

# The production of light by matter - from light bulbs to monolithic blue LEDs

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# Context

## The Nobel Prize in Physics 2014



Hisamu Akasaki



Hiroshi Amano



Shuji Nakamura

“ « for the invention of efficient blue light-emitting diodes which has enabled bright and energy-saving white light sources».

# Outline

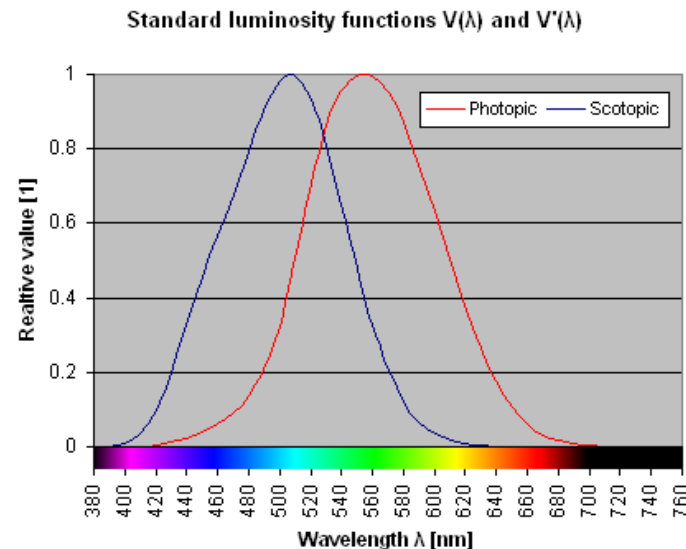
- Short review of white light sources
- Breaking down the light emitting diode
  - *Electronic states in bulk semiconductors*
  - *Light-matter interaction*
  - *PN-junction and electroluminescence*
  - *Characterizing and improving light emitters*
- Gallium nitride and the blue emitting diode
  - *Nitride growth*
  - *Control of doping*
  - *The blue diode*
  - *Nanostructures and perspective*

# Comparing white light emitters

- **Luminous flux:** measure of the perceived power of light.

$$\text{Luminous flux (Lm)} = \text{radiant flux (W)} \times \text{luminosity function}$$

- **Luminosity function:** spectral sensitivity of human visual perception of brightness.
- In **daylight**, the **maximum sensitivity** is at **555 nm** (in the green region). This type of vision is called (*photopic vision*).
- In **low light** conditions, **maximum sensitivity** is at **507 nm** (in the blue-green region) and red light is almost invisible (*scotopic vision*).



# Comparing white light emitters

- **Luminous flux:** measure of the perceived power of light.

Luminous flux (Lm) = radiant flux (W) × luminosity function

- **Luminous efficacy:** ratio of flux to power consumption.

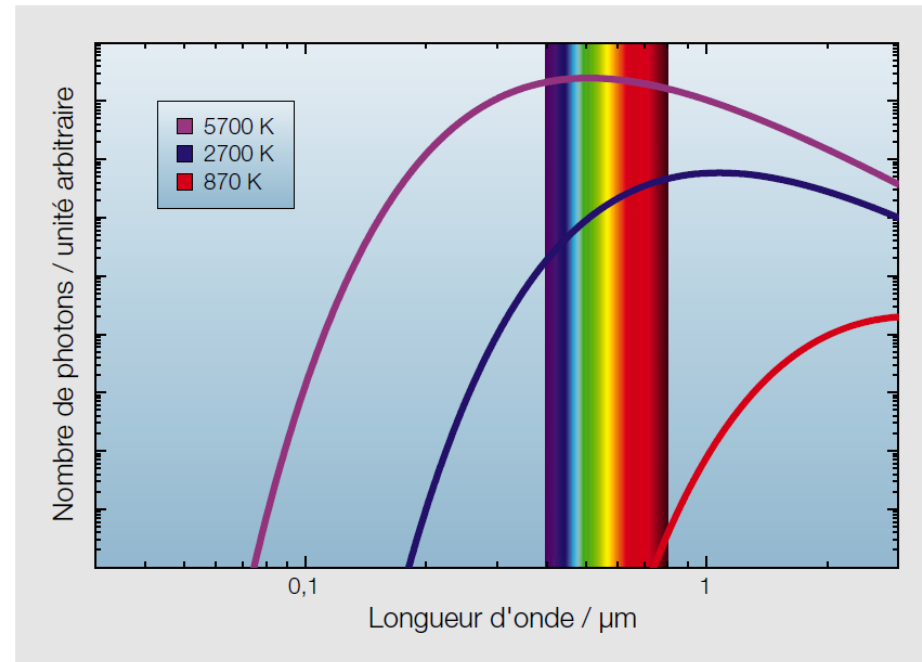
$$\text{Efficacy} = \frac{\text{Flux(Lm)}}{\text{Power (W)}}$$

- **Color rendering index (CRI) :** characterize the quality of the white light produced by a source. A value of 100 corresponds to the natural day light.

# Incandescence light bulbs



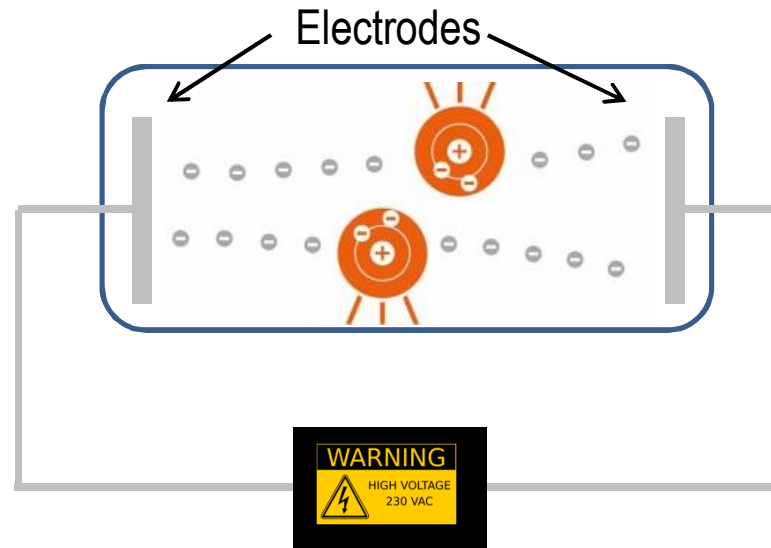
- Invented by **Joseph W. Swan** in **1879**, improved by **Thomas Edison**.
- **Principle:**
  - A **tungsten filament** is heated to a high temperature by an electric current passing through it until it glows.
  - The emitted light displays a “black body” spectrum: its shape is given by the **Planck’s law** and depends only on temperature.



**Figure 1** Spectre d’émission du corps noir pour différentes températures : soleil (5700 K), lampe à incandescence (2700 K) et 600 °C (870 K), ce qui correspond au début de l’émission dans le visible.

Efficacy (lm/W)	15
CRI	100
Lifetime(h)	1000

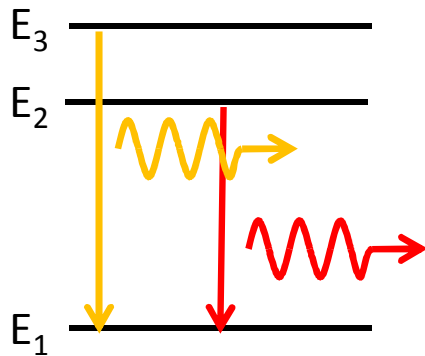
# Gas-discharge lamps



## ■ Principle:

- A gas of atoms in a sealed tube is excited by an electric discharge.
- In the plasma, mobile electrons collide with atoms, transfer energy to the atoms and elevate them to an excited state.
- Atoms relax towards their ground state by emitting light.

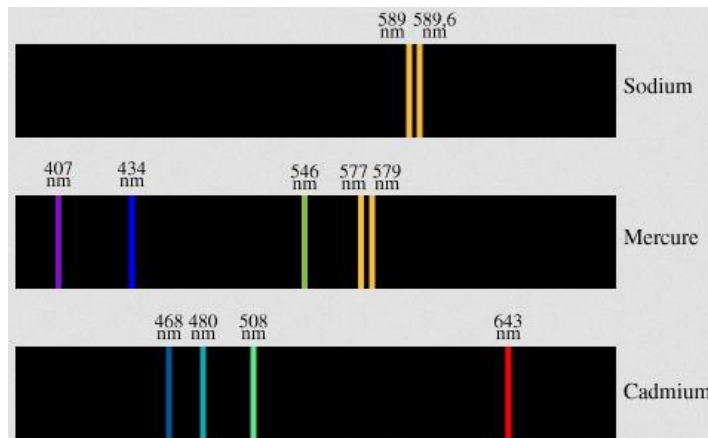
# Gas-discharge lamps



## ■ Principle:

- Electronic levels in atoms are discrete.
- The emission spectrum displays series of line the wavelengths of which satisfy the equation:

$$E_i - E_f = \frac{hc}{\lambda}$$



⇒ *The emission spectrum is characteristic of the element.*



# Gas-discharge lamps

- **Example:** High pressure sodium lamp, motorway lighting.



Efficacy (lm/W)	90-140
CRI	25
Lifetime(h)	25000

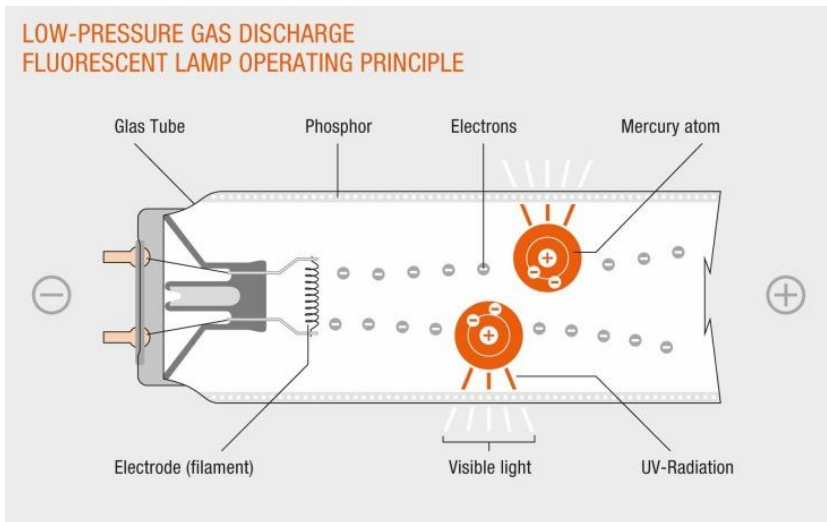
[www.marjoy.com](http://www.marjoy.com)

# Compact fluorescent tubes



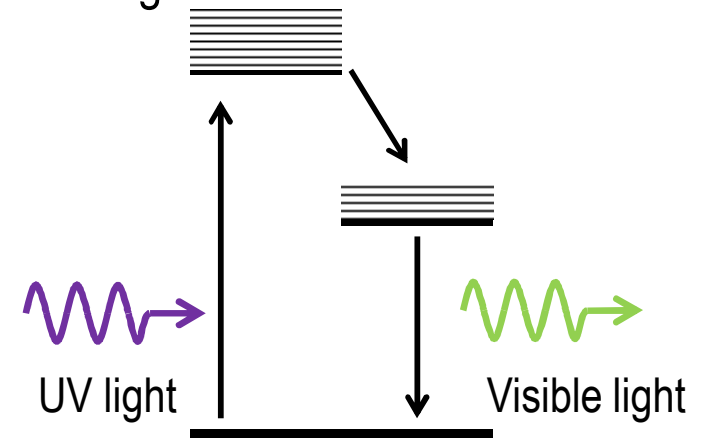
## Principle:

A gas discharge **mercury lamp** radiates **ultraviolet light**. The emitted ultraviolet light is **converted into visible light** as it strikes the phosphorescent coating on the bulb.



