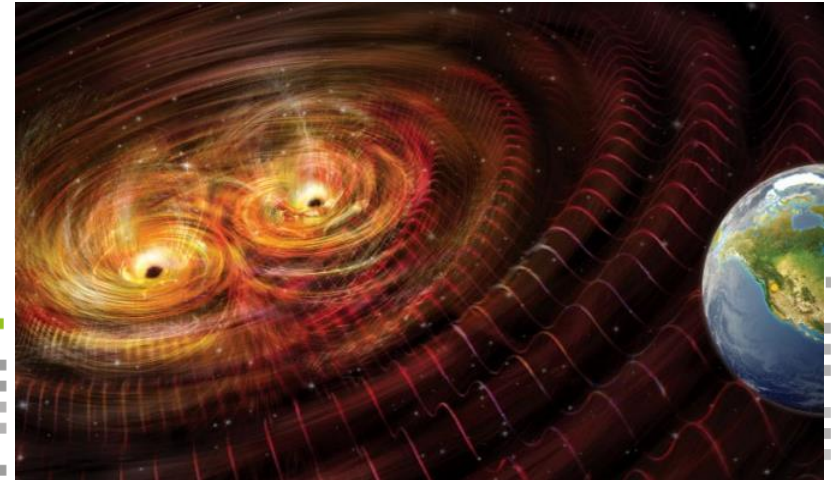


leti
cea tech



LAPP



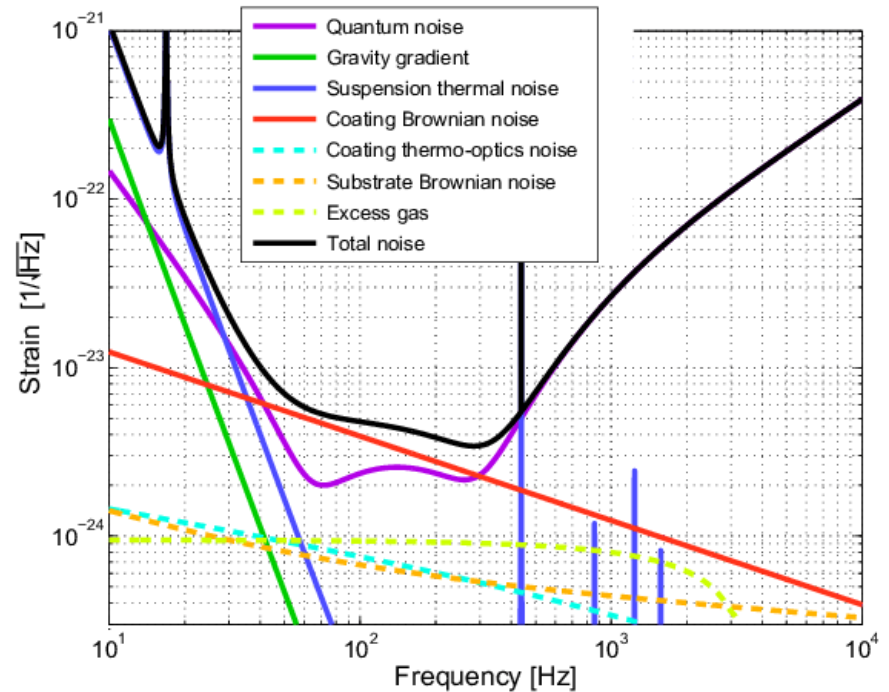
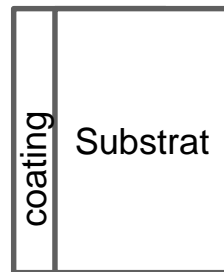
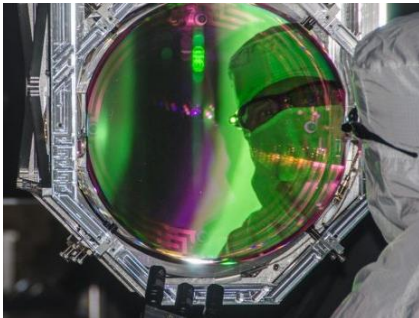
TRANSFER OF AN ALGAAS/GAAS MULTILAYER COATING ON A SILICA SUBSTRATE

GdR Ondes Gravitationnelles| Hui Victor| 12/10/21

Supervisors :
Christophe Dubarry (CEA LETI)
Raffaele Flaminio (LAPP)

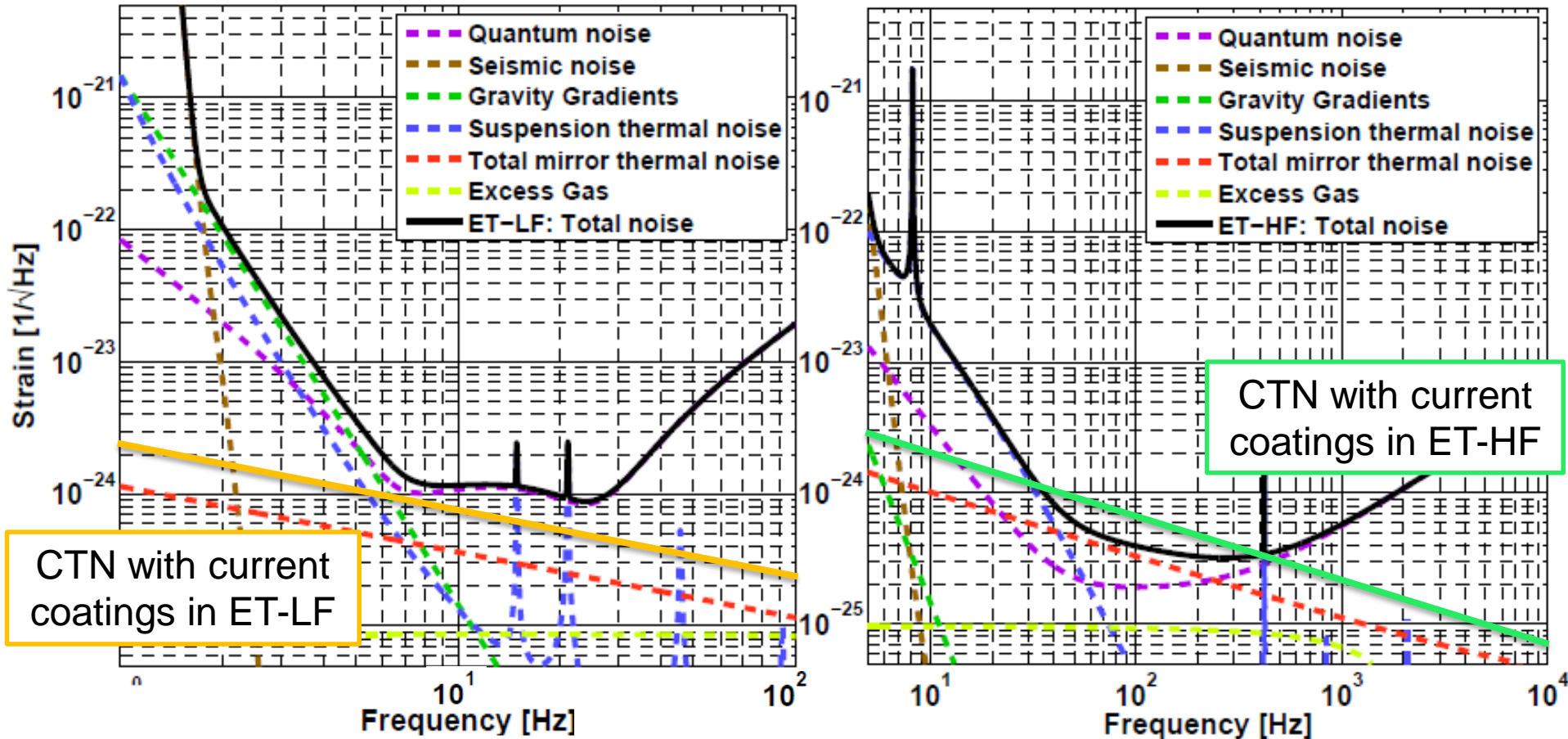
MOTIVATION

- One of the main limitation of the sensitivity of GW detectors is the thermal noises on the interferometer's mirrors
- Two main thermal noises in the coating:
 - Brownian noise
 - Thermo-optic noise



MOTIVATION

- Comparison of coating thermal noise in ET with current coatings



Need to reduce thermal noise to enhance sensitivity

Credit : Design report update 2020 for ET

- **Brownian noise** : due to mechanical damping of the coating layers driven by internal losses

$$S_B(f, T) \propto \frac{k_B T}{\omega} \phi$$

Loss angle

- **Thermo-optic noise** : contribution of Thermo-elastic noise and Thermorefractive noise

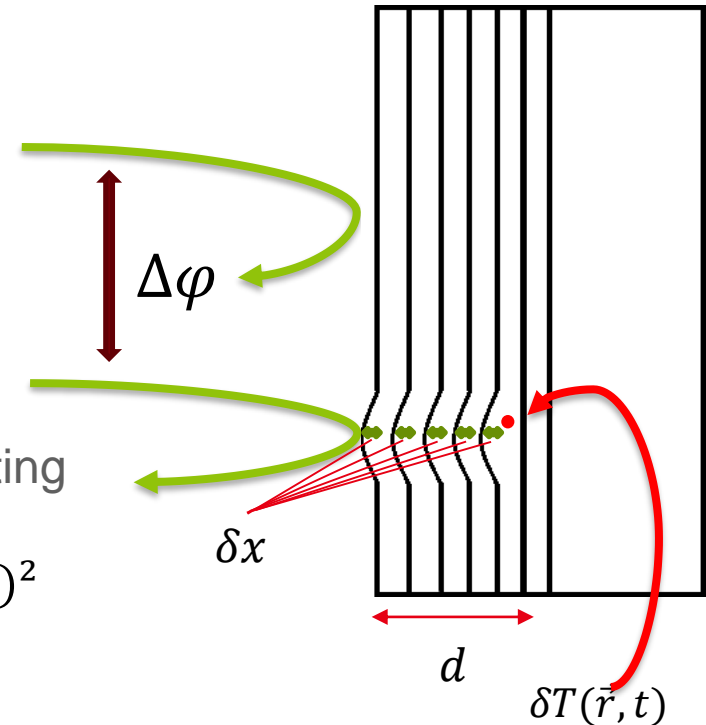
- Thermo-elastic noise : apparent expansion of the mirror coating causing phase shift in the reflected beam
- Thermorefractive noise: physical change of the coating layers size and change in refractive index with temperature in the coating

$$S_{TE} \propto \frac{k_B T^2}{\sqrt{\omega}} (\alpha_c d)^2$$

α_c : coating coefficient of thermal expansion

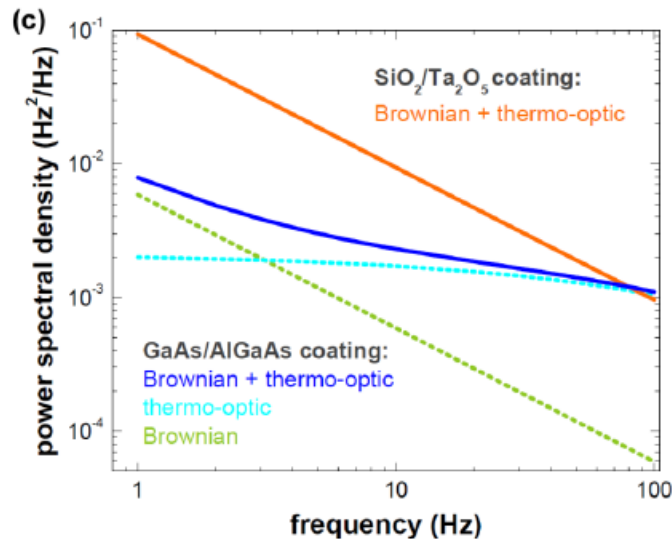
$$S_{TR} \propto \frac{k_B T^2}{\sqrt{\omega}} (\beta \lambda)^2$$

β : coefficient of thermorefraction



STATE OF ART THERMAL NOISE REDUCTION

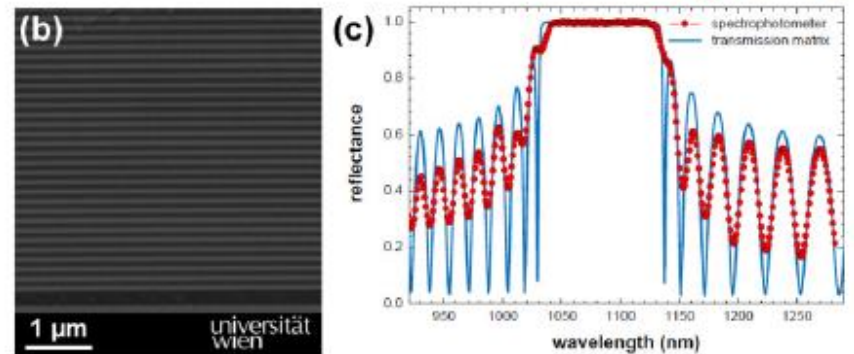
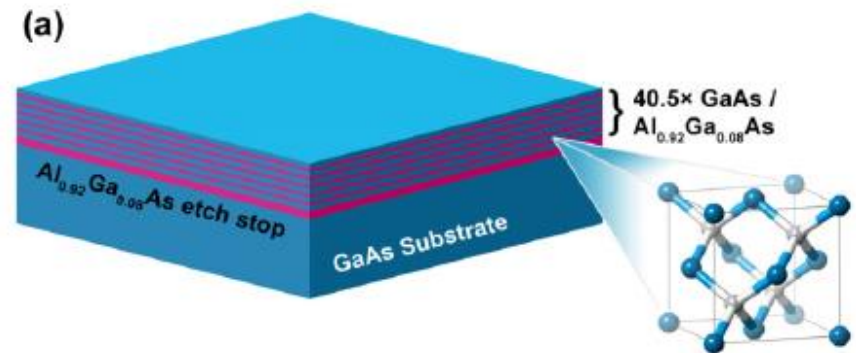
- **Coatings used until now: amorphous**
 - Ta_2O_5/SiO_2
 - Made at LMA (Laboratoire des Matériaux Avancés)
- **Thermal noise reduction**
 - Cryogenic environment
 - Development of new coatings => Crystalline coatings