<http://www.osram.com>

Optical pumping of phosphorescent coating

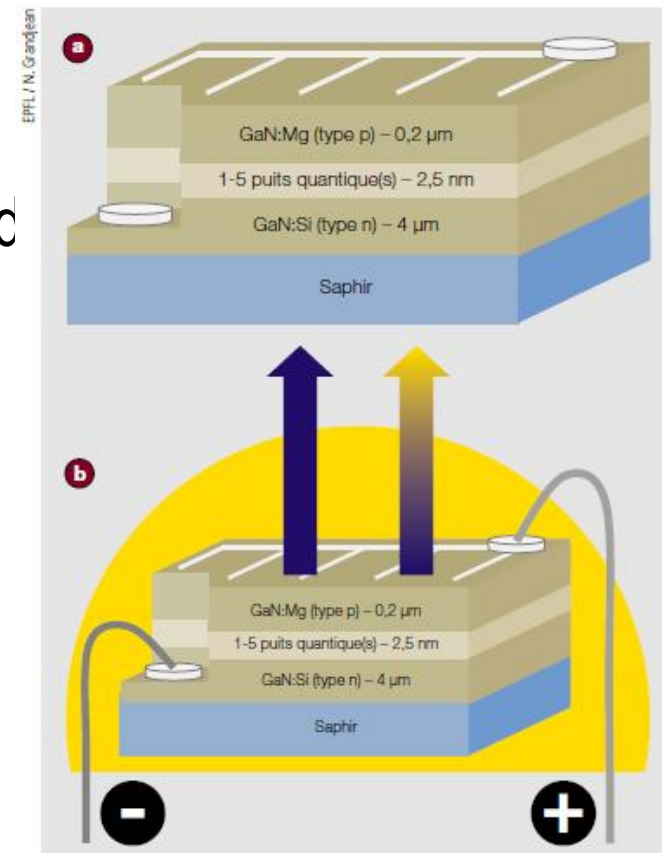


Efficacy (Lm/W)	100 (tubes), 40-60 (bulbs)
CRI	Up to 85
Lifetime (h)	~10000

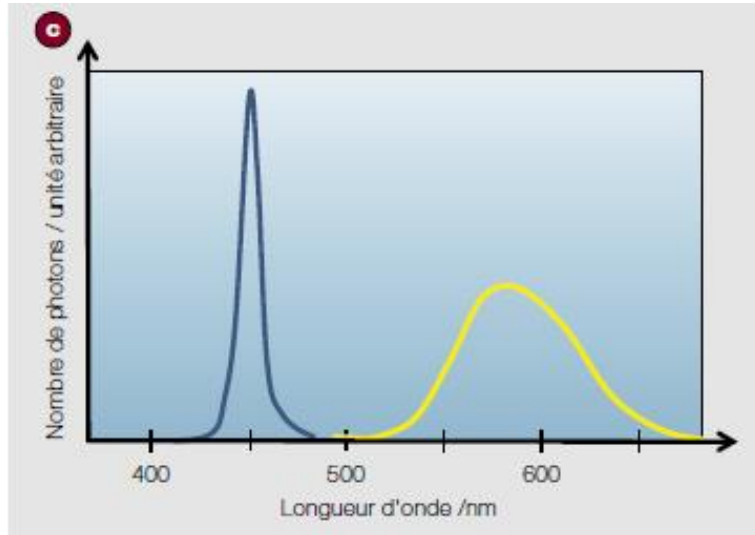
# Electroluminescent diodes

## ■ Principle:

- Radiative recombination of negative (electrons) and positive charges (« holes ») that are electrically injected in a **semiconductor diode**.
- Build around a blue emitting diode.



# White LEDs



Efficacy (Lm/W)	100
IRC	80-90
Lifetime (h)	> 15000

- The blue LED has a **double function**:
  - It supplies the **blue part** of the spectrum at 450 nm.
  - It **Optical pumps** a yellow **phosphorescent compound**.
- White light=blue emission from a LED (450-460 nm) + yellow emission from a phosphor.

# LED history

- **1907:** H.J. Round (Marconi Electronics) discovers **electroluminescence** of **SiC**.
- **1961:** **first infrared LED** made of **GaAs** (Bob Biard and Gary Pittman, Texas instruments).
- **1962:** using **GaAsP** alloy enable the of the **first visible LED** (710 nm)(Nick Holonyak Jr. and Sam Bevacqua, General electric).
- **1960-70:** yellow and green LEDs.
- **1972:** **GaN** green emitting LED prototype (J.Pankove,RCA ).
- **1989:** **GaN** UV emitting LED prototype (380 nm) (H. Akasaki & H. Amano).
- **1994:** **InGaN** based blue diode(450 nm) S.Nakamura, Nishia.
  - ⇒ *Different materials emit light in distinct ranges of wavelength.*

# Semiconductor compounds

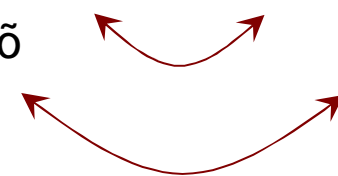
$\geq$	<u>1</u> I A	<u>2</u> II A	<u>3</u> III B	<u>4</u> IV B	<u>5</u> V B	<u>6</u> VI B	<u>7</u> VII B	<u>8</u> VIII B	<u>9</u> VIII B	<u>10</u> VIII B	<u>11</u> I B	<u>12</u> II B	<u>13</u> III A	<u>14</u> IV A	<u>15</u> V A	<u>16</u> VI A	<u>17</u> VII A	<u>18</u> VIII A						
$\geq$	1 H																	2 He						
2	3 Li	4 Be																	5 B	6 C	7 N	8 O	9 F	10 Ne
3	11 Na	12 Mg																	13 Al	14 Si	15 P	16 S	17 Cl	18 Ar
4	19 K	20 Ca	21 Sc	22 Ti	23 V	24 Cr	25 Mn	26 Fe	27 Co	28 Ni	29 Cu	30 Zn	31 Ga	32 Ge	33 As	34 Se	35 Br	36 Kr						
5	37 Rb	38 Sr	39 Y	40 Zr	41 Nb	42 Mo	43 Tc	44 Ru	45 Rh	46 Pd	47 Ag	48 Cd	49 In	50 Sn	51 Sb	52 Te	53 I	54 Xe						
6	55 Cs	56 Ba	*	72 Hf	73 Ta	74 W	75 Re	76 Os	77 Ir	78 Pt	79 Au	80 Hg	81 Tl	82 Pb	83 Bi	84 Po	85 At	86 Rn						
7	87 Fr	88 Ra	**	104 Rf	105 Db	106 Sg	107 Bh	108 Hs	109 Mt	110 Ds	111 Rg	112 Uub	113 Uut	114 Uuq	115 Uup	116 Uuh	117 Uus	118 Uuo						



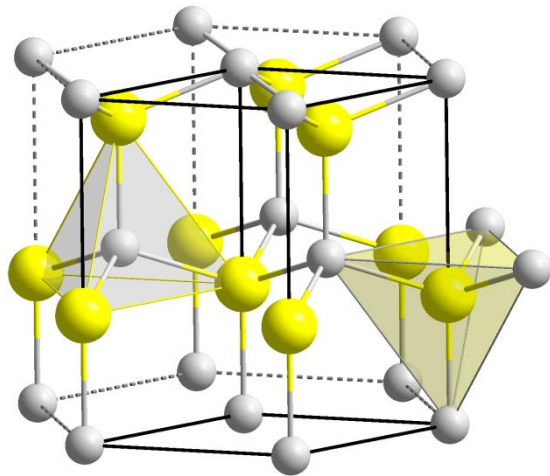
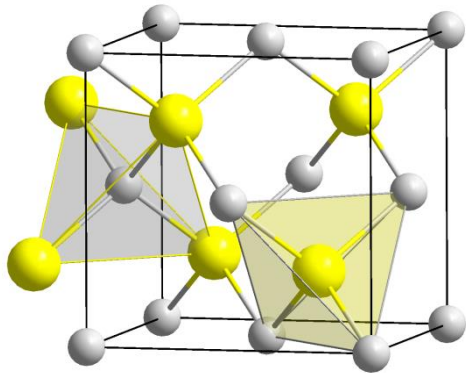
* <b>Lanthanides</b>	57 La	58 Ce	59 Pr	60 Nd	61 Pm	62 Sm	63 Eu	64 Gd	65 Tb	66 Dy	67 Ho	68 Er	69 Tm	70 Yb	71 Lu
** <b>Actinides</b>	89 Ac	90 Th	91 Pa	92 U	93 Np	94 Pu	95 Am	96 Cm	97 Bk	98 Cf	99 Es	100 Fm	101 Md	102 No	103 Lr

III-V compounds: GaN, GaAs, InP

II-VI compounds: ZnS, CdTe



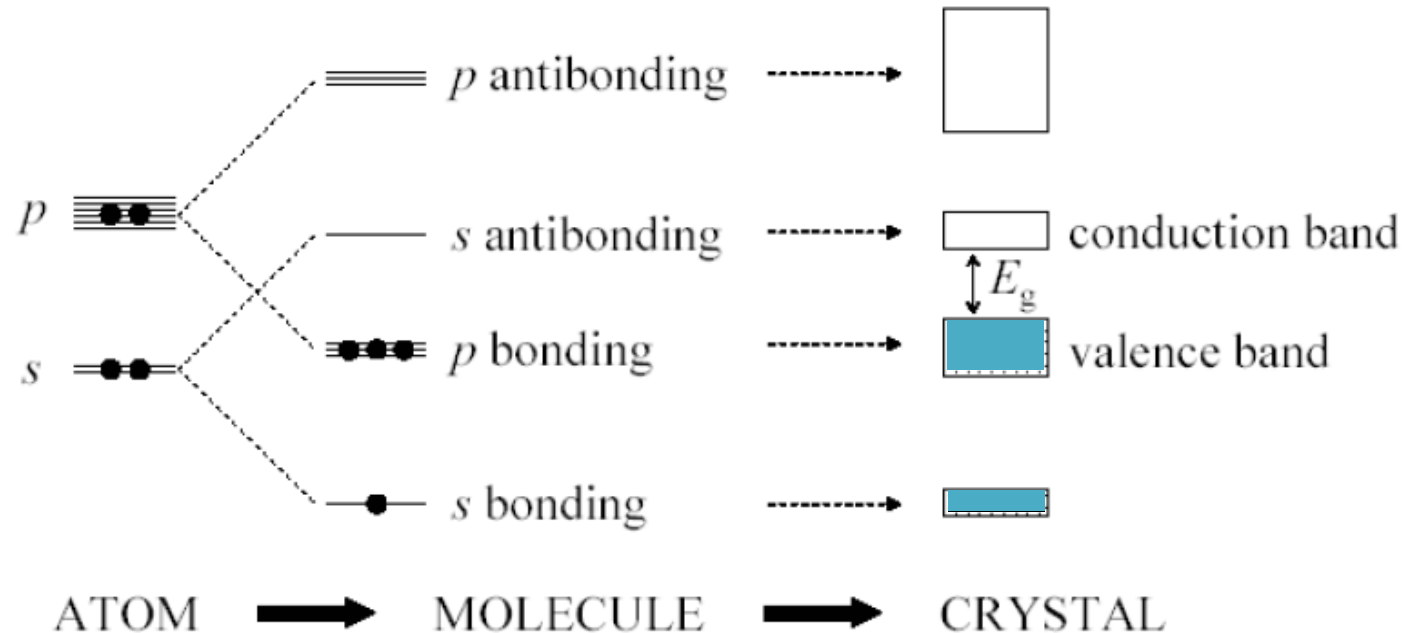
# Structure cristalline



- zinc-blende (cubic) and wurtzite (hexagonal) structures are the most common.
- The arrangement of atoms is **periodic** in real space.
- Crystal electrons evolve in the **periodic Coulomb potential** created by atoms..

Semiconductor	Structure	Lattice parameter
Si	Diamond	$a = 5,430 \text{ \AA}$
GaAs	Zinc-blende	$a = 5,653 \text{ \AA}$
GaN	Wurtzite	$a = 3,189 \text{ \AA}$ $c = 5,185 \text{ \AA}$

# From atomic states to energy bands



- The electronic states of the periodic lattice are built by **coupling the valence states** of the isolated atoms.
- The coupling **lifts the degeneracy** between the atomic states and opens **bandgaps** in the band structure.

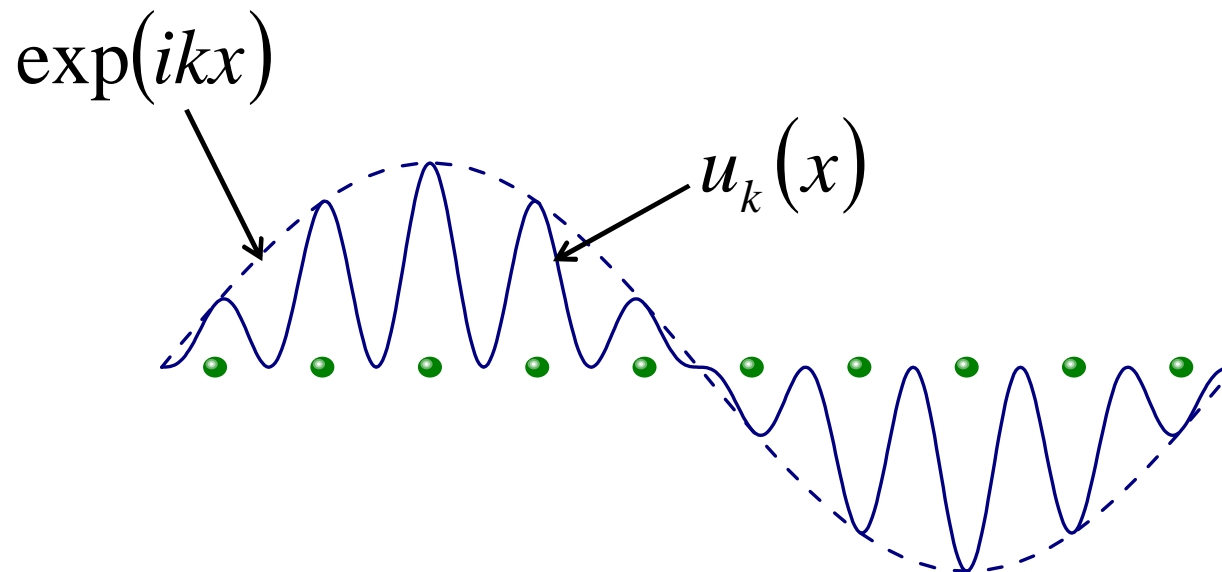
⇒ *The bandgap energy is characteristic of a semiconductor.*



# Electronic states

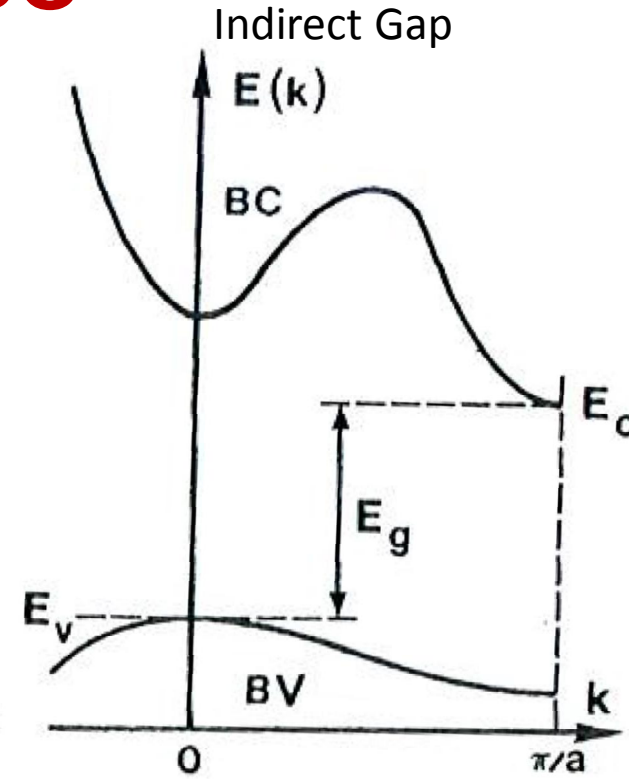
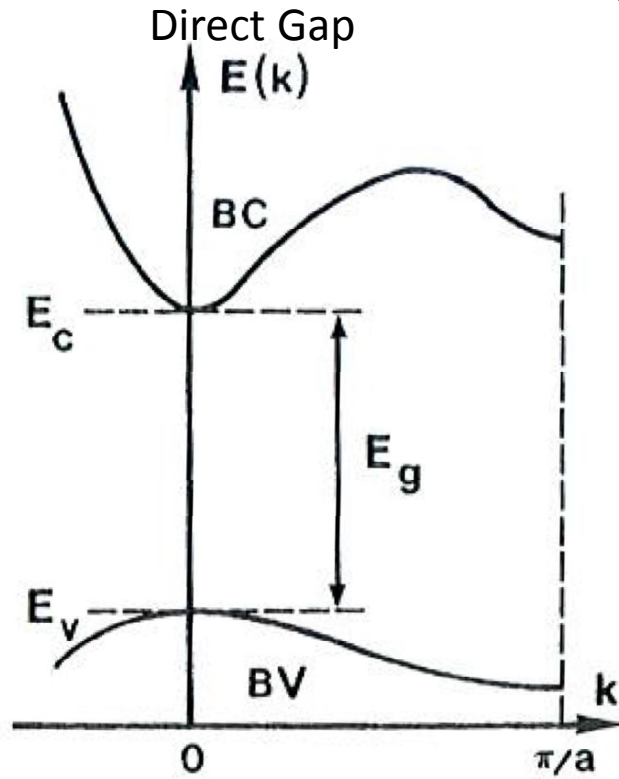
- Cristal: periodic arrays of atoms, translational **symmetry**.
- **Electrons** can **delocalize** on crystal atoms and are characterized by a **wavefunction**:

$$\psi_k(x) = \exp(ikx) \times u_n(x)$$

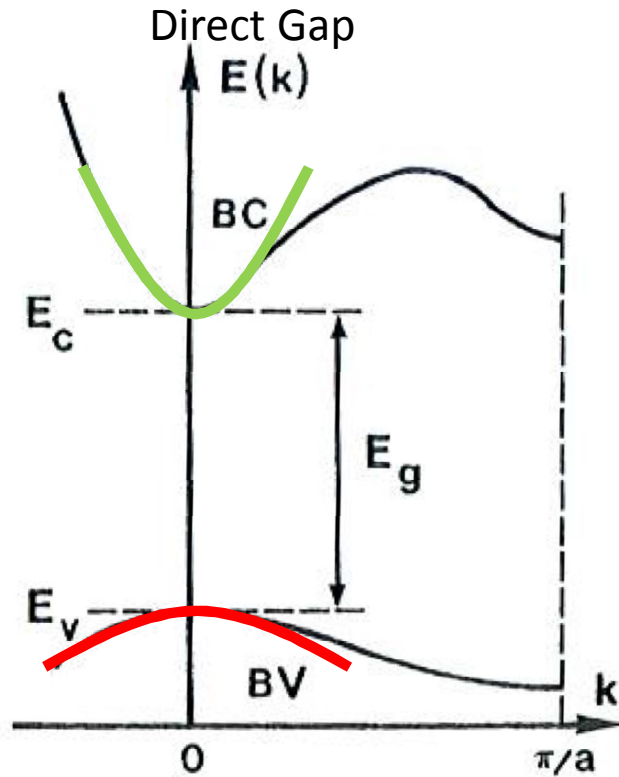


⇒ **Energy depends on quantum number  $k$**

# Band structure in reciprocal space



# Effective mass



Near the band maxima:

$$E(k) = E(k_0) + \underbrace{k \frac{\partial E}{\partial k}}_{=0} + \frac{k^2}{2} \frac{\partial^2 E}{\partial k^2} + \dots$$

By introducing the electron « effective mass »

$$m_e^* = \hbar^2 \left( \frac{\partial^2 E}{\partial k^2} \right)^{-1}$$

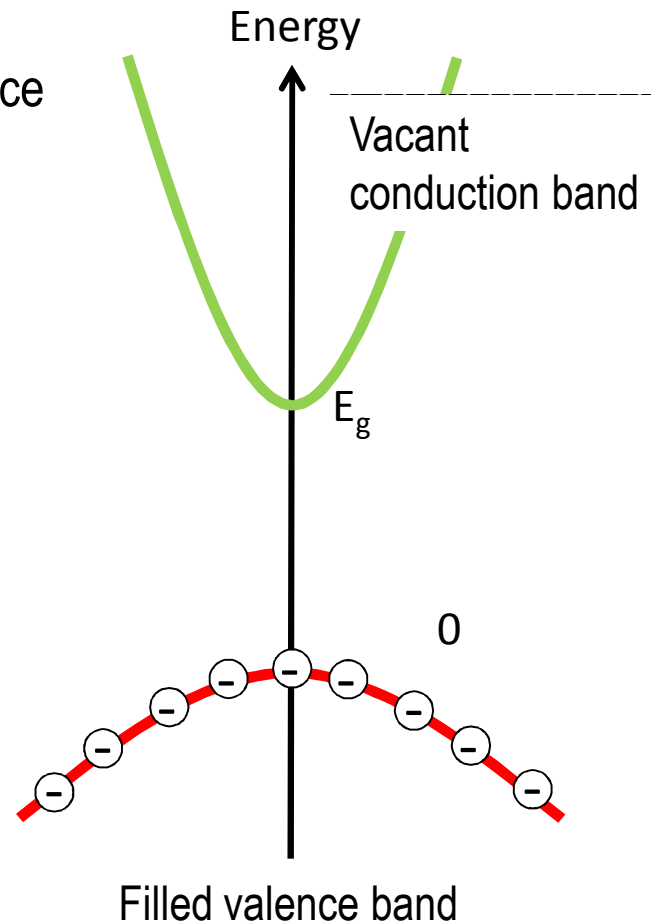
The energy dispersion is similar to the one of a free electron of mass  $m^*$ .

$$E(k) \approx E(k_0) + \frac{\hbar^2 k^2}{2m^*}$$

⇒ *Effective medium with mass  $m_e^*$ .*

# Intrinsic semiconductors

- At  $T = 0$  K, all electrons occupy valence states.
- The crystal is **insulant**.