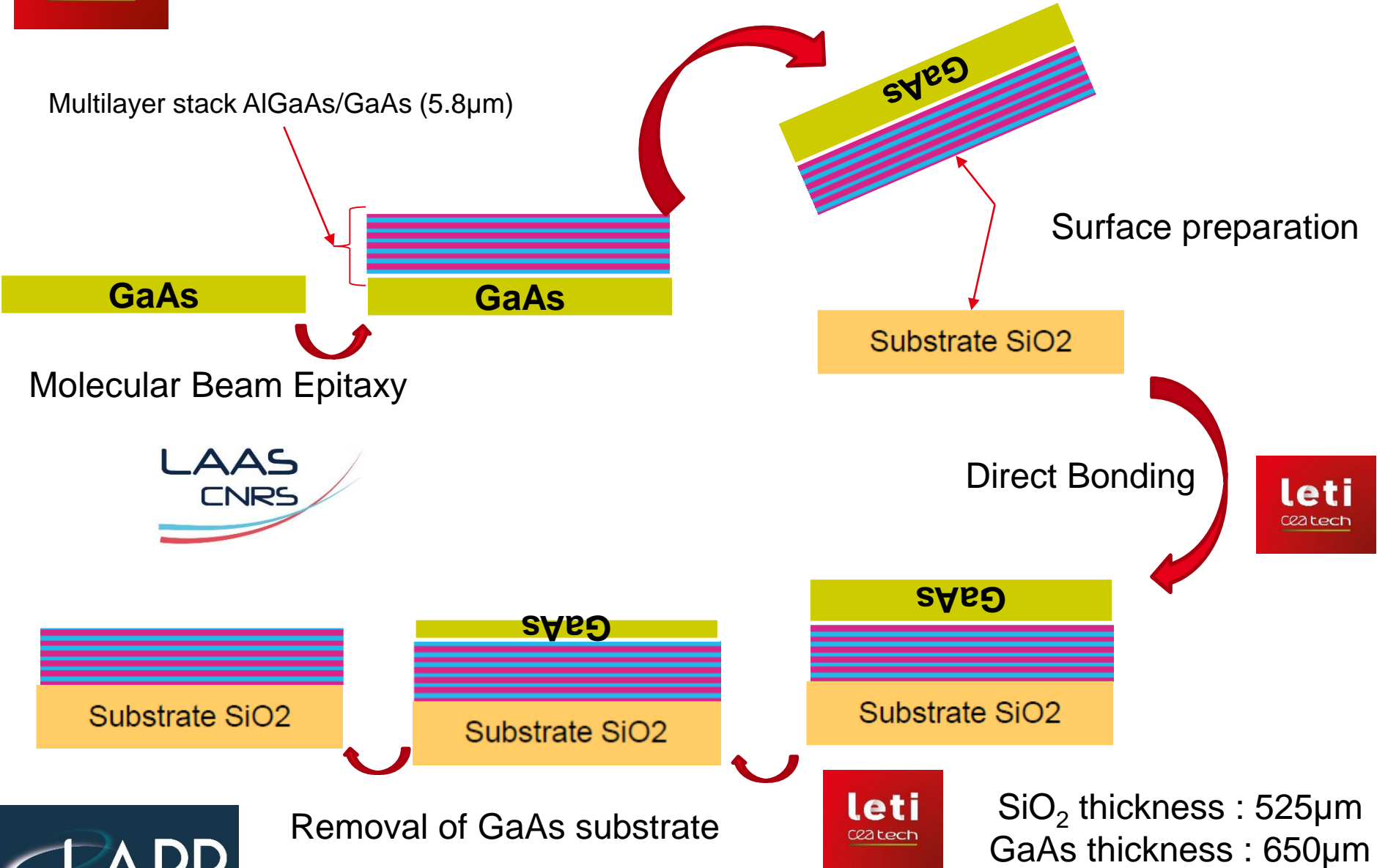
Crystalline coatings : Reduction
x10 on Thermal noise

G. Cole et al. 2013

- **G. Cole et al. 2013**
Tenfold reduction of Brownian noise in optical interferometry
 - Crystalline Bragg mirror (AlGaAs/GaAs)
 - Excellent optical performances ($R > 0.9998$, 4ppm diffusion)
 - Very low mechanical losses
=> Quality factor $Q > 4 \cdot 10^4$ ($Q \sim 1/\phi$)
 $Q(\text{Ta}_2\text{O}_5/\text{SiO}_2) \sim \text{a few } 10^3$



GENERAL PROCESS



BONDING TECHNIQUE

Surface conditioning

(cleaning, roughness, binding states, etc.)

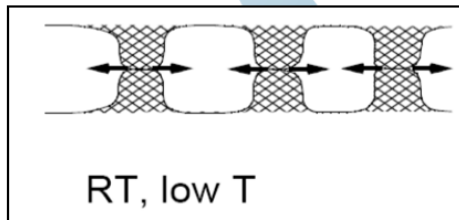
- Roughness of ~0.3 nm RMS
- No particule
- No chemical contamination



Direct Bonding

Room temperature and atmospheric pressure bonding:
Van de Waals adhesion

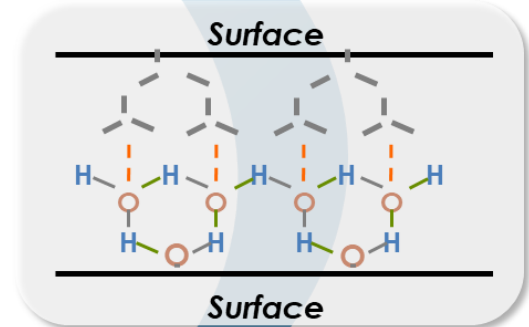
Ziplock model



Fr. Rieutord et al., ECS 2006

Annealing

Binding states evolution upon annealing: from Van der Waals to covalent bonding



- **Heterostructure bonding**

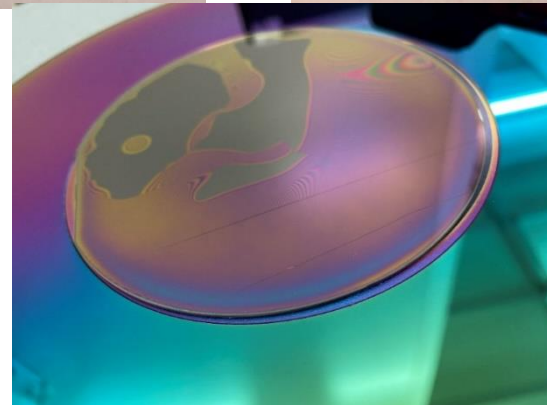
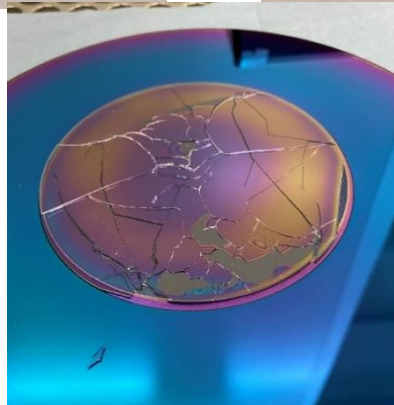
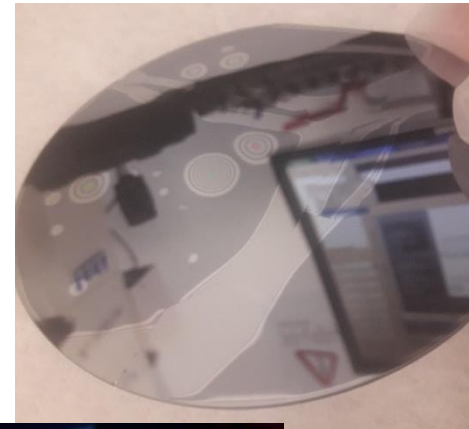
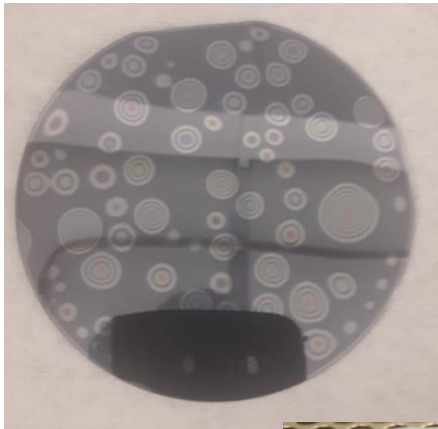
- GaAs naturally oxidize => native oxide layer GaAsO



- Direct bonding difficult : Difference of thermo-mechanic properties between GaAs and SiO₂ (large mismatch of Coefficient of Thermal expansion **CTE(GaAs)=5.3e-6; CTE(SiO₂)=5.1e-7**)
- Thermal stress depends on wafer thickness (the more wafers are thick, the more stresses are important) => Difficult to anneal after direct bonding

MAIN CHALLENGES

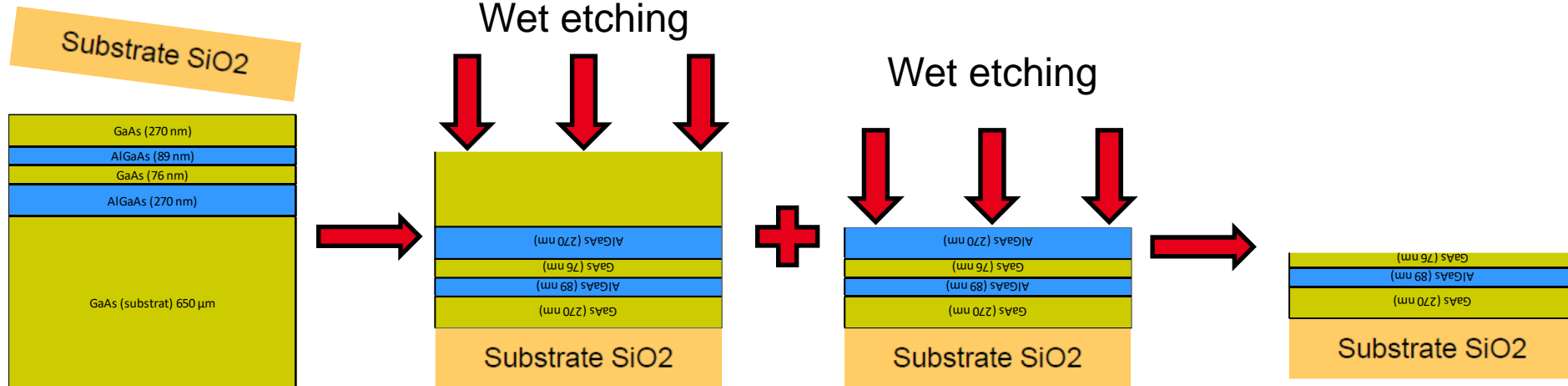
- Heterostructure bonding
 - Results after direct bonding + annealing 100°C



Local debonding + breaking of the heterostructure due to material expansion (CTE(GaAs)= $5.3e-6$; CTE(SiO₂)= $5.1e-7$)

MAIN CHALLENGES

- **Removal of GaAs substrate**
 - Need to stop properly on the multilayer
 - Selective chemical solution that etches GaAs and not AlGaAs
 - Selective chemical solution that etches AlGaAs and not GaAs

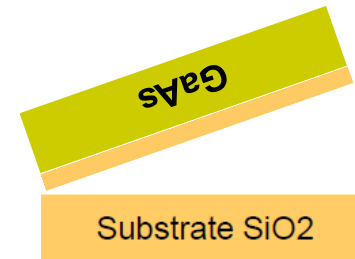


- **Heterostructure bonding**

- Wet cleaning : Removal of GaAs native oxide
 - Cleaning with NH₄OH solution
- Deposition SiO₂ by PECVD : Removal of GaAs native oxide + improvement of bonding quality (Oxide-Oxide direct bonding)
 - Preclean with plasma
 - Deposit of 500nm of SiO₂



PECVD



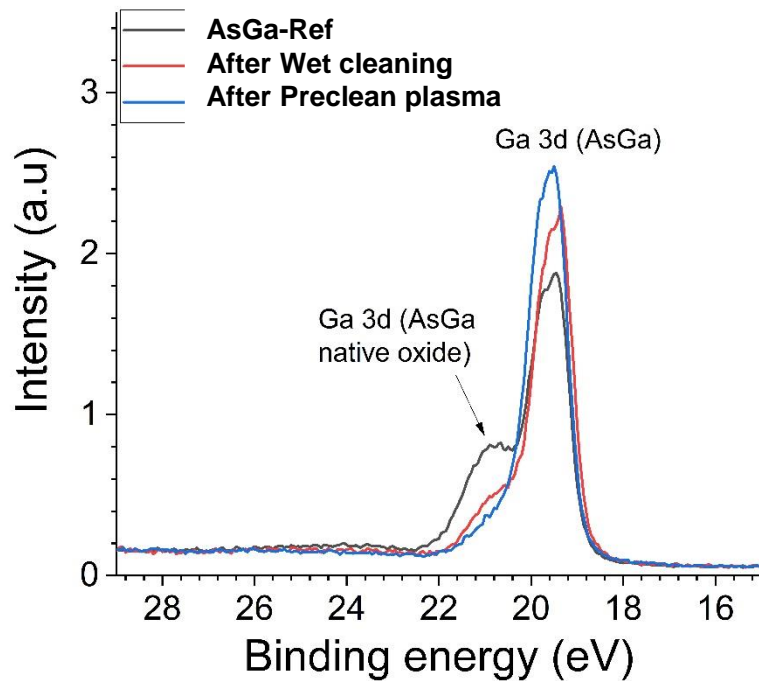
Direct bonding

- CMP (Chemico-Mechanical Polishing) => improve roughness/micro-topography
- Grinding of GaAs (physical thinning of GaAs before annealing)

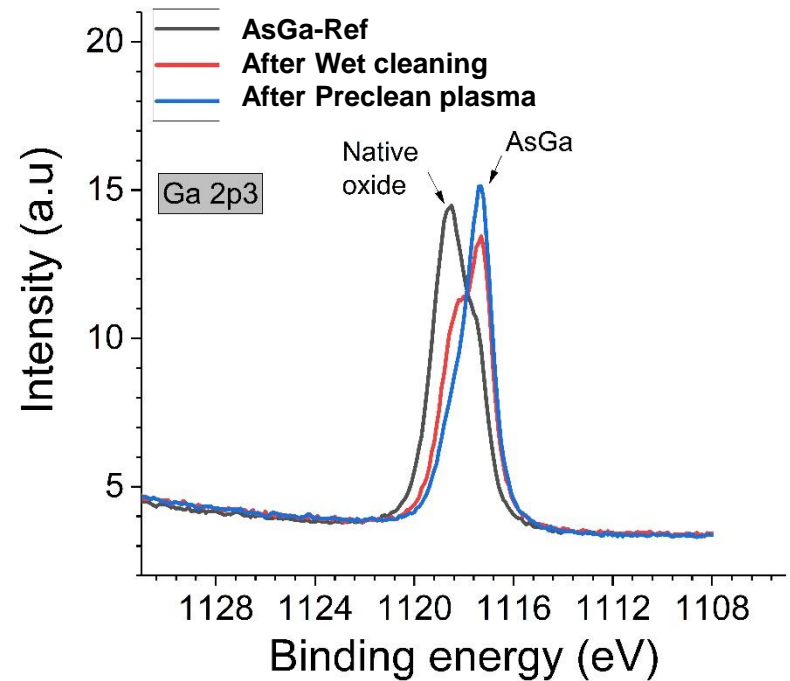
- **Surface treatment**
 - Wet Cleaning of GaAs wafers
 - Plasma treatment + SiO₂ deposition (500nm)
- **Direct bonding**
 - Chemical-Mechanical Polishing on SiO₂ wafers
 - Chemical-Mechanical Polishing on GaAs wafers after deposition
 - Direct bonding
- **Removal of GaAs**
 - Grinding of GaAs (until 100μm thickness)
 - Chemical wet etching of GaAs (until 20μm thickness)
- **Annealing 2h @100°C**

- X-Ray photoelectron spectrometry on GaAs

Ga 3d : depth of analysis ~ 7 nm



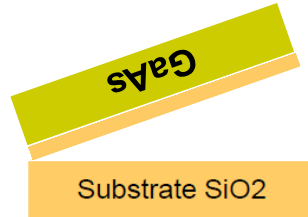
Ga 2p : depth of analysis ~ 2 nm



Treatment of native oxide validated

CHARACTERIZATION OF ROUGHNESS : POLISHING EFFECT

- AFM (Atomic force microscopy) on substrate



SiO2

		Before CMP 10µmX10µm	After CMP 10µmX10µm
centre	RMS, nm	0,622	0.26
Mi-rayon	RMS, nm	0,852	0.26
bord	RMS, nm	0,737	0.28

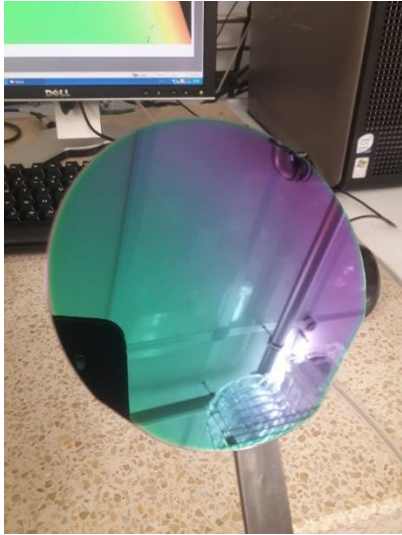
Ok for direct bonding

GaAs with SiO2 deposit

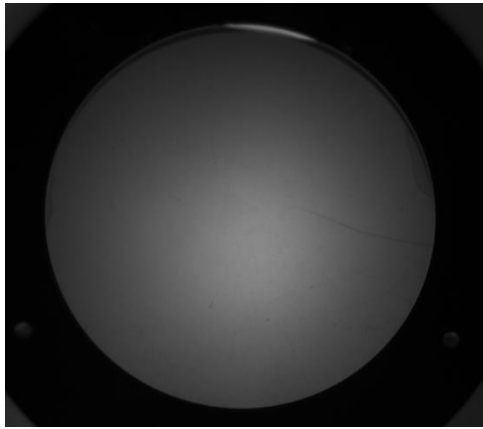
		Before CMP 10µmX10µm	After CMP 10µmX10µm
Centre	RMS,nm	0,521	0,215
Mi-rayon	RMS,nm	0,600	0,212
Bord	RMS,nm	0,579	0,198

Ok for direct bonding

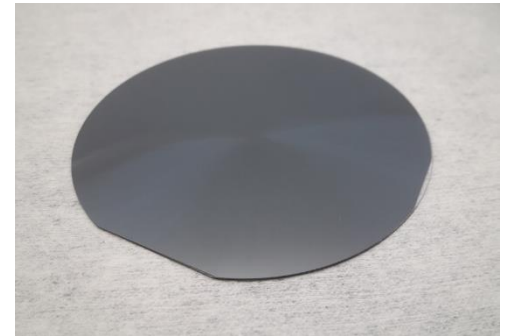
MAIN RESULTS



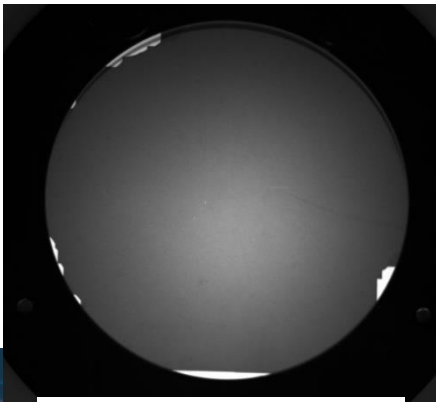
Preclean + SiO2 deposit



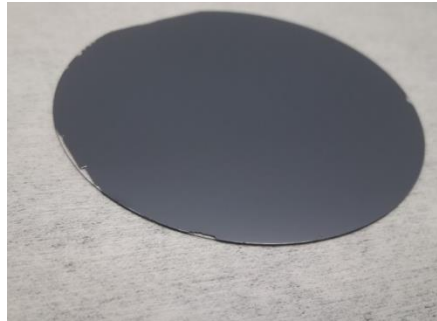
Direct bonding



Grinding GaAs (650μ->100μ)



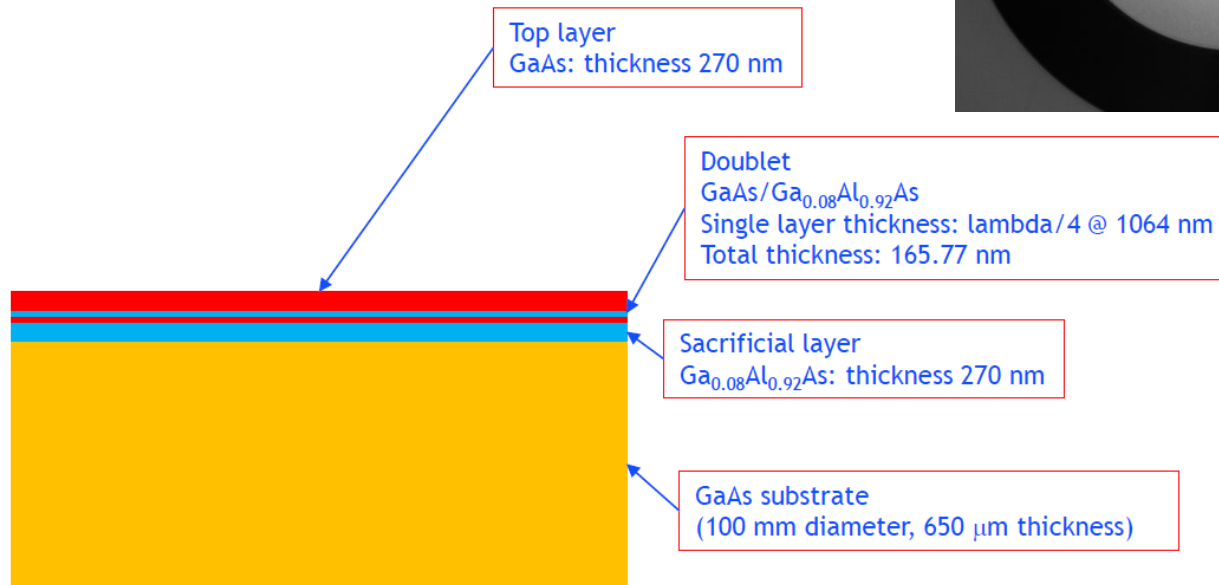
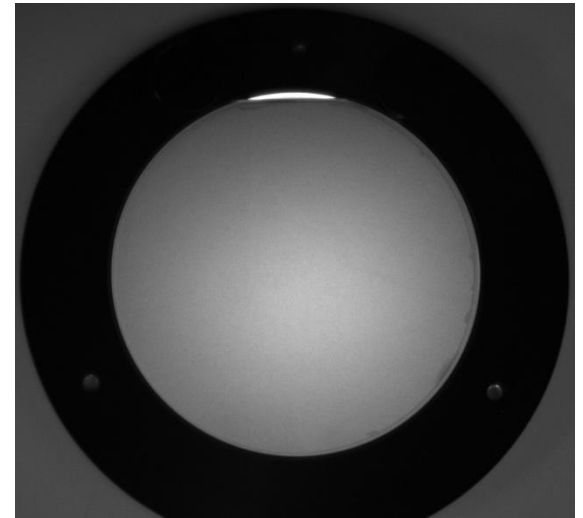
Final annealing



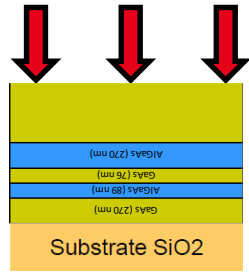
Etching GaAs (100μ->20μ)

TRANSFER OF A DOUBLET

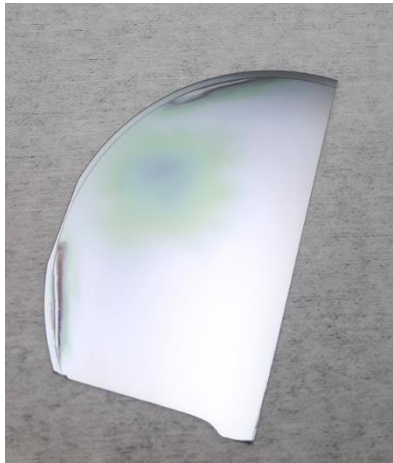
- Process applied to the doublet
 - Cleaning, PECVD SiO₂, CMP, Bonding



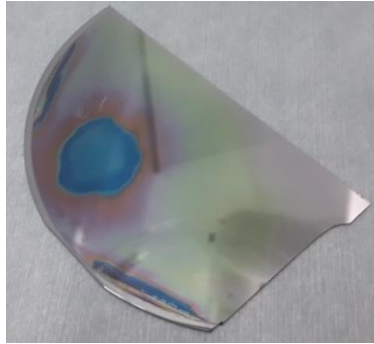
HARD POINT : WET ETCHING



- Removal of GaAs substrate with $\text{NH}_4\text{OH}/\text{H}_2\text{O}_2/\text{H}_2\text{O}$
 - Results after wet etching tests



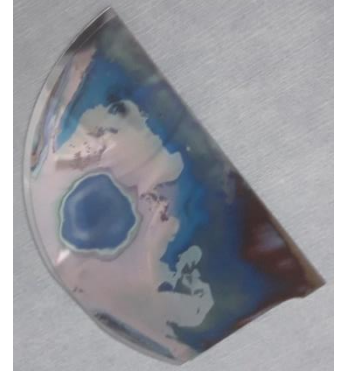
Etching t+75s



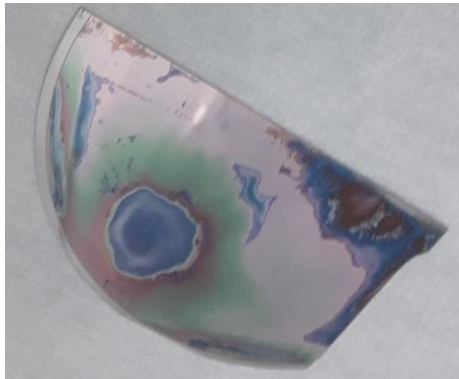
Etching t+90s



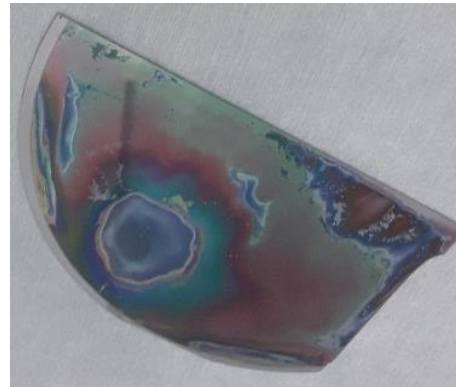
Etching t+105s



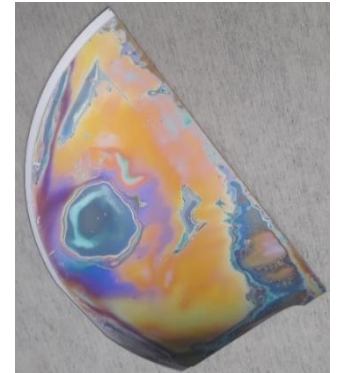
Etching t+120s



Etching t+135s



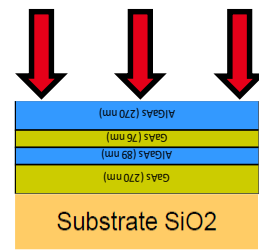
Etching t+150s



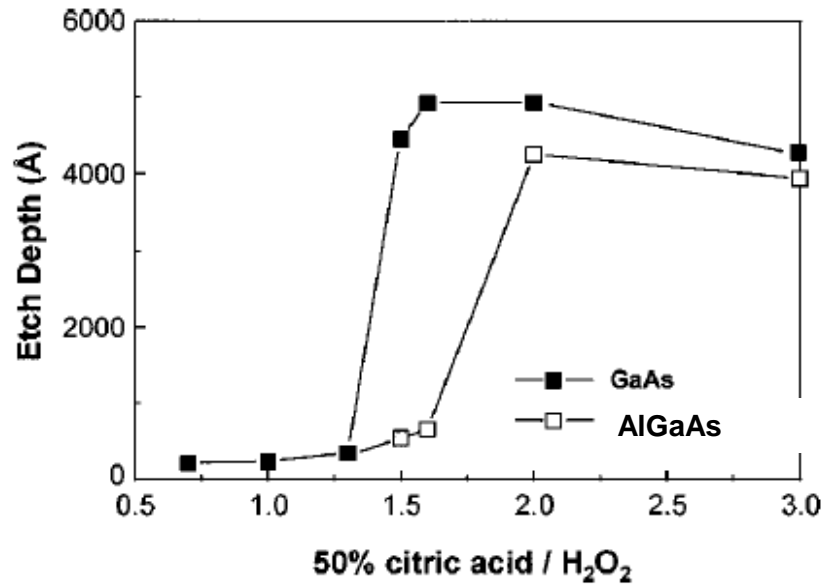
Etching t+195s

Non homogeneous etching : Etching chemistry non selective on GaAs over AlGaAs

WET ETCHING GAAS/ALGAAS



- NH₄OH/H₂O₂ not selective for wet etching of GaAs over AlGaAs
- Possible solution :
 - Citric Acid/H₂O₂