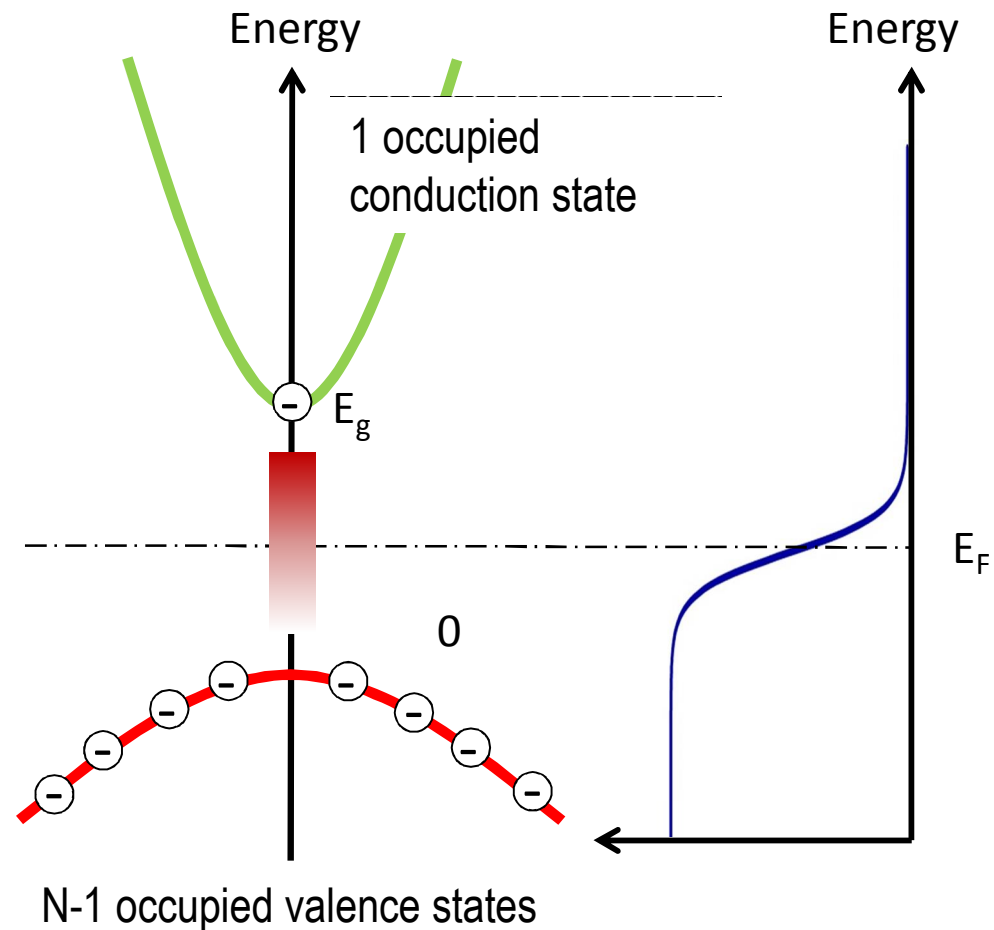


# Intrinsic semiconductors

- For  $T \neq 0$ , electrons can be **thermally excited** across the energy gap into the conduction band.
  - Conduction electrons = “free electrons”
- The probability that a state is occupied is given by the **Fermi-Dirac distribution**:

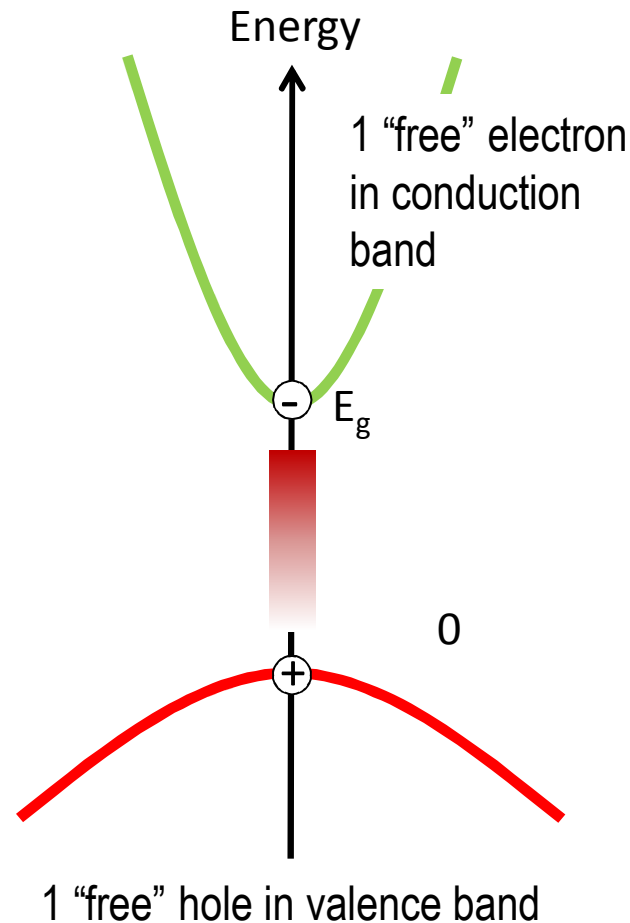
$$f(E) = \frac{1}{\exp\left(\frac{E - E_F}{k_B T}\right) + 1}$$

$$E_F \approx \frac{E_G}{2}$$

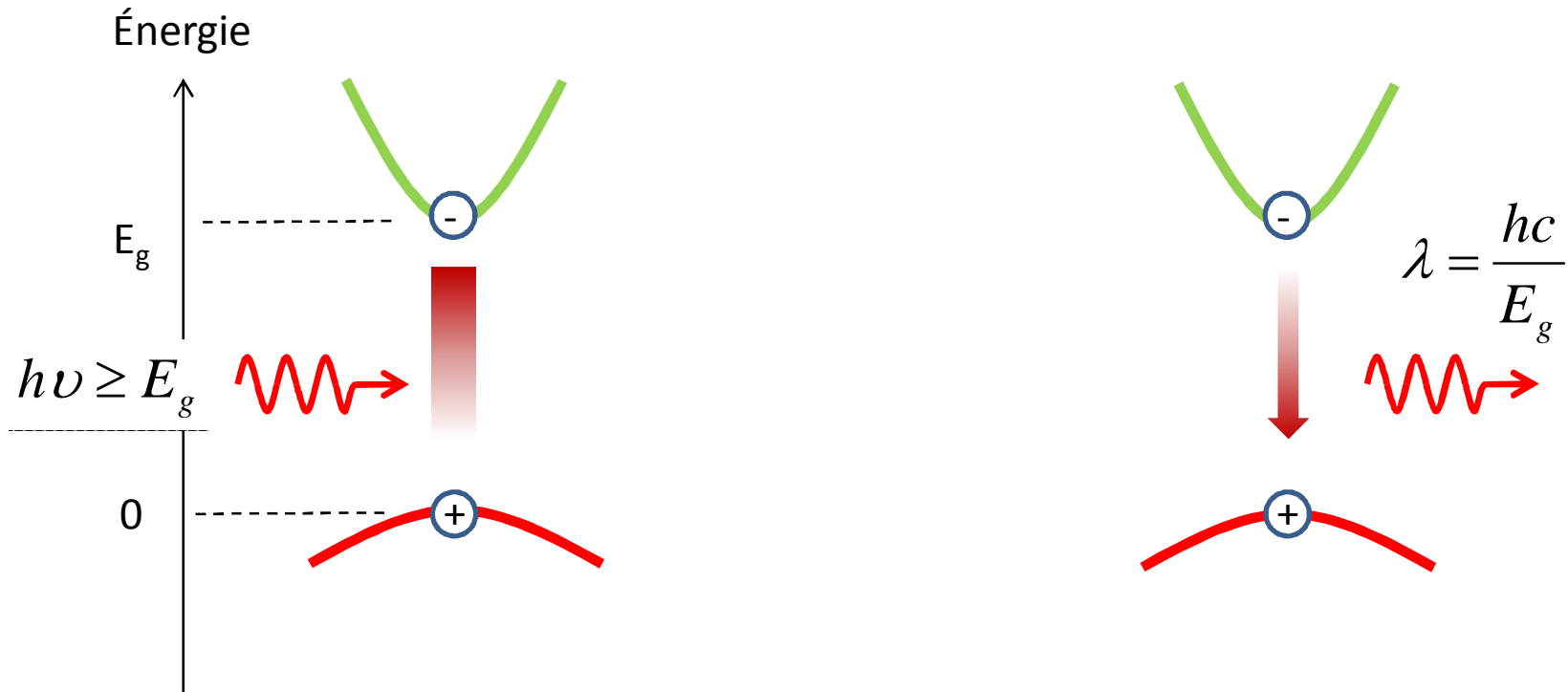


# Intrinsic semiconductors

- The excited electron leaves behind **one unoccupied state** in the valence band that is equivalent to a quasi-particle **carrying a positive charge**.
  - This lack of negative charge is called a **“hole”**.
- ⇒ *In semiconductors, electric current is mediated by electrons and holes.*



# Optical processes



- **Absorption:**

A photon which energy is larger than the bandgap  $E_g$  can be absorbed to create one electron in the CB and one hole in the VB.

- **Spontaneous emission:**

An electron at the bottom of the CB can recombine with a hole at the top of the VB to emit one photon which energy is close to the bandgap energy.

# Selection rules

Quantum mechanics  $\Rightarrow$  probability of optical transition between from an initial state  $|i\rangle$  to a final state  $|f\rangle$

*Fermi's golden rule:*

$$P = \frac{2\pi}{\hbar} |\langle f | V | i \rangle|^2 \rho(f) \delta(E_f - E_i \pm \hbar\omega)$$

Transition probability/s

Matrix element of light-matter interaction

Final density of state

Energy conservation



# Application to semiconductors

Eigen functions:

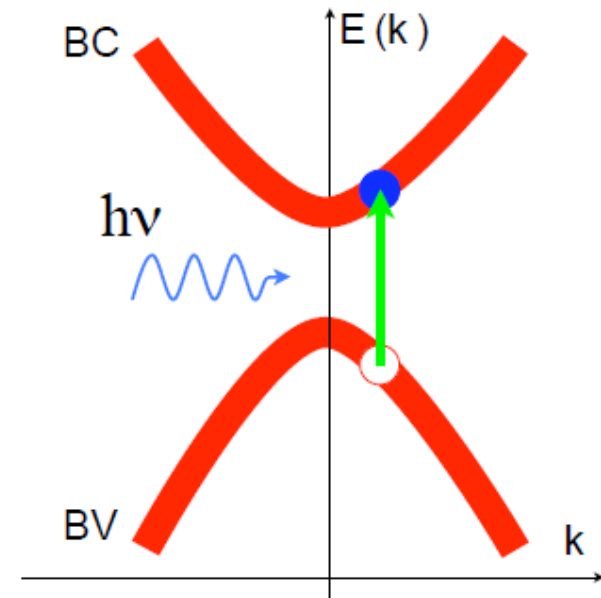
$$|\psi_{\vec{k}}\rangle = u_{\vec{k}}(\vec{r}) e^{i\vec{k}\vec{r}}$$