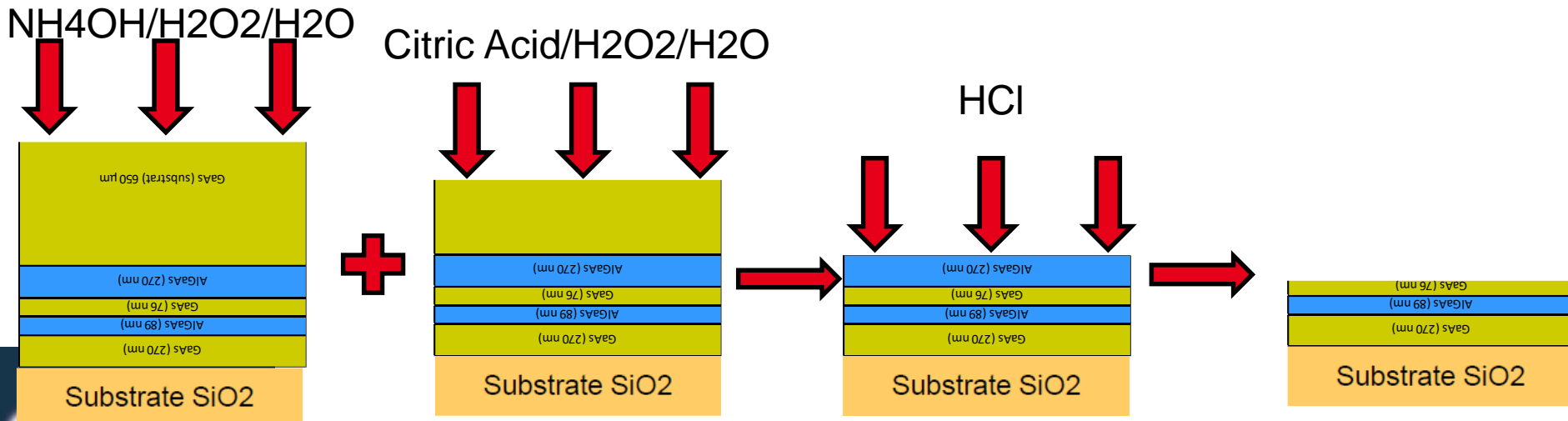
Eun-A Moon et al. (1998)

doi.org/10.1063/1.368571

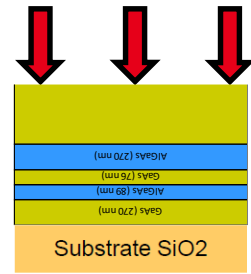
• Process

- HCl etching (de-oxidation)
- NH₄OH/H₂O₂/H₂O etching
- Citric Acid/H₂O₂/H₂O etching
- HCl etching

Courtesy to Isabelle Sagnes for improvement process compatible with multilayer design



ELLIPSOMETRY CHARACTERIZATION



- Wet etch AC/H2O2



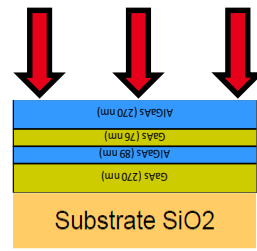
Theoretical stack

AlGaAs	270 nm
GaAs	76,4 nm
AlGaAs	89,3 nm
GaAs	270 nm
SiO2	525 μm

Measured stack

AlGaAs : 257,391 nm
GaAs : 78,284 nm
AlGaAs : 89,657 nm
GaAs : 273.888 nm
SiO2: 0.52 μm

ELLIPSOMETRY CHARACTERIZATION



- Wet etch HCl 10%



Doublet transfer successful!

Theroretical stack

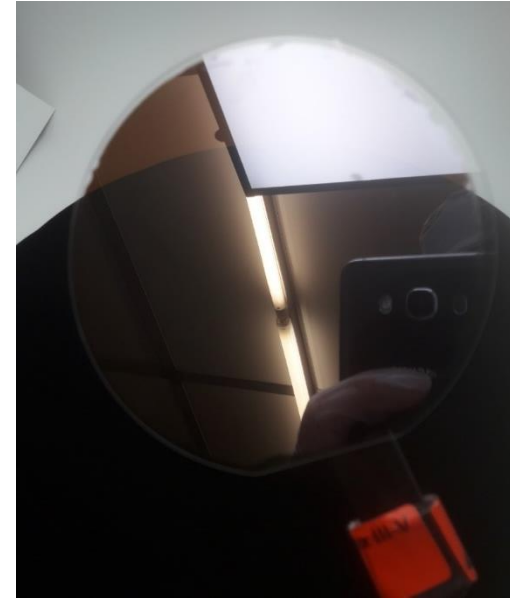
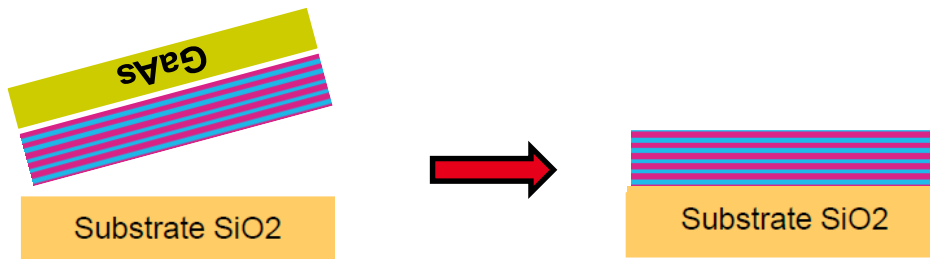
GaAs	76,4 nm
AlGaAs	89,3 nm
GaAs	270 nm
SiO2	525 μm

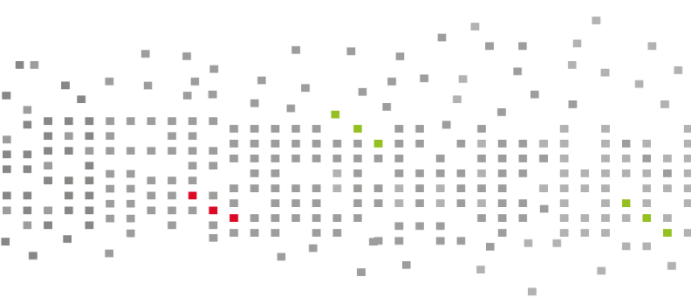
Measured stack

GaAs : 77.863nm
GaAs : 89,083 nm
GaAs : 274.402 nm
SiO2 : 0.52 μm

CONCLUSION

- Development of a process to transfer GaAs on SiO₂ for 100mm substrates
- Complete validation of each step of the process
- Doublet successfully transferred on SiO₂
- Next step : Applying the process to the transfer of the final multilayer

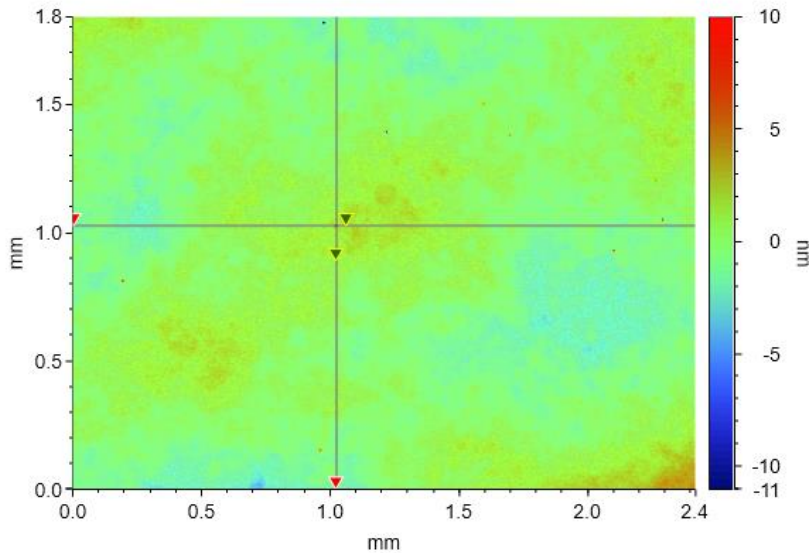




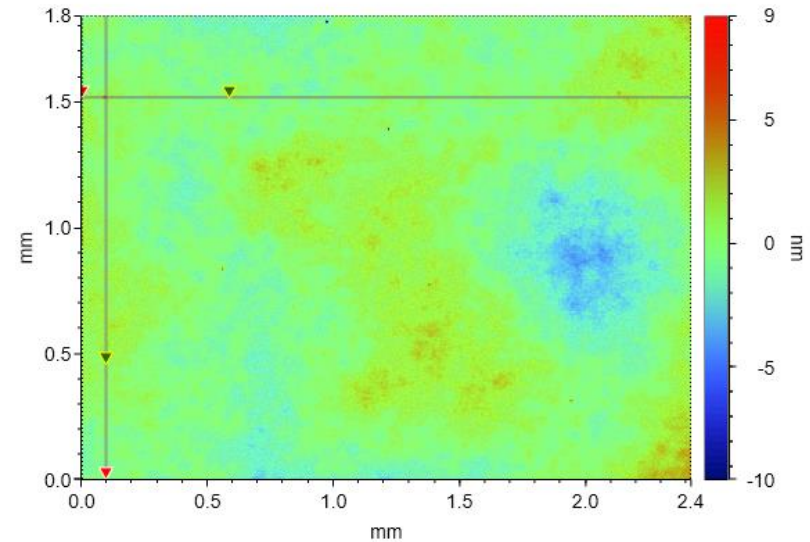
MERCI POUR VOTRE ATTENTION!

THANK YOU FOR YOUR ATTENTION!

- Current Multilayers



A1195



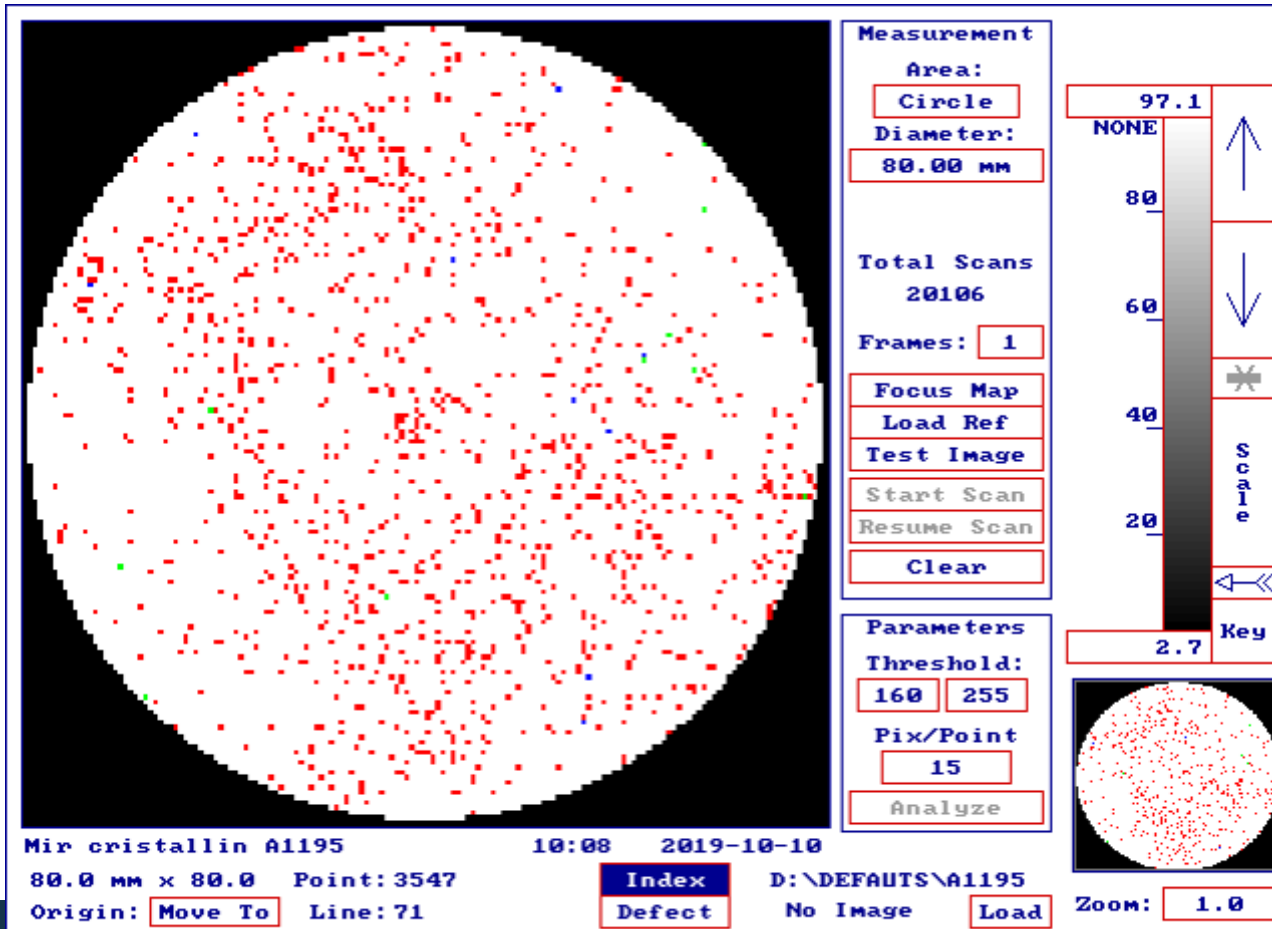
A1197

	A1195	A1197
RMS,nm	1,03	1,1
Nb defects	~8	~6
Nb defects/mm ²	1,8	1,4

Better surface quality than in 2018 but planarization necessary for direct bonding (RMS>0.2nm) => CMP