Periodic                      Plane wave

Optical matrix element:

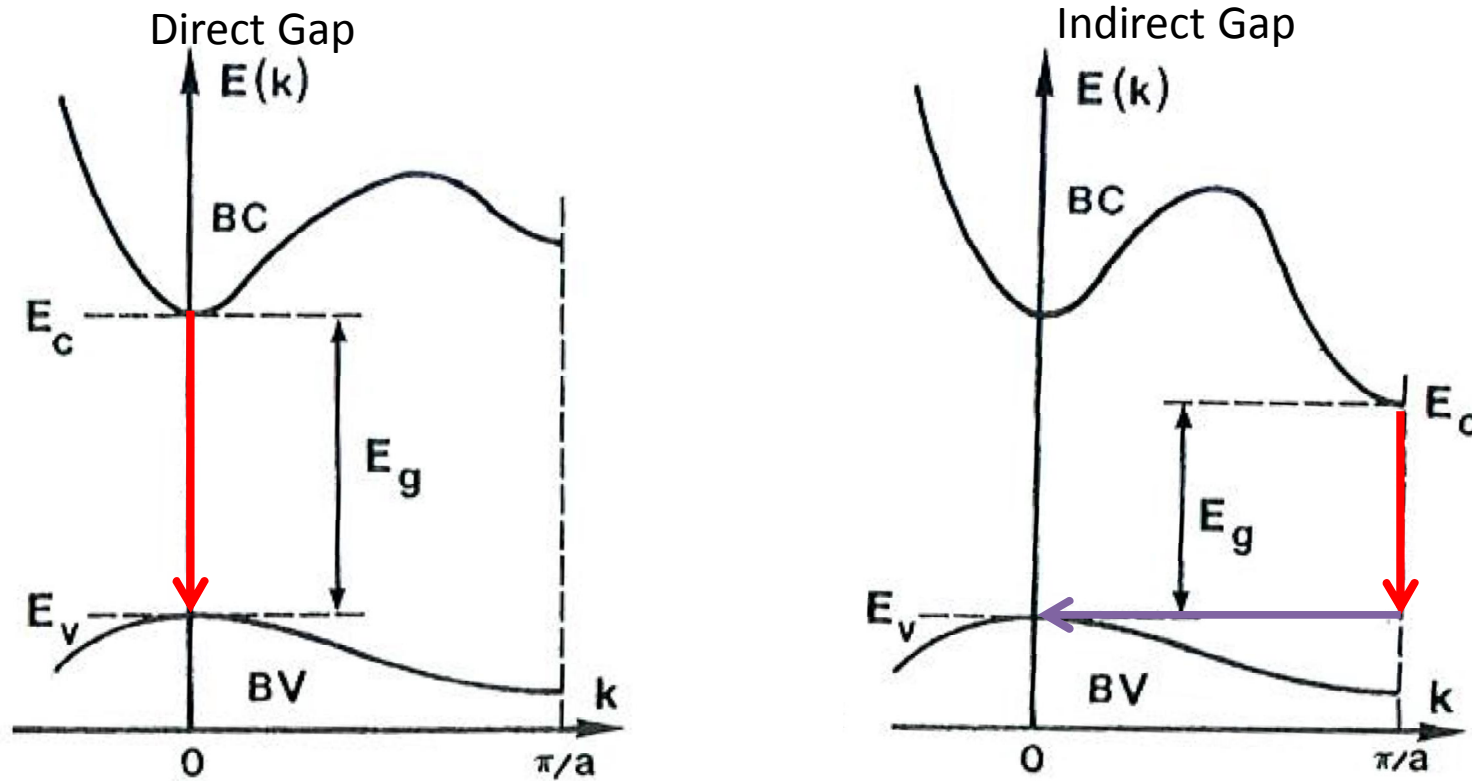
$$\langle f|V|i\rangle \propto \langle u_i|V|u_f\rangle \times \delta(\vec{k}_f - \vec{k}_i - \vec{q})$$

$\vec{q} = 2\pi/\lambda$ , light wavevector



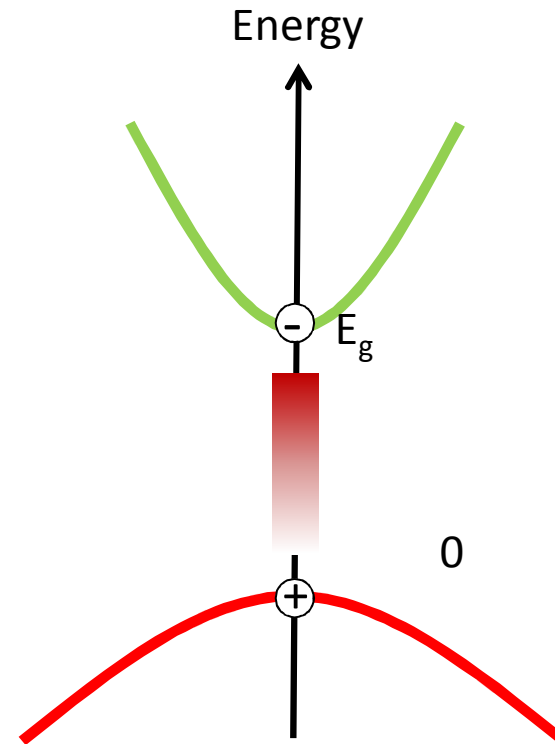
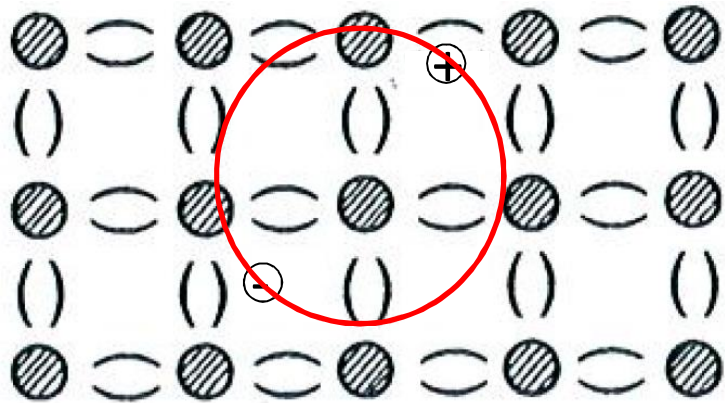
*Selections rules show vertical transitions  $\Delta k=0$*

# Direct and indirect bandgaps



⇒ *Bulk semiconductors with indirect bandgap are weak light emitters.*

# Excitons



- The electron and the hole are bound par **Coulomb interaction**.
- The exciton can freely **propagate** or **localize** on defects.  
⇒ *It is a very fine probe of matter, very sensitive to imperfections..*

# Hydrogen-like description

Exciton seen as an hydrogen atom in a effective medium (*defined by  $m_e^*$ ,  $m_h^*$  and  $\epsilon_r$* )

$$E_{ex}(n, K) = E_g - \frac{R_y^*}{n^2} + \frac{\hbar^2 K^2}{2M}$$

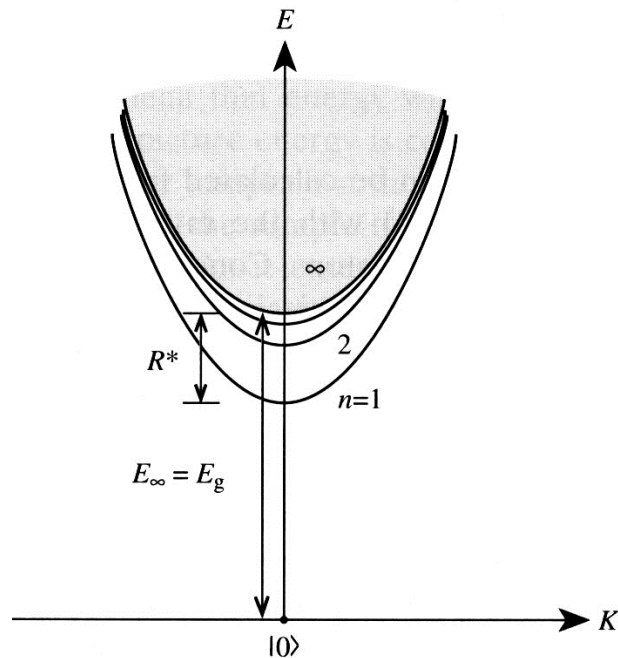
$$R_y^* = 13.6 \frac{\mu}{\epsilon_r^2 m_0} \quad \text{Exciton binding energy}$$

$$M = m_e + m_h \quad K = k_e + k_h \quad \text{Mass and wave vector of the exciton}$$

$$\frac{1}{\mu} = \frac{1}{m_e} + \frac{1}{m_h} \quad \text{Reduced mass for the relative motion}$$

$$a_B^* = a_B \frac{m_0}{\mu \epsilon_r} = 0.052 \frac{m_0}{\mu \epsilon_r} \quad \text{Bohr radius}$$

# Light-matter interaction in semiconductors

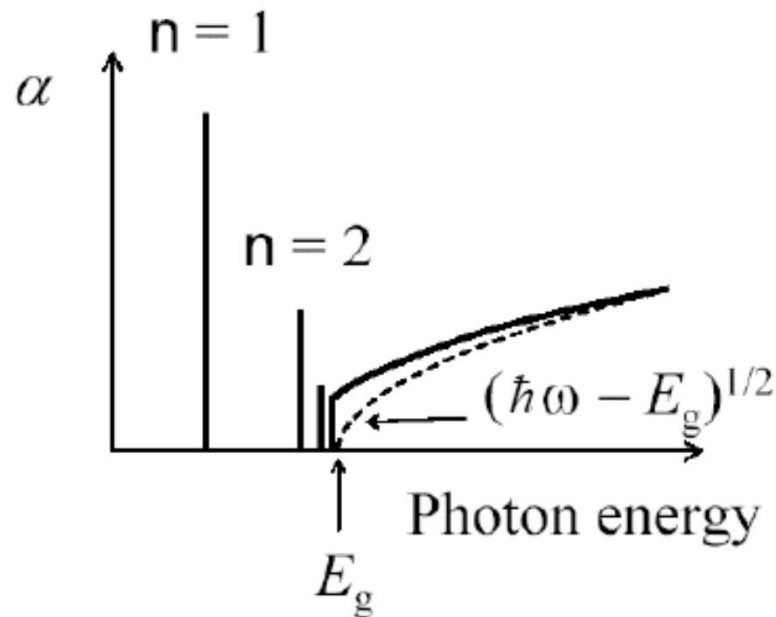


**Fig. 6.21.** The energy states of a Wannier exciton showing both its bound states  $n = 1$  to 3 and the continuum states.  $E_g$  is the bandgap and  $R^*$  the exciton binding energy

Crystal	$E_g$ (eV)	$R_x$ (meV)	$a_x$ (nm)
GaN	3.5	23	3.1
ZnSe	2.8	20	4.5
CdS	2.6	28	2.7
ZnTe	2.4	13	5.5
CdSe	1.8	15	5.4
CdTe	1.6	12	6.7
GaAs	1.5	4.2	13
InP	1.4	4.8	12
GaSb	0.8	2.0	23
InSb	0.2	(0.4)	(100)

$R_y$  increases with  $E_g$

## Light-matter interaction in semiconductors



“ Coulomb interaction between e and h:  
bound states

“ Hydrogen-like lines

$$E_x = E_g \cdot R_y/n^2$$

“ Absorption increased if compared to the results of a simple band-to-band absorption model for  $\hbar\omega > E_g$

“ Observable in good quality crystals

“ The excitonic signature disappears for  $kT > R_y$

⇒ *In good quality crystals, excitons are responsible for strong and sharp resonances in optical spectra (absorption, reflexion and photoluminescence).*

## Light-matter interaction in semiconductors

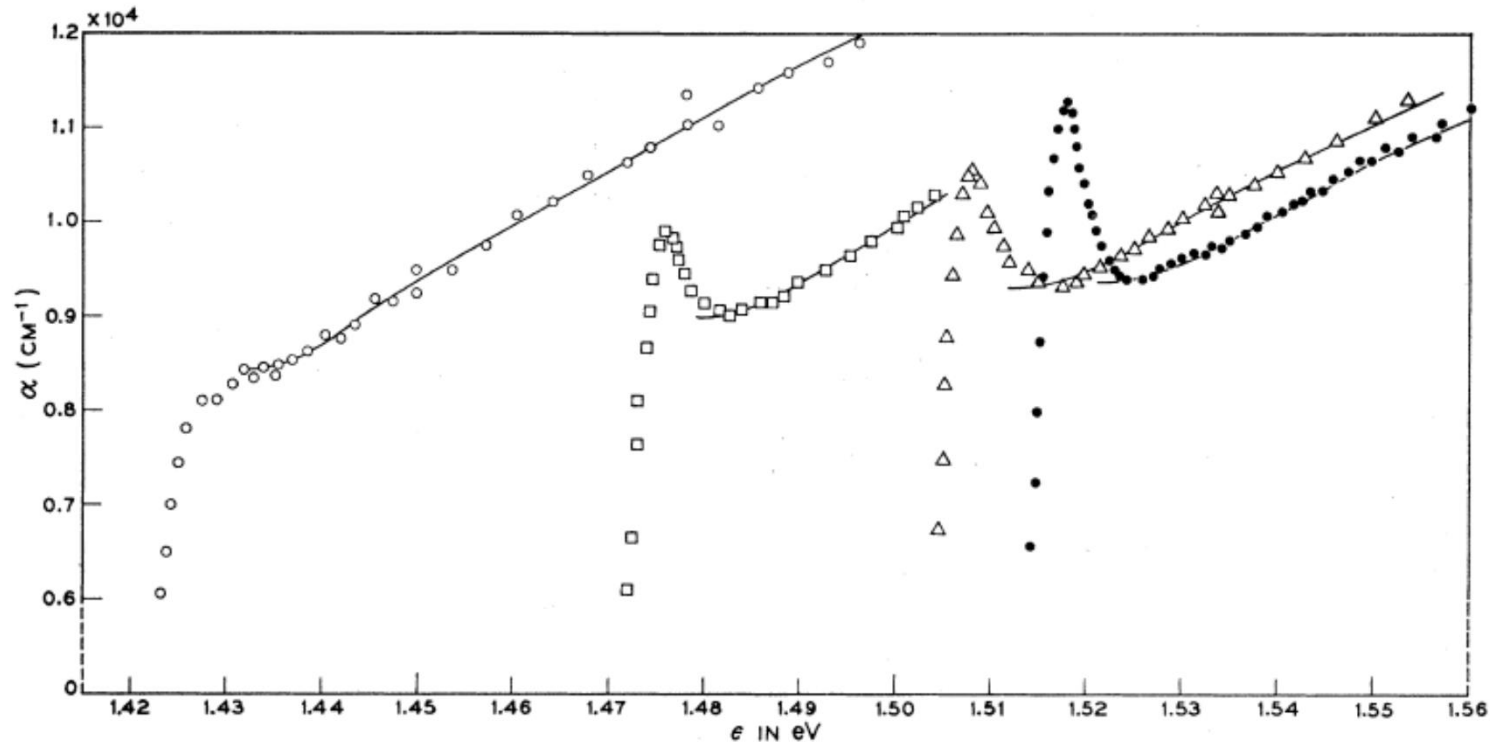
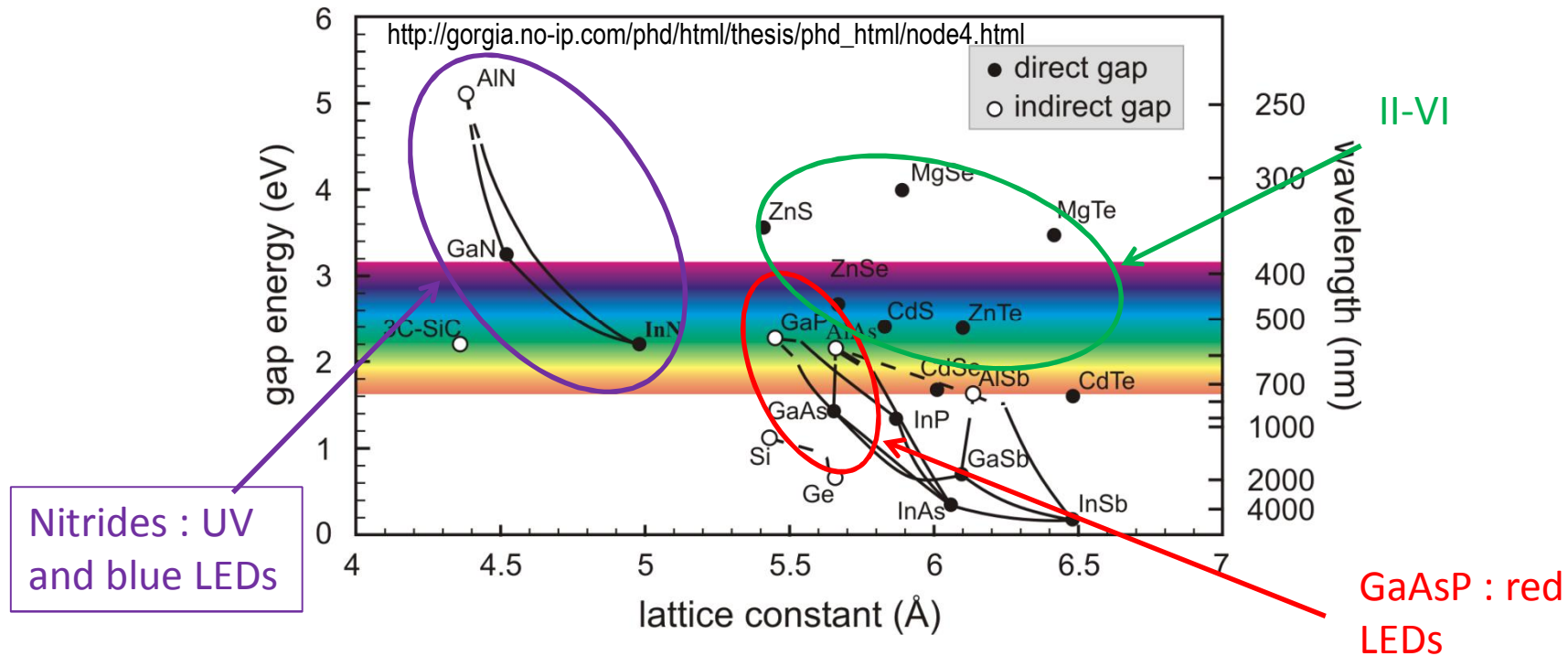


FIG 3 Exciton absorption in GaAs;  $\circ$  294°K,  $\square$  186°K,  $\Delta$  90°K,  $\bullet$  21°K.

“For a free exciton, as well as for a confined exciton, one searches for an exciton which is stable at high temperature: **high exciton binding energy or high confinement.**”

# Application

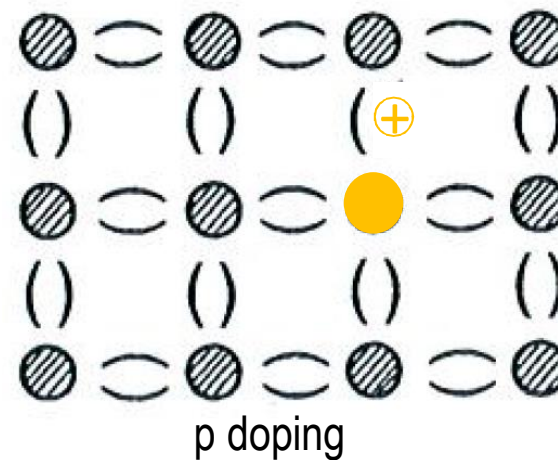
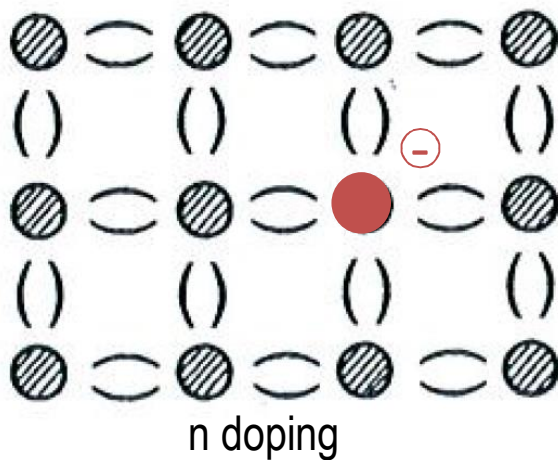
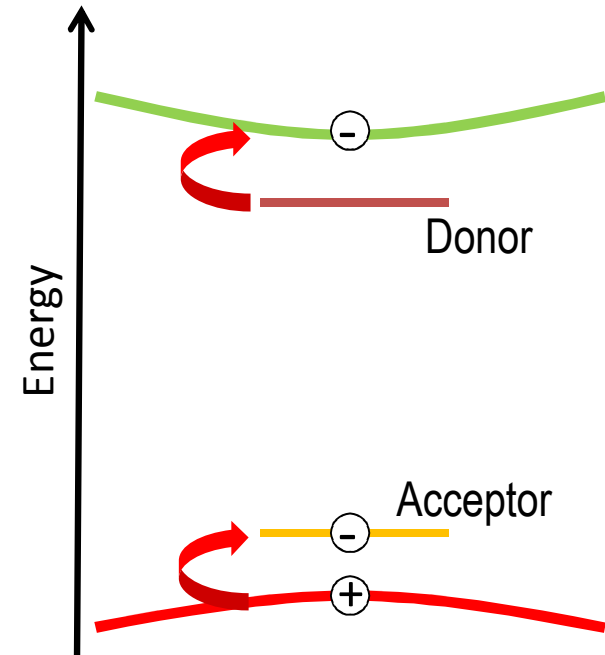
- The **emission wavelength** depends on the **bandgap energy** that is an intrinsic property of a semiconductor.
- By synthesizing different semiconductor alloys it is possible to generate **all wavelengths** from the UV to infrared.