INPUT CHARACTERIZATION ON MULTILAYERS AT LMA : DEFECTS MAP

- Defects map of A1195

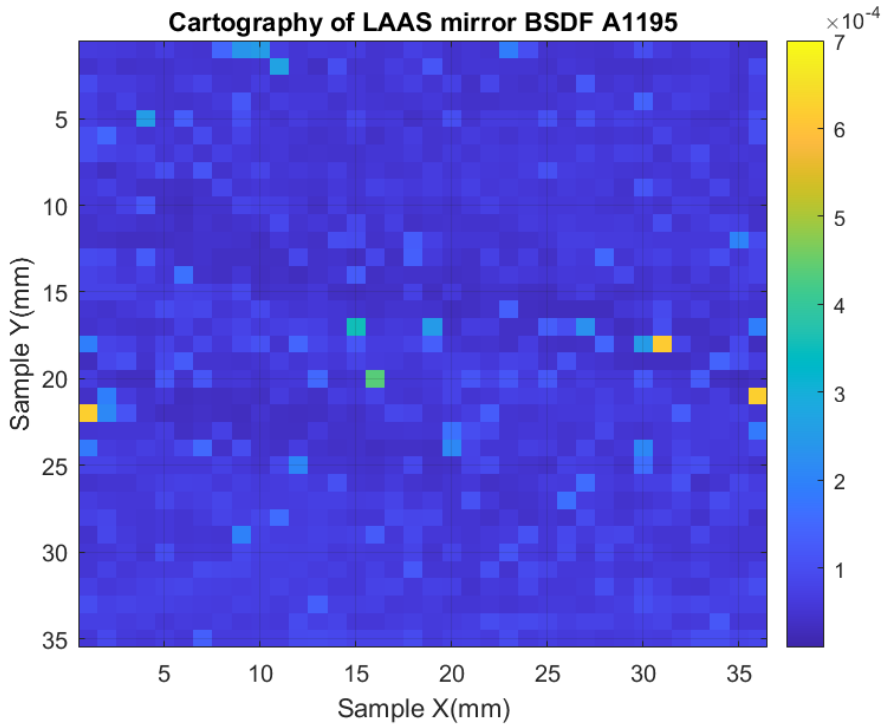


Nb tot of defects :	3618
Nb defects/mm ²	1,4

Match characterization
with interferometric
microscopy

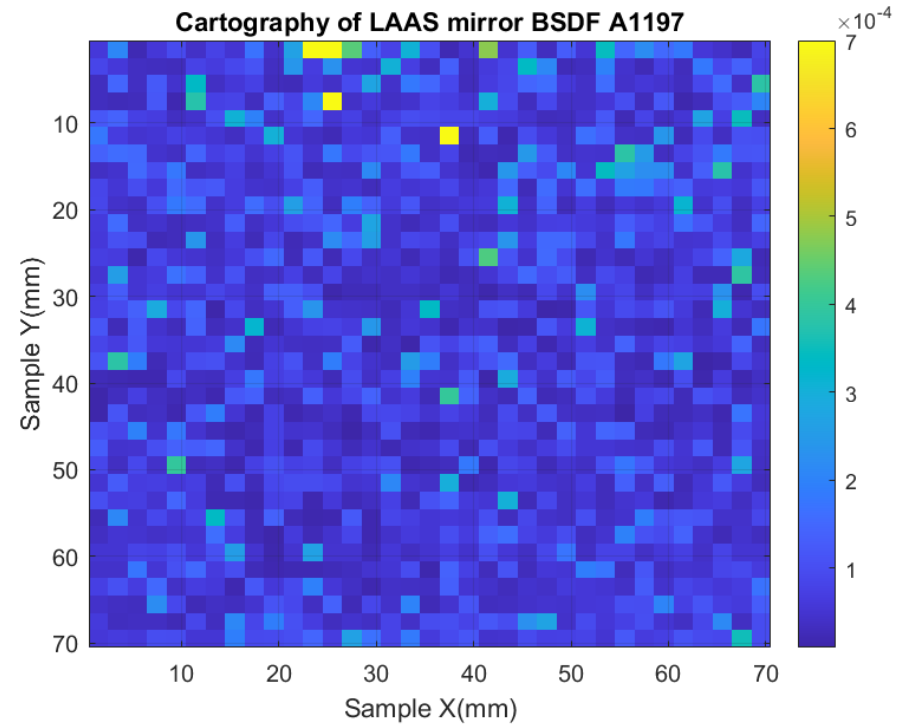
INPUT CHARACTERIZATION ON MULTILAYERS AT LMA : DIFFUSION MAP

Cartography of LAAS mirror BSDF A1195



Mean diffusion : 45 ppm

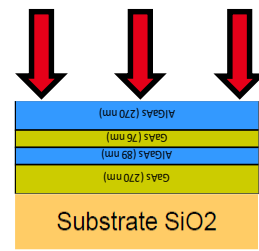
Cartography of LAAS mirror BSDF A1197



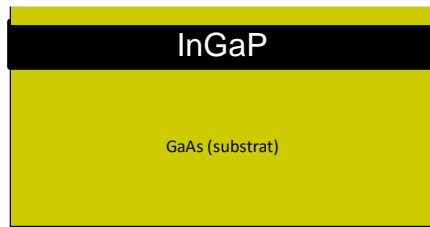
Mean diffusion : 60 ppm

Virgo mirrors : 5-10 ppm

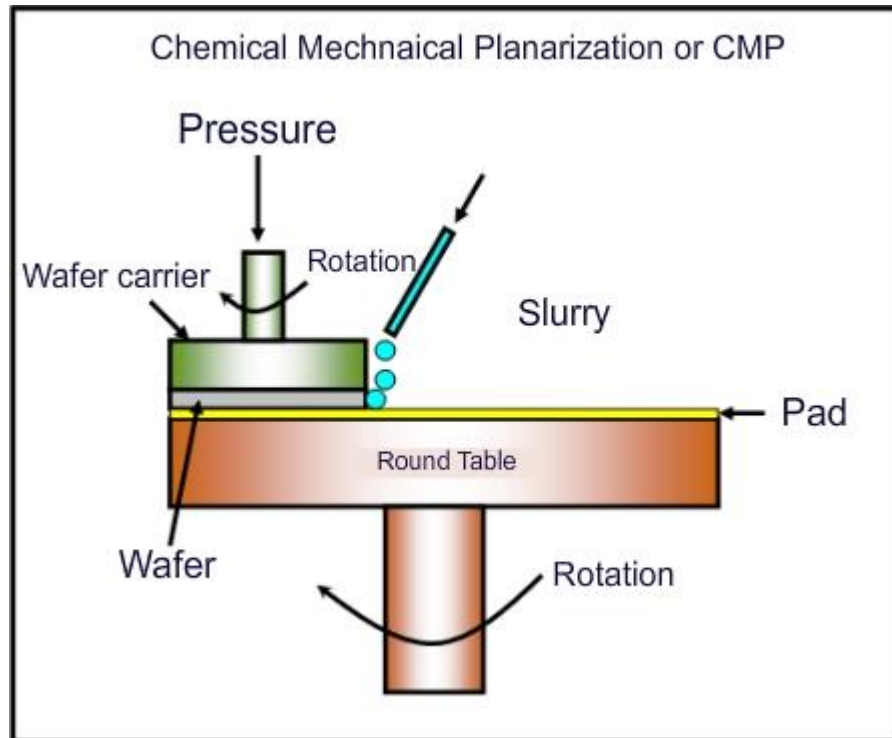
WET ETCHING GAAS/ALGAAS

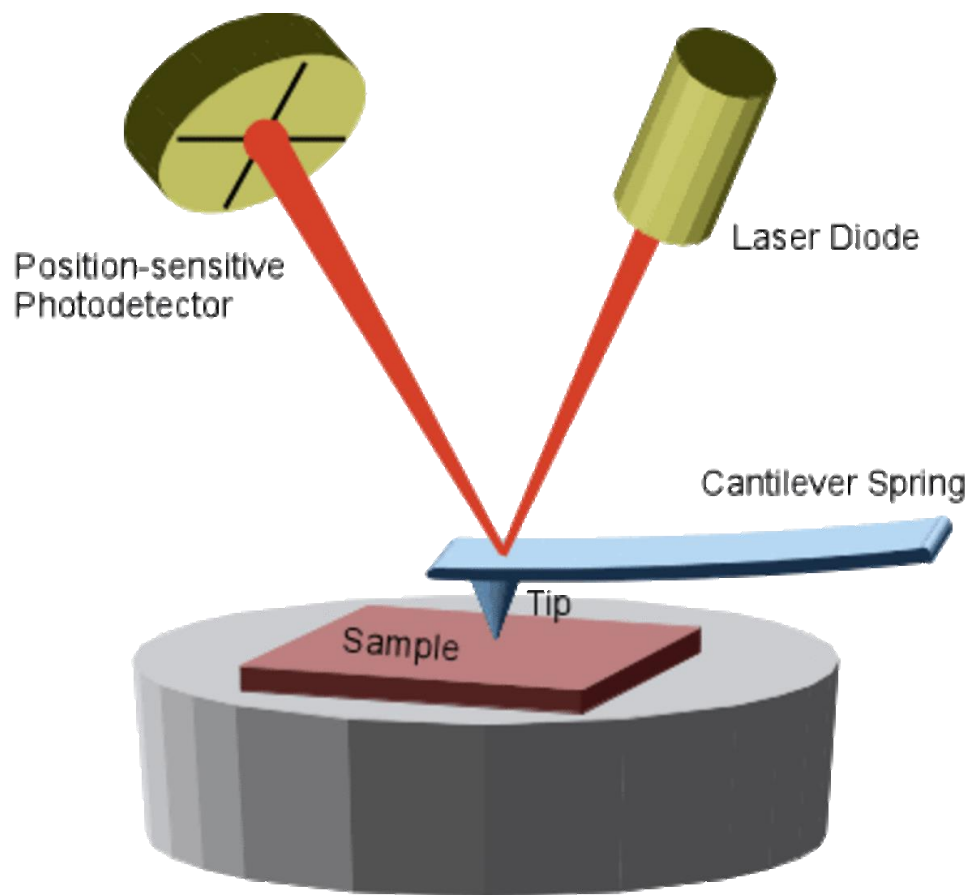


- NH4OH/H2O2 not selective for wet etching of GaAs over AlGaAs
- Possible solutions
 - Indium based etch stop layer (InGaP)
 - Selective etching of GaAs over InGaP well known at CEA (H3PO4/HCl)

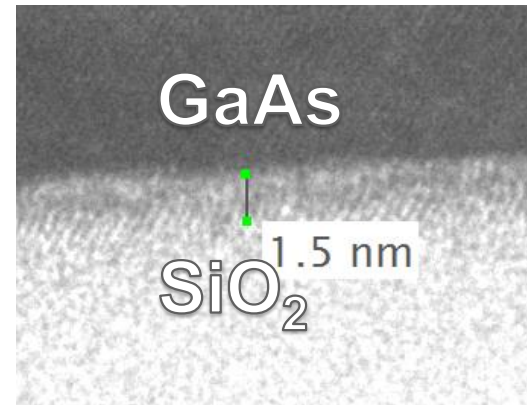
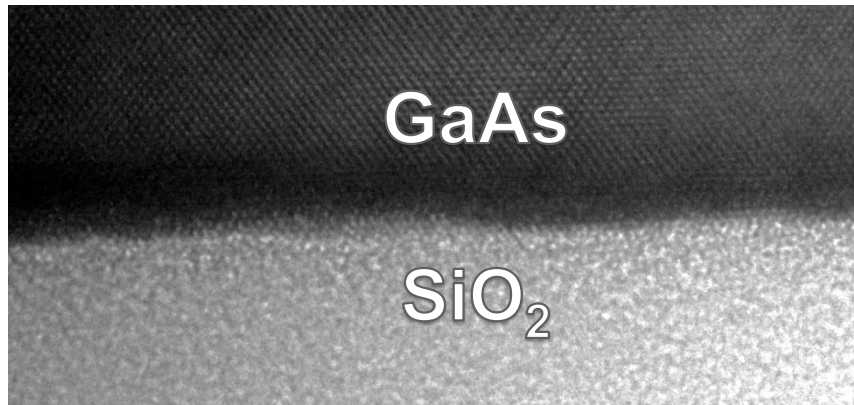
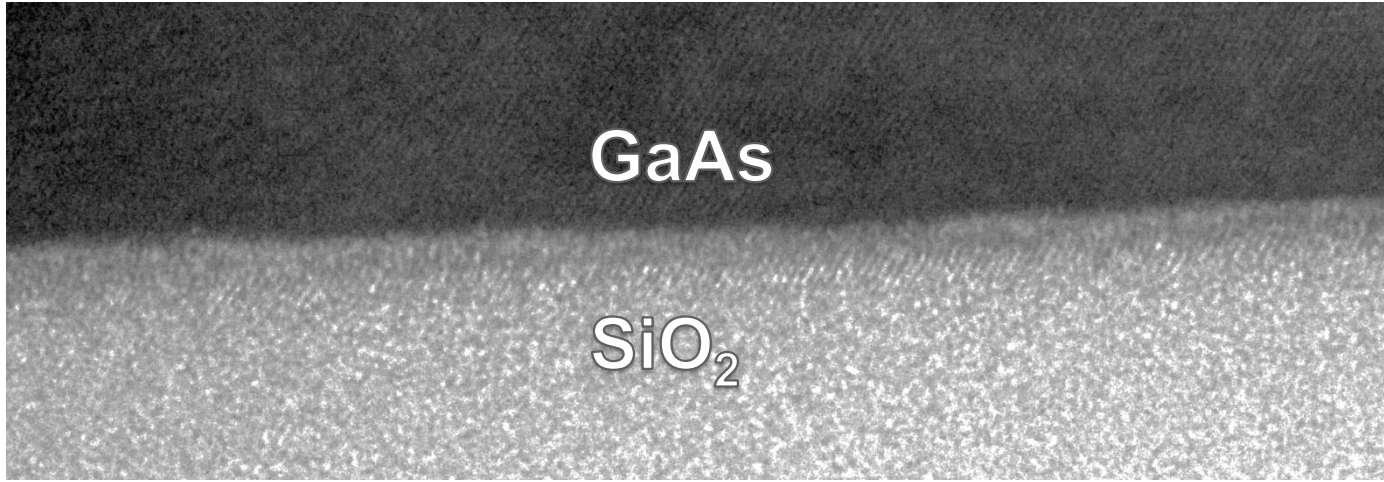


InGaP layer (70nm + GaAs cap layer (20nm)



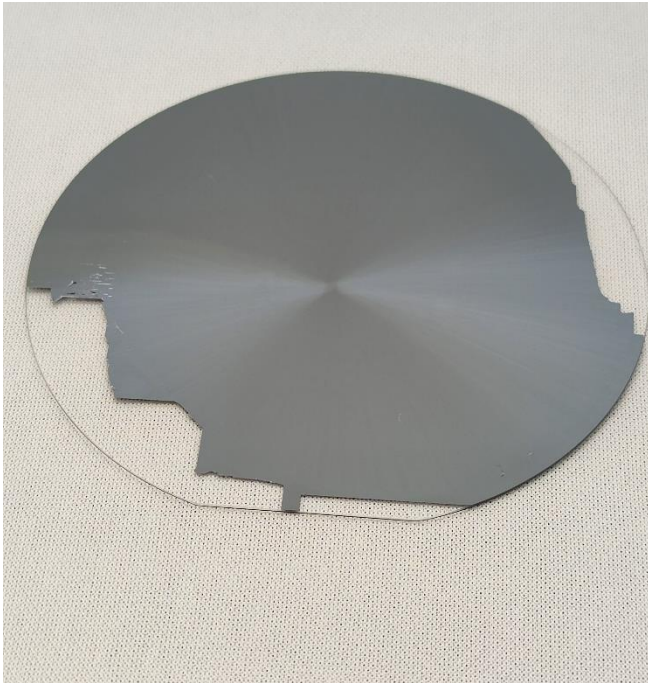


TEM CHARACTERIZATION OF BONDING INTERFACE

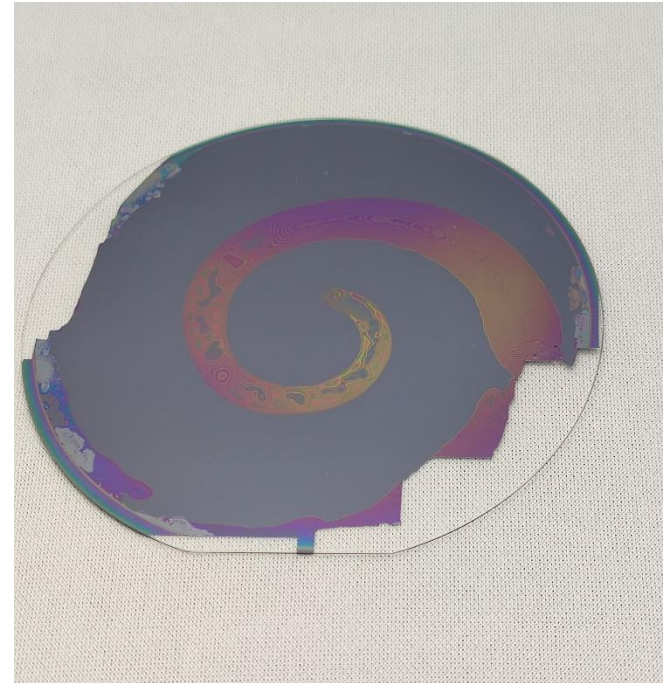


Heterostructure reconstruction OK

- Grinding multilayer (A1195)