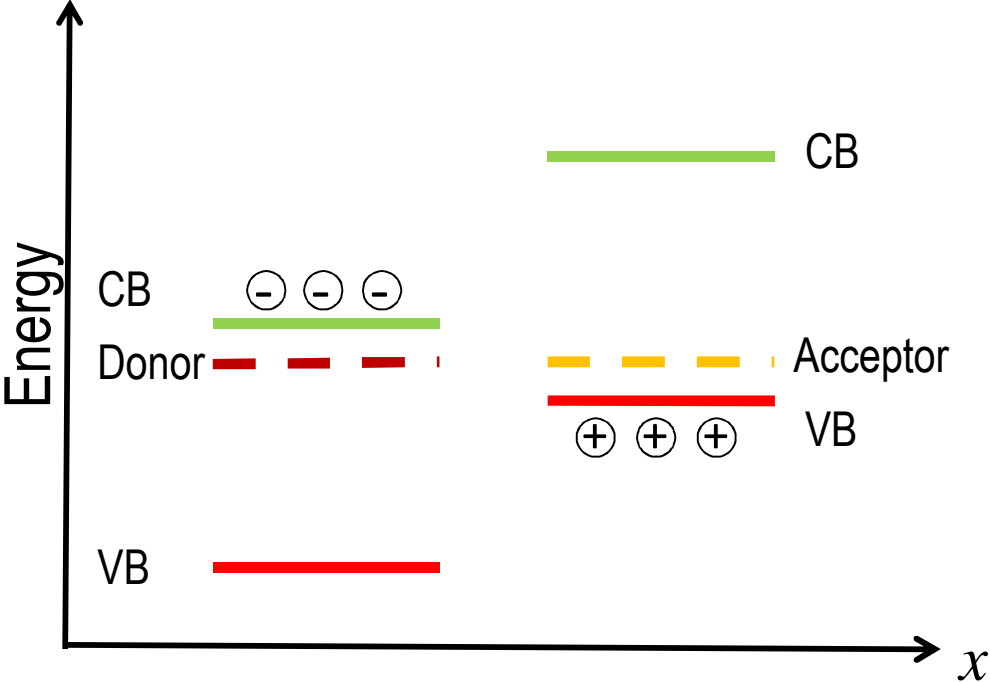


# Doping

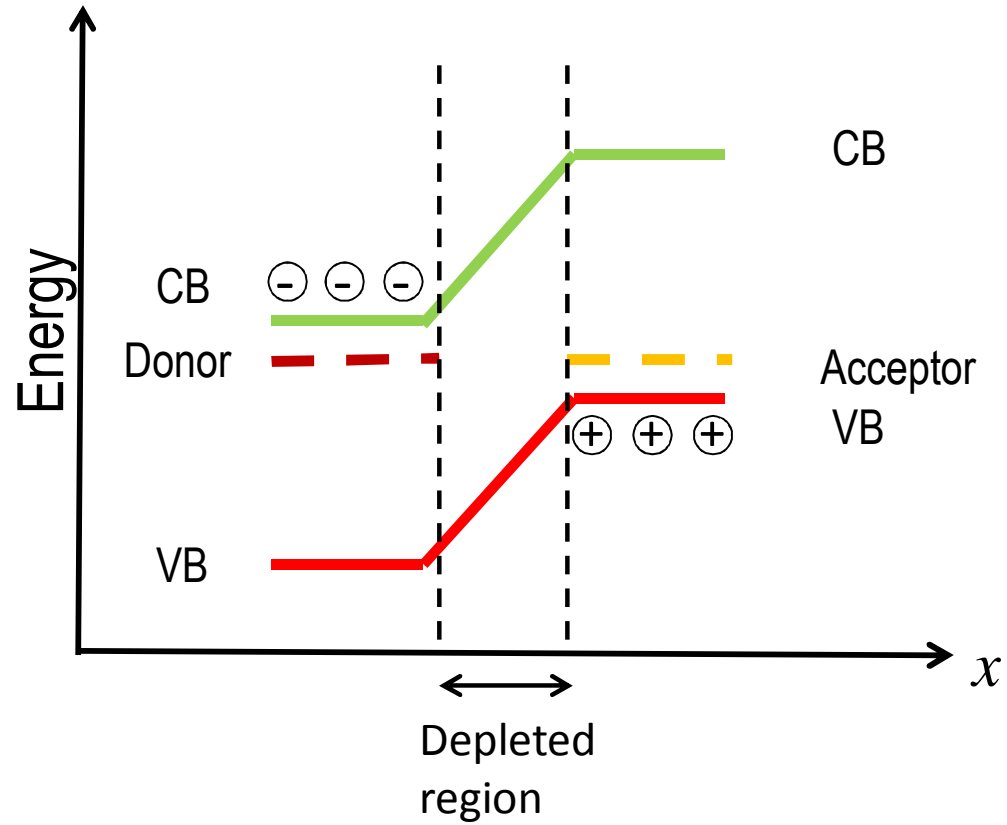
- Making **optoelectronic** devices requires the control of **free carrier density**.
- Addition of **extrinsic atoms**:
  - Impurity with one extra valence electron  $\Rightarrow$  *donor*
  - Impurity with one less valence electron  $\Rightarrow$  *acceptor*
- Typical impurity density  $\sim 10^{16}$ - $10^{18}$  cm<sup>-3</sup>



# PN junction

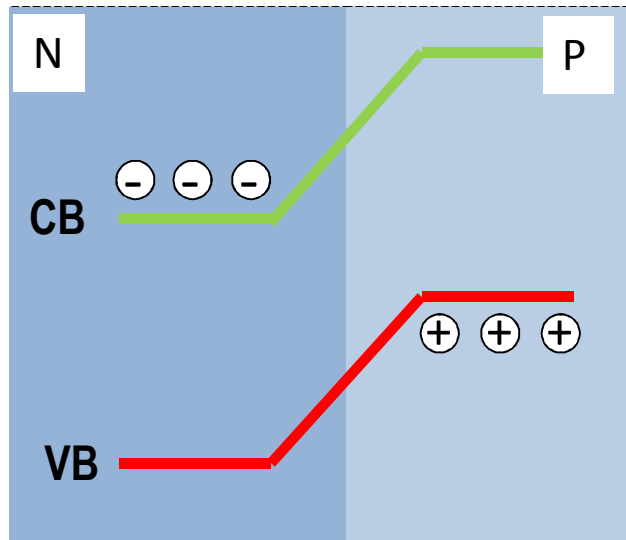


# PN junction



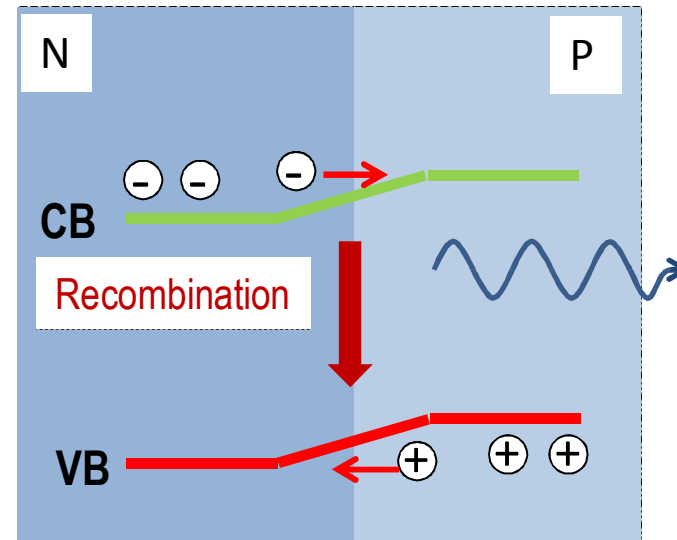
# PN Junction electroluminescence

Zero bias



- A **potential barrier** hinders the **diffusion** of electron and hole across the depletion zone.

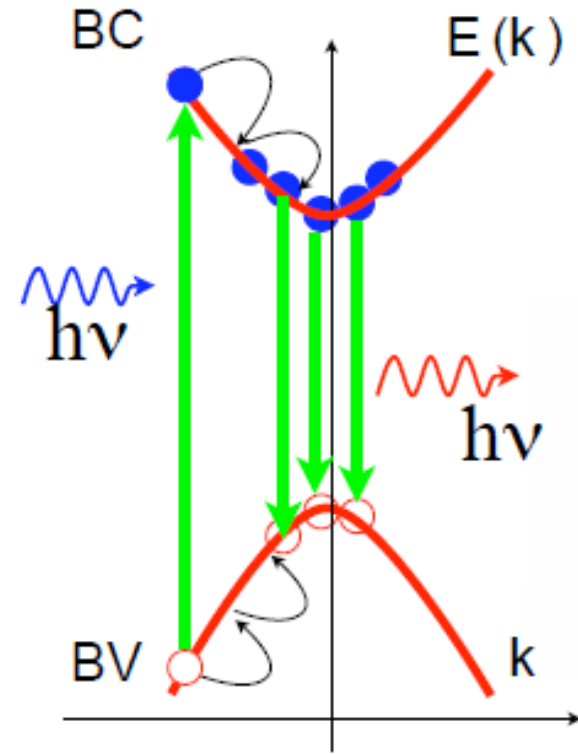
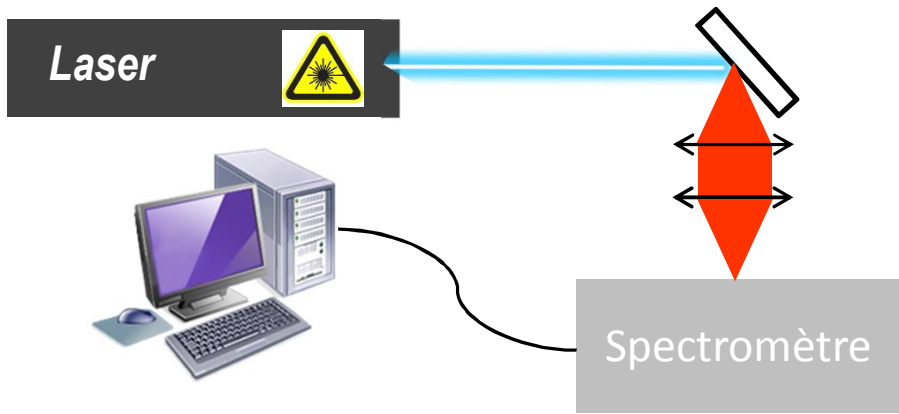
Forward bias



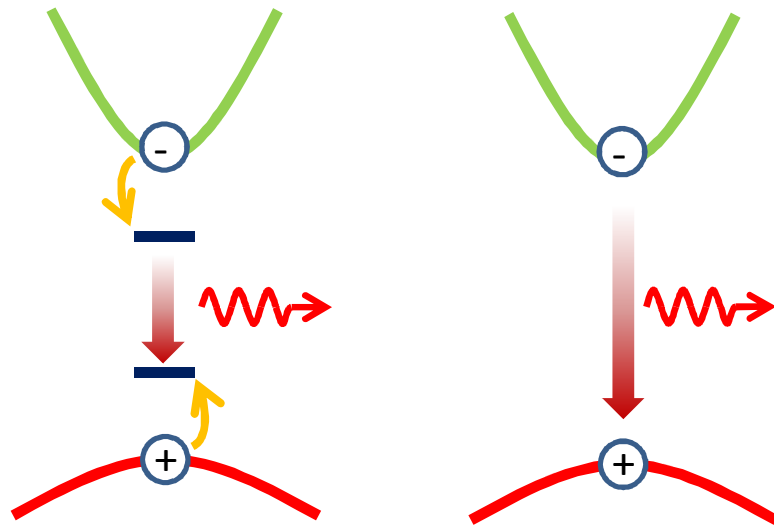
- **Electron from n region** and **holes from p region** diffuse and meet in the depletion zone where they **recombine** to **emit light**.

$$\text{Quantum efficiency} = \frac{\text{Number of emitted photons per unit time}}{\text{Number of injected electrons per unit time}}$$