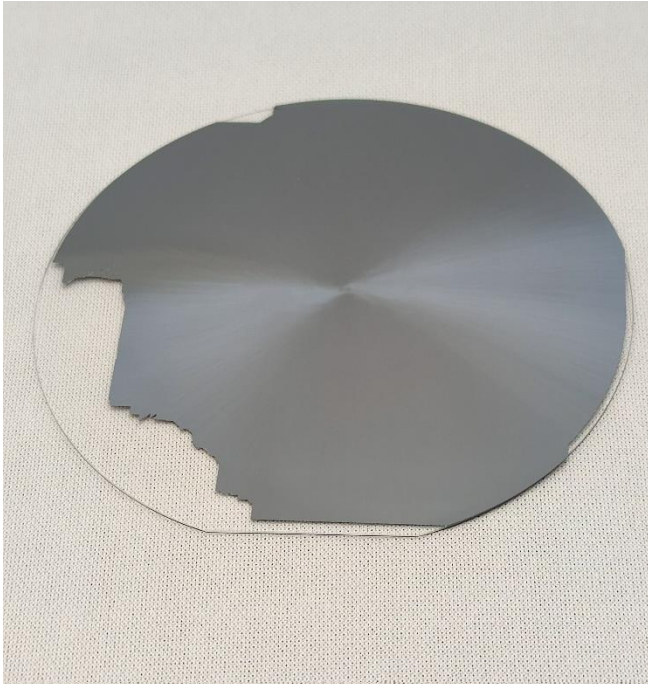


P01 Front face (GaAs)

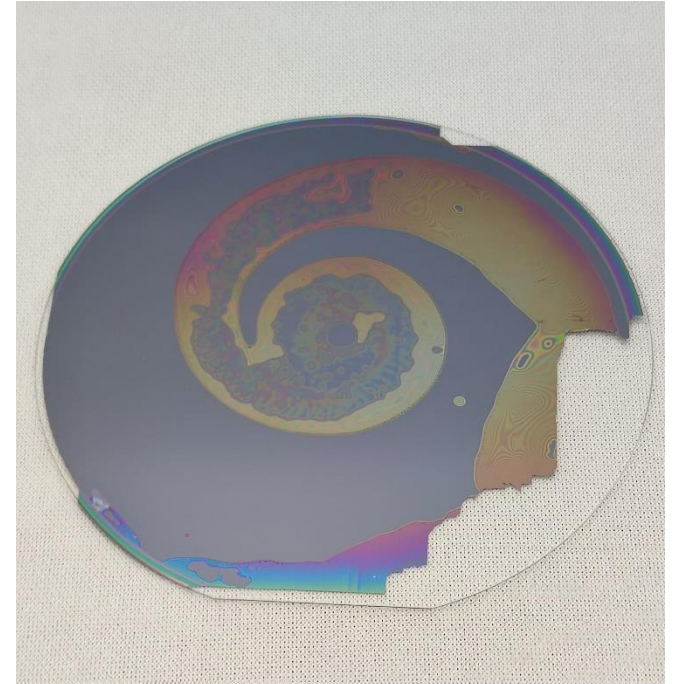


P01 Back face (SiO2)

- Grinding multilayer (A1197)

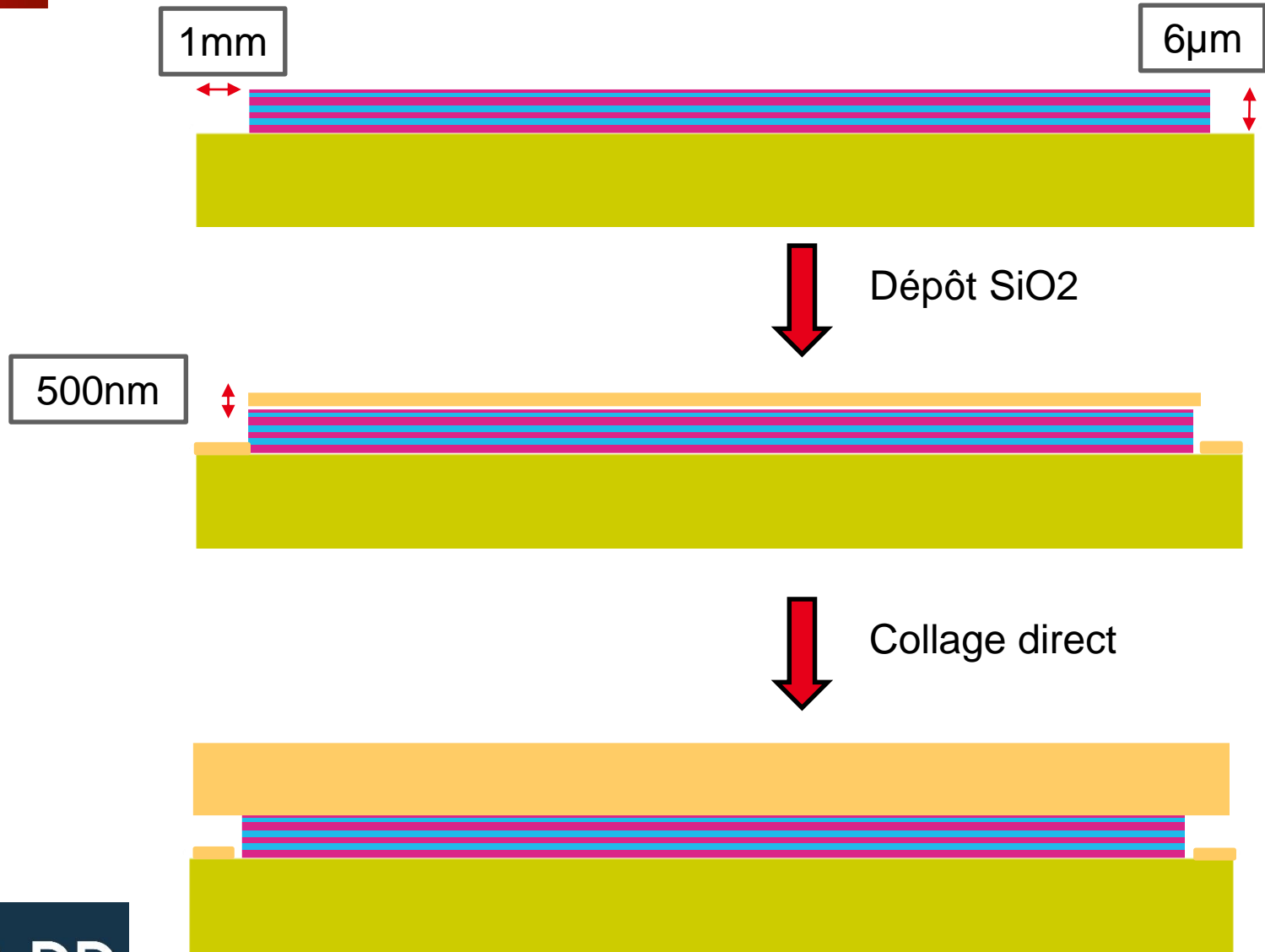


P02 Face avant (côté substrat GaAs)

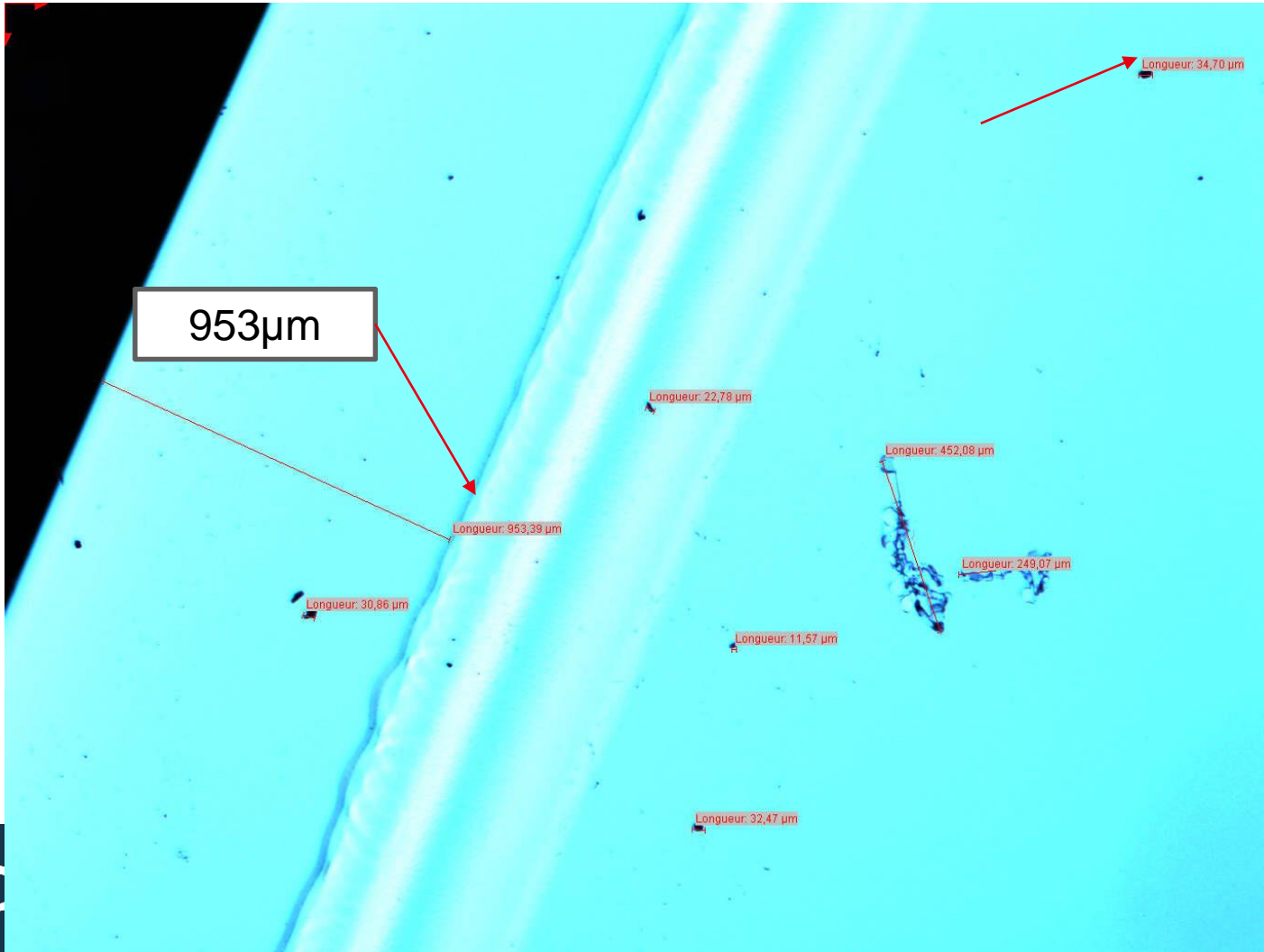


P02 Face Arrière (côté SiO2)

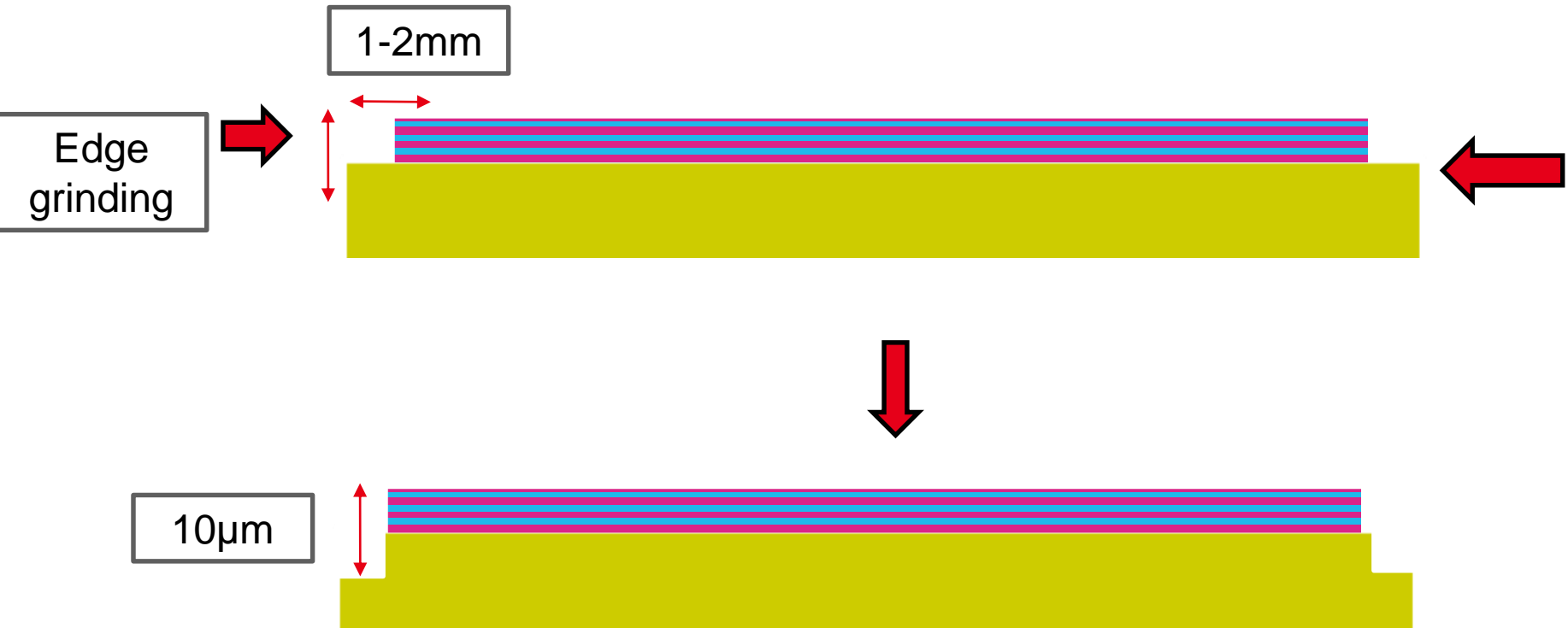
EXPLANATION : EDGE TOPOLOGY



- A1195 bord x2,5



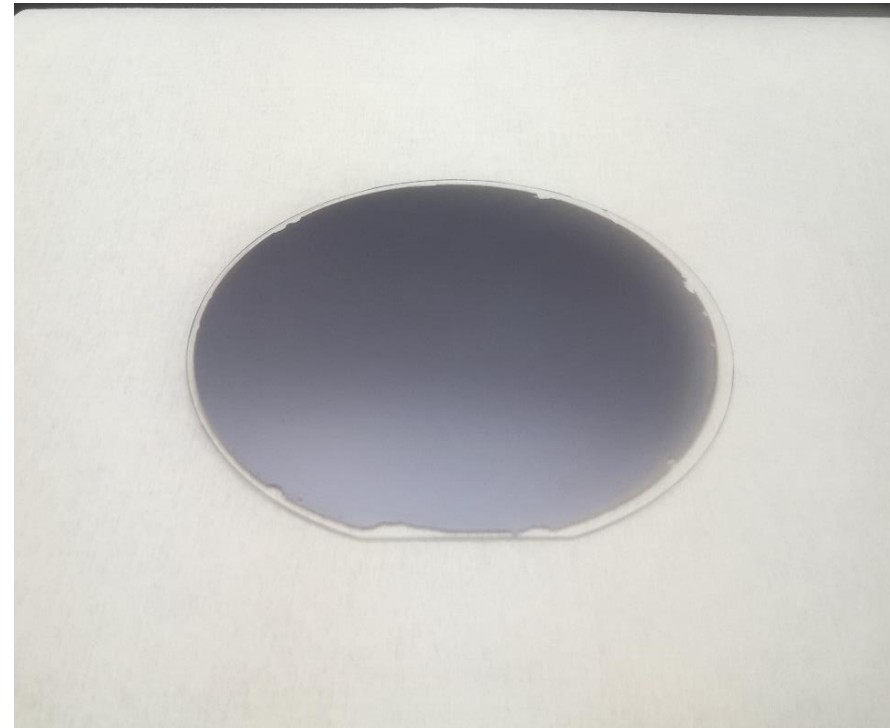
SOLUTION : EDGE GRINDING



- Gravure 2x30s

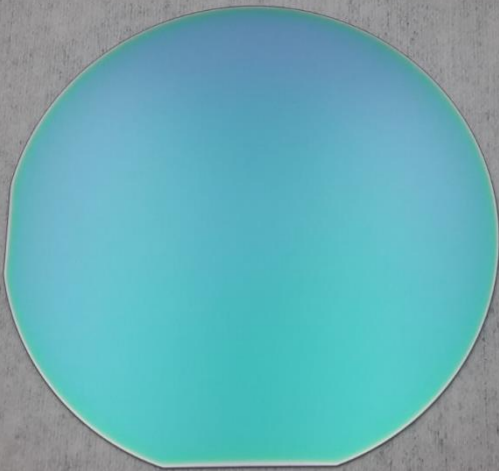


Après 30s de gravure

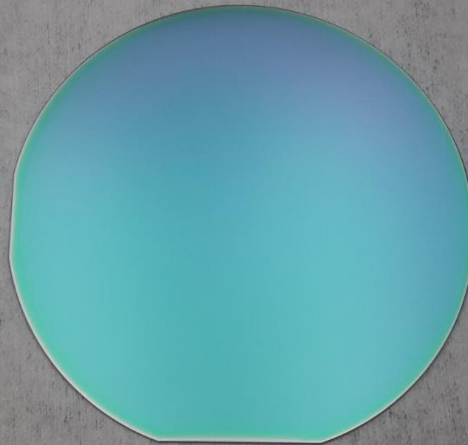


Après 60s de gravure

- Photos lumière rasante



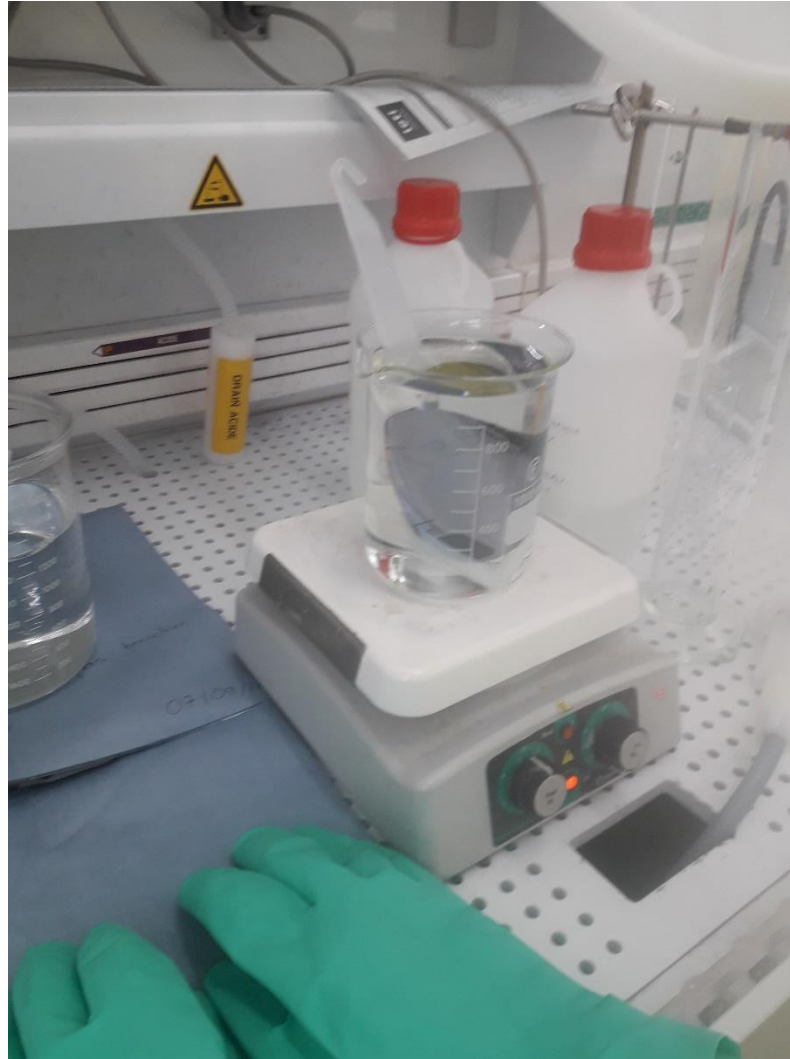
P01



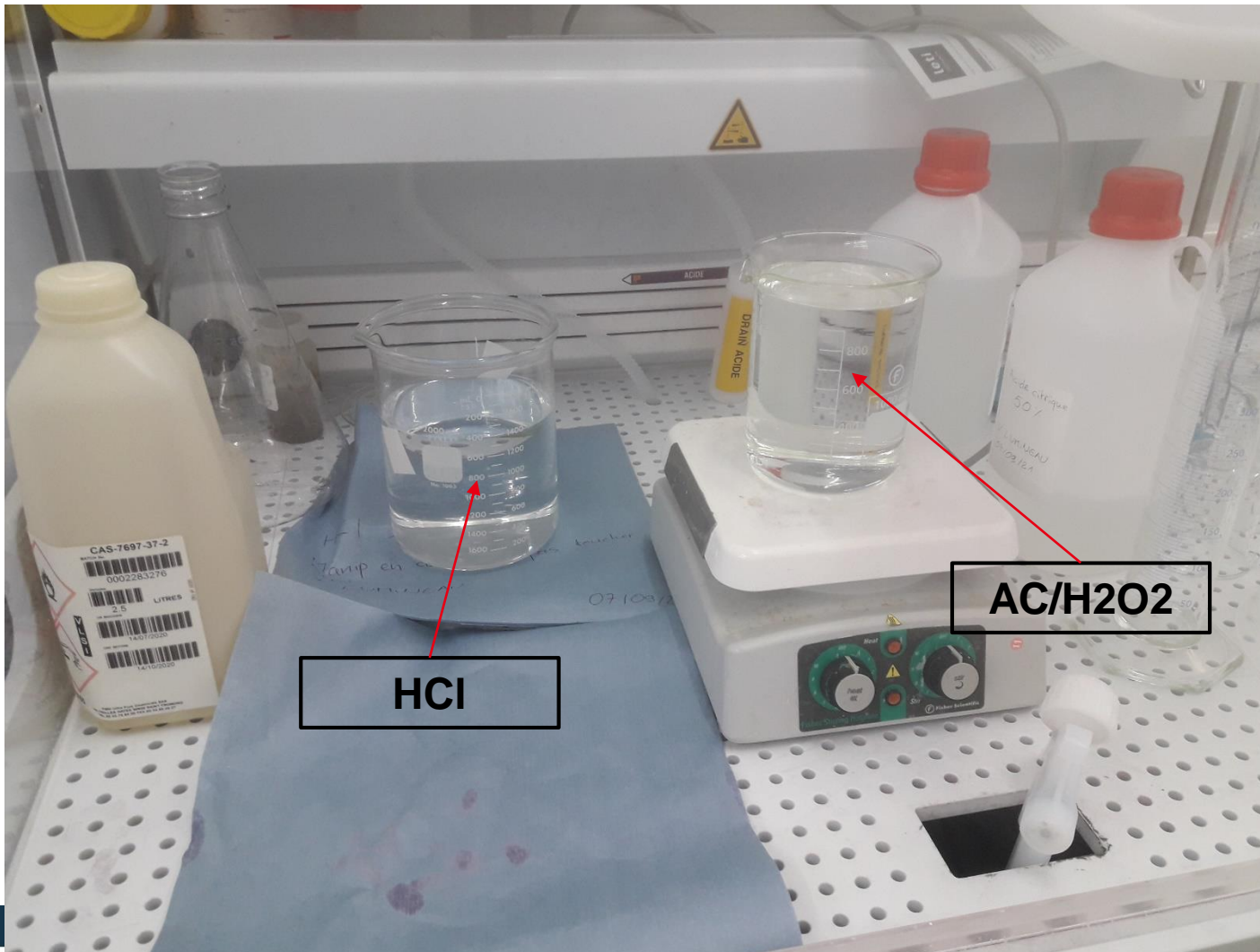
P04



P05



GRAVURE CHIMIQUE



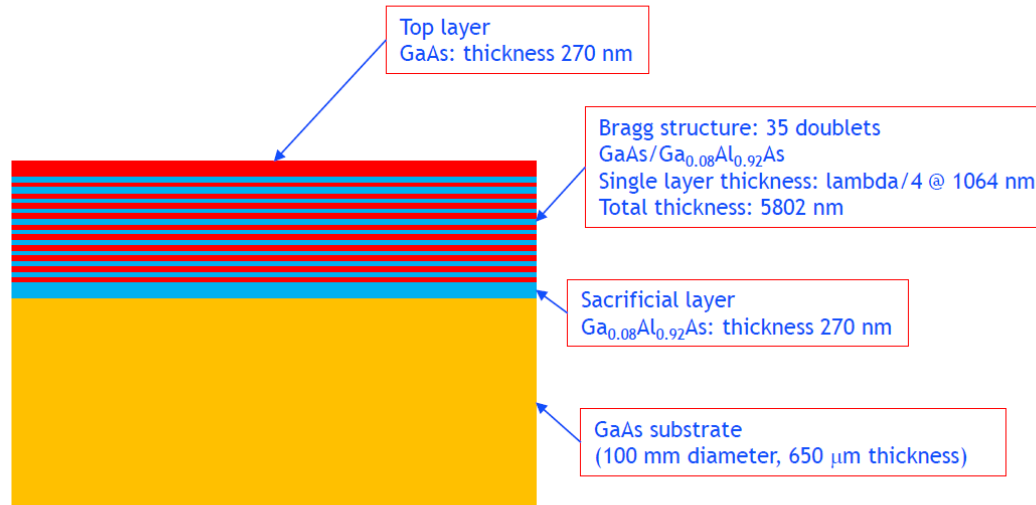
HCl

AC/H₂O₂

GRINDING



Multicouche GaAs/GaAlAs

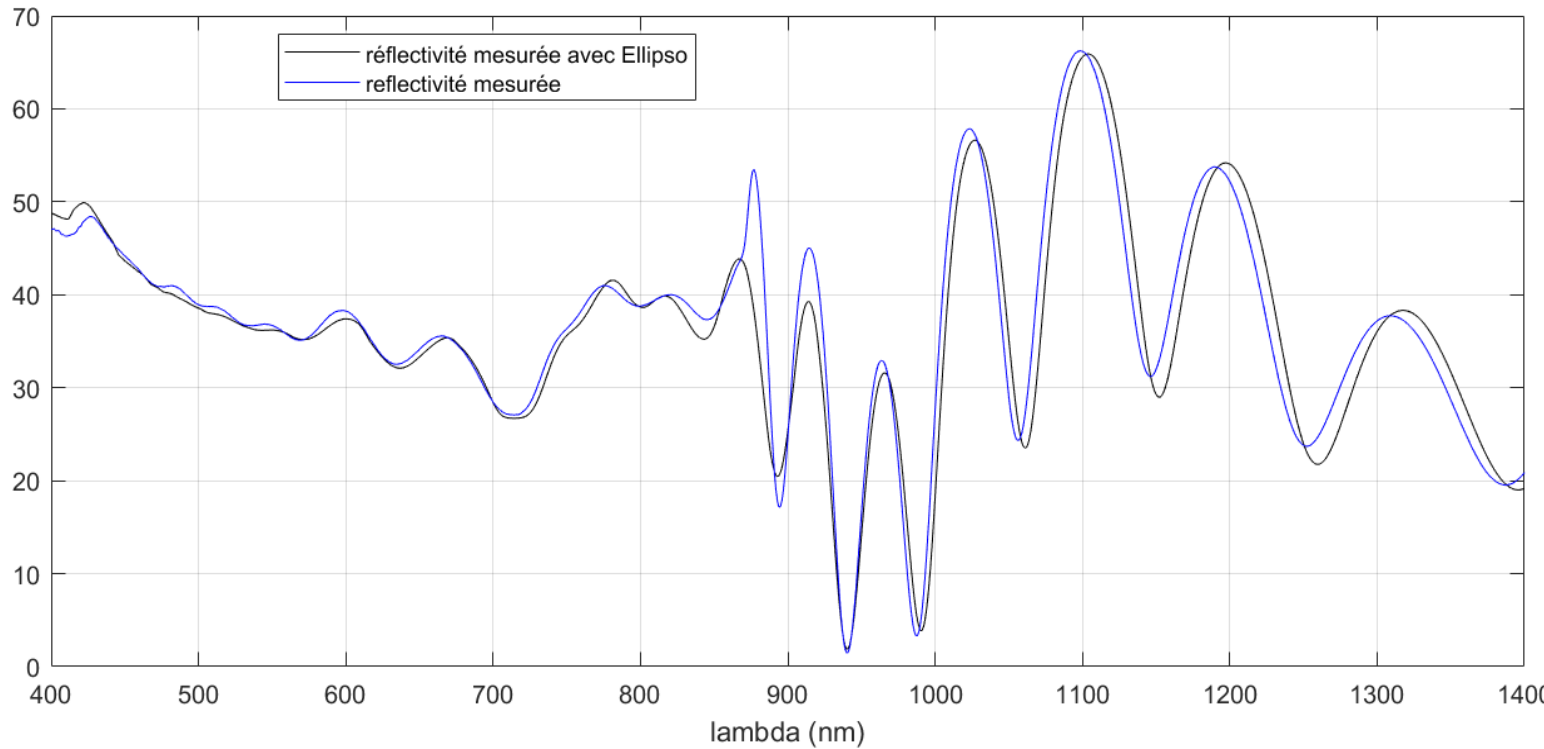


COMPARAISON SPECTRE DE REFLEXION AVEC ÉPAISSEURS MESURÉS AU CEA ET AU LMA

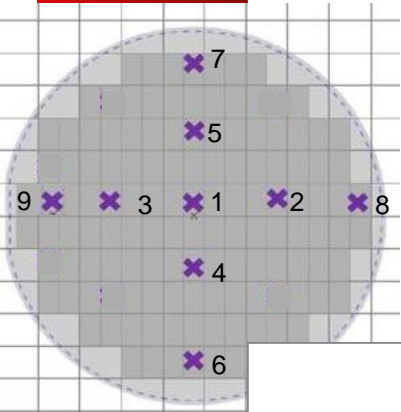
Mesures LMA



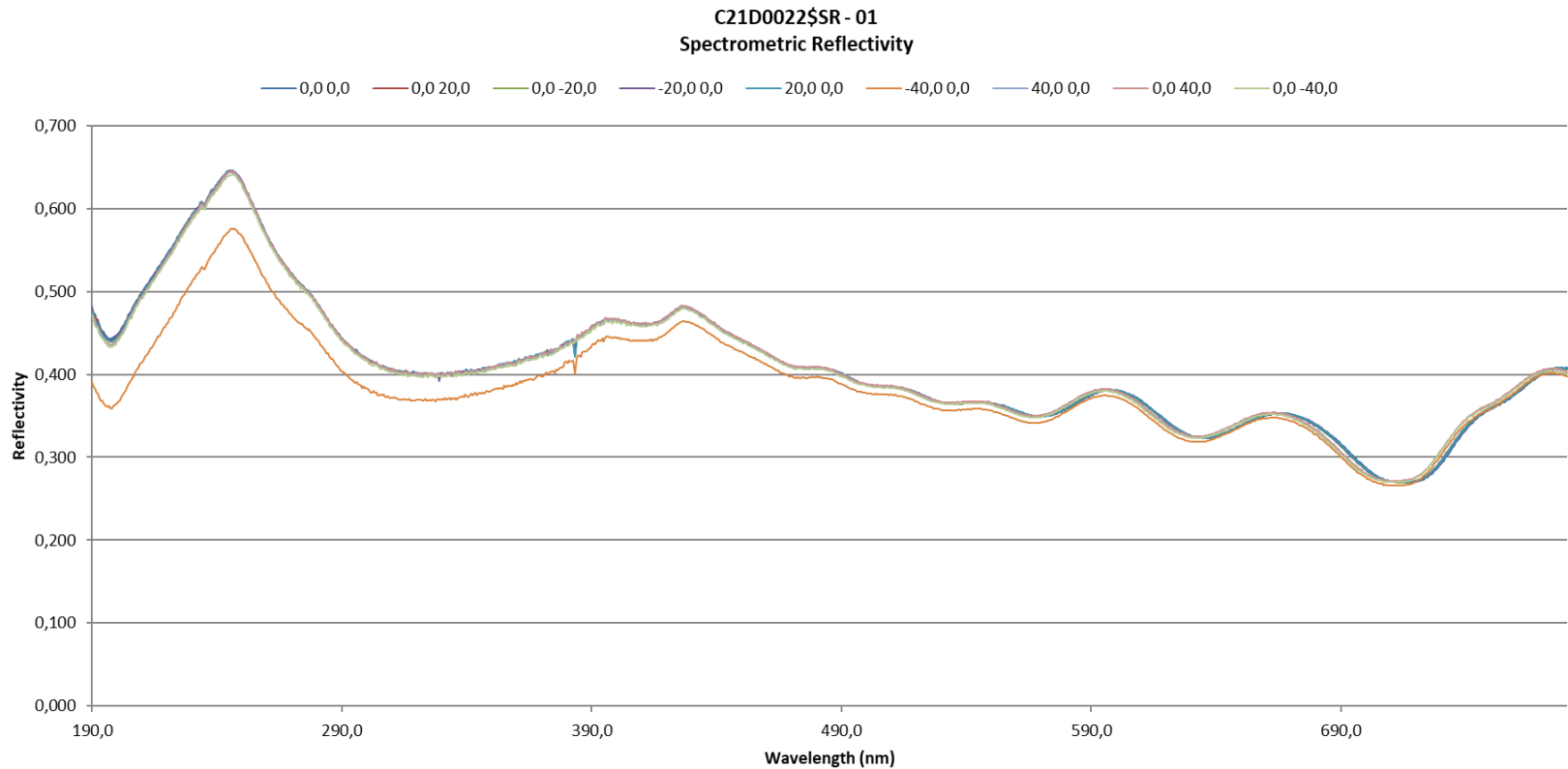
Simu avec
mesures CEA



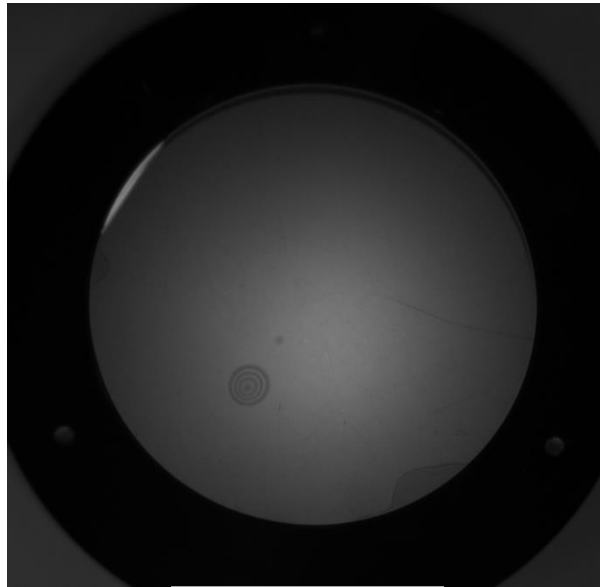
CARTO REFLEXION SUR NANOMETRICS



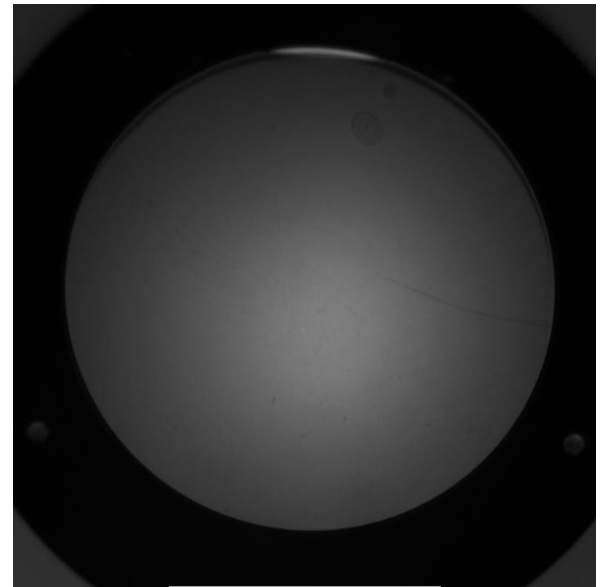
- Différence sur l'amplitude pour $X=-40\text{mm}$ et $Y=0\text{mm}$
Différence majoritairement dans l'UV => variation sur l'oxydation/état de surface



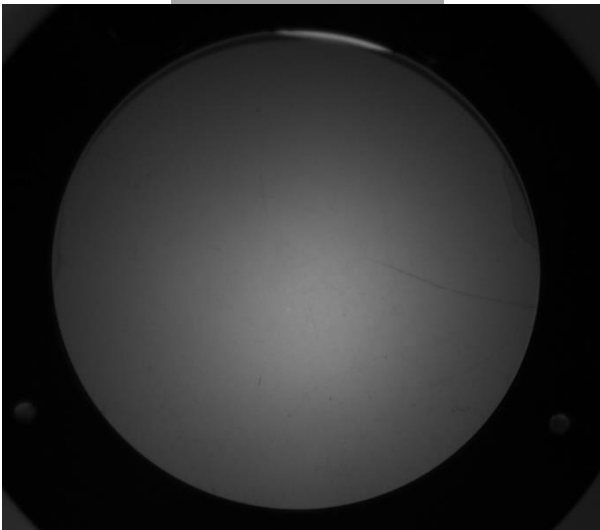
DIRECT BONDING GAAS/SIO2



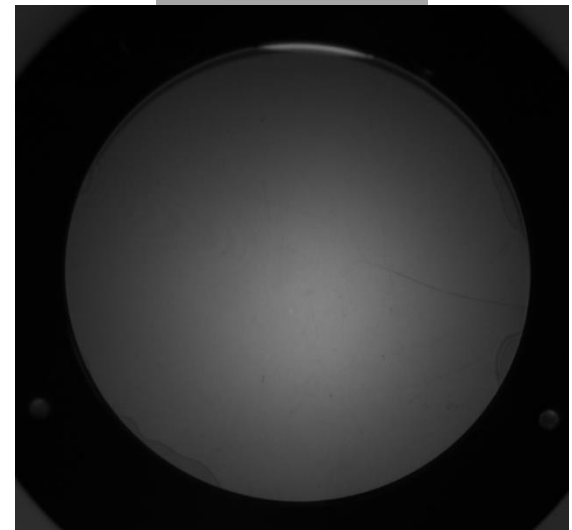
P01/P11



P02/P12



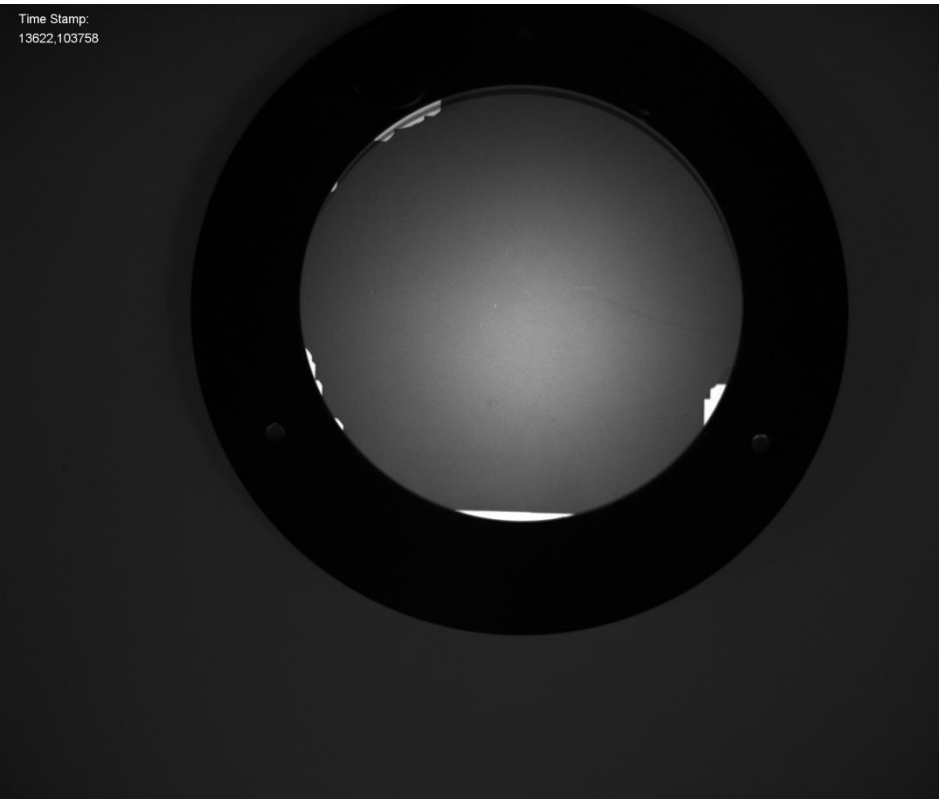
P03/P13



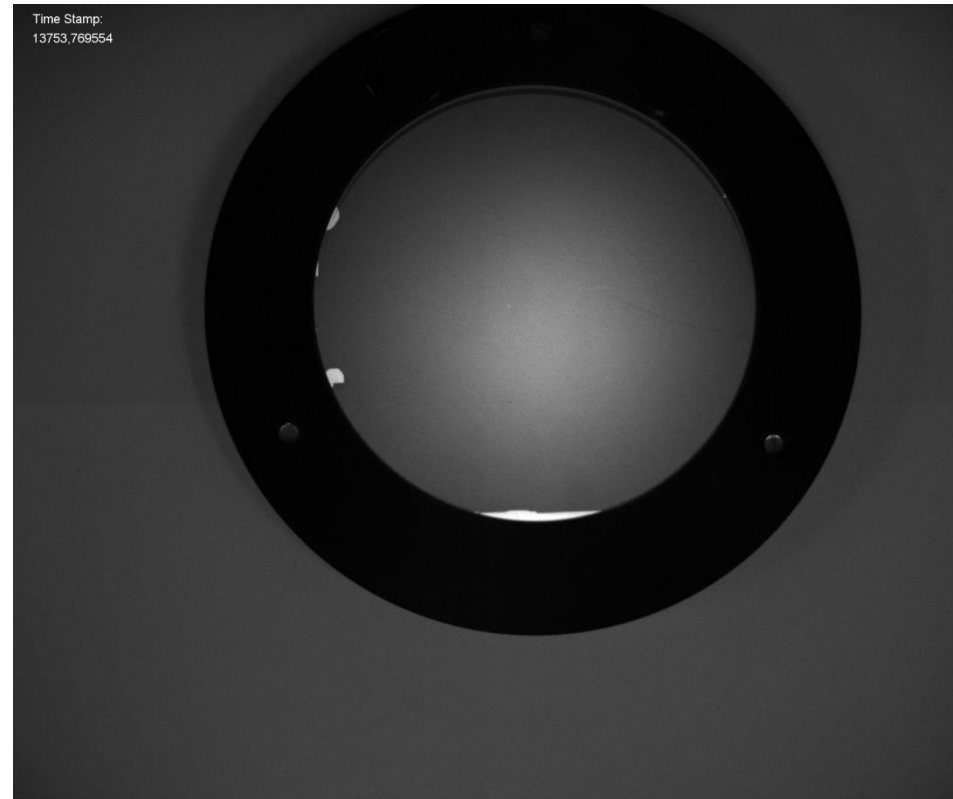
P04/P14

PHOTOS IR P12 AND P14

- P12 : grinding @100 μ m + etching @20 μ m+ recuit à 100°C sous N2
- P14 : grinding @100 μ m + annealing 100°C sous N2



P12



P14

WAFERS AFTER DELAMINATION (C20D0054)