# Photoluminescence

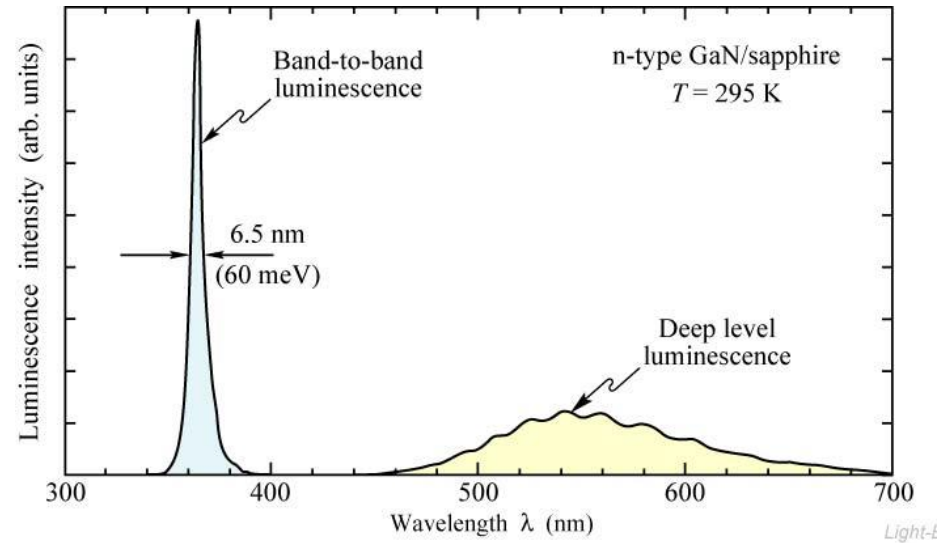


# Photoluminescence



Intrinsic PL  
Band to band

Extrinsic PL



⇒ *Photoluminescence reveals the crystal purity and gives information on the presence of doping impurities.*

# Lifetime and radiative efficiency

Soit une population de  $n$  paires électron-trou :

$$\frac{dn(t)}{dt} = -n(t) \left( \frac{1}{\tau_R} + \frac{1}{\tau_{NR}} \right)$$

*Probability of radiative recombination*

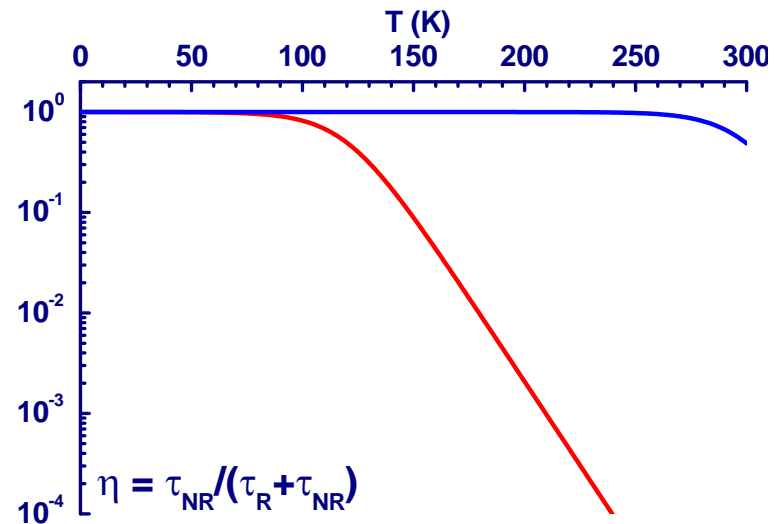
*Probability of non-radiative recombination*

Radiative efficiency :

$$\eta = \frac{1}{1 + \frac{\tau_R}{\tau_{NR}}}$$

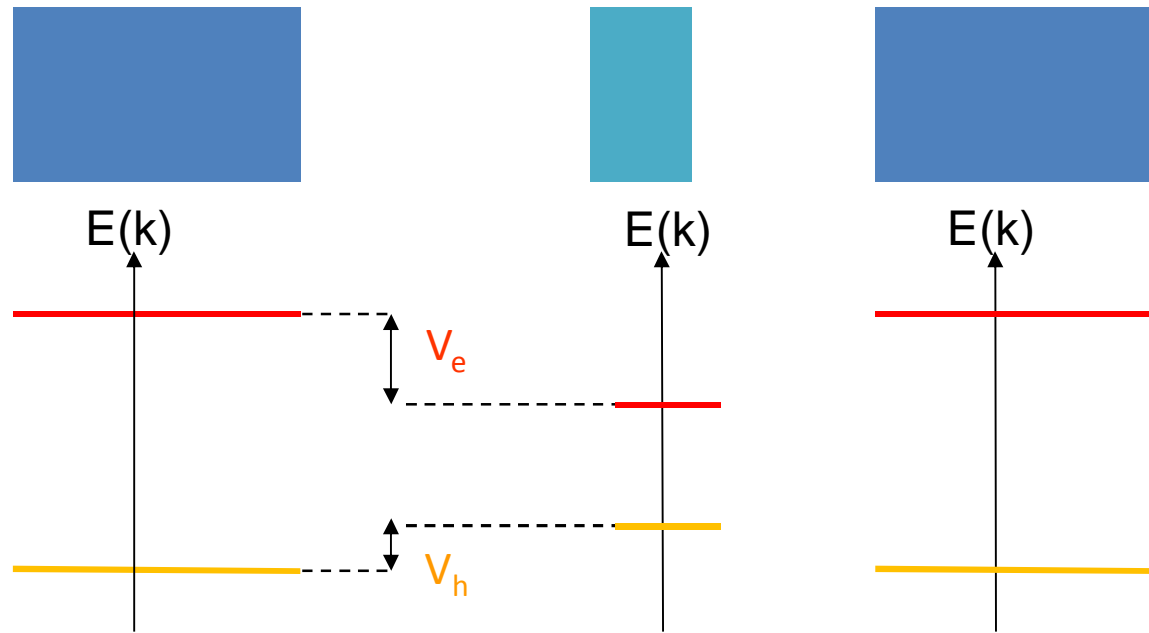
For light-emitting applications, need to maximize the radiative efficiency by decreasing  $\tau_R / \tau_{NR}$ :

- ⇒ *improve crystalline quality*
- ⇒ *Localize carriers (excitons)*



Characterizing and improving light emitters

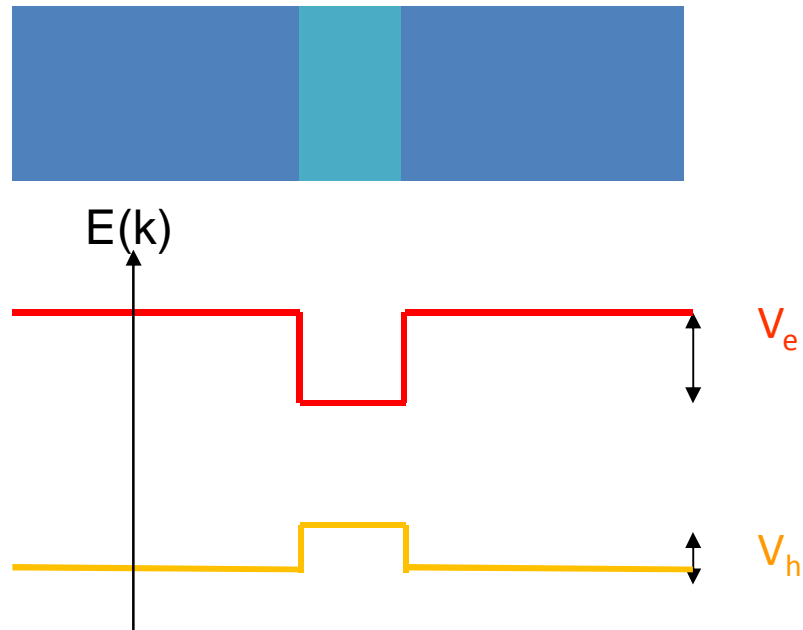
# Heterojunctions, quantum structures





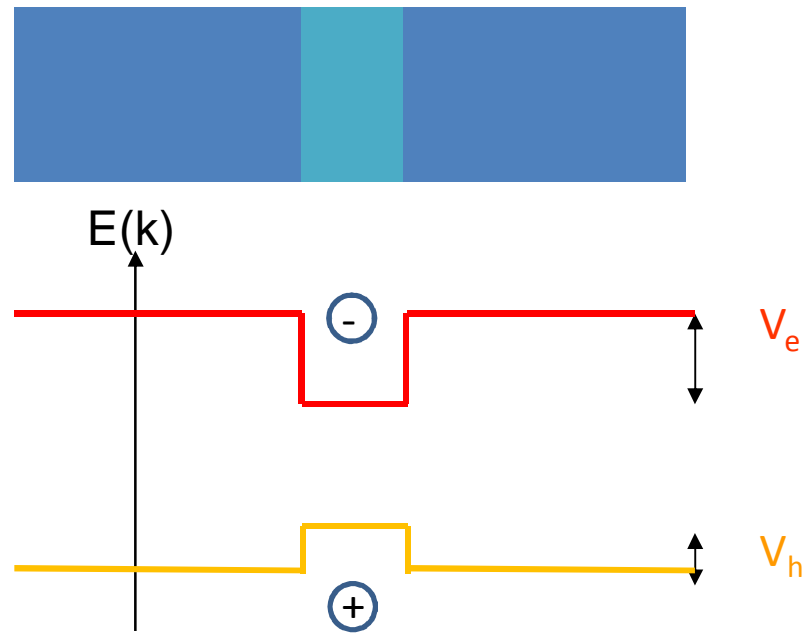
Characterizing and improving light emitters

# Heterojunctions, quantum structures



Characterizing and improving light emitters

# Heterojunctions, quantum structures



- ⇒ *Electrons and holes are confined in the same region of the sample.*
- ⇒ *Energies of optical transitions depend on the well width.*

# Electronic states in QWs

Bulk material: carriers free to propagate in 3D

$$\mathbf{k}=(k_x, k_y, k_z)$$

$$\psi_c(\vec{k}, \vec{r}) = \frac{1}{\sqrt{V}} e^{i\vec{k}\vec{r}} u_c(\vec{k}, \vec{r})$$

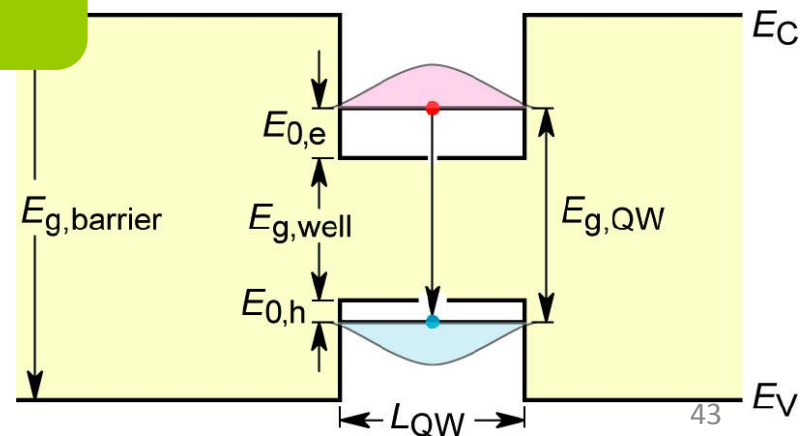
envelope  
= plane wave

Periodic part

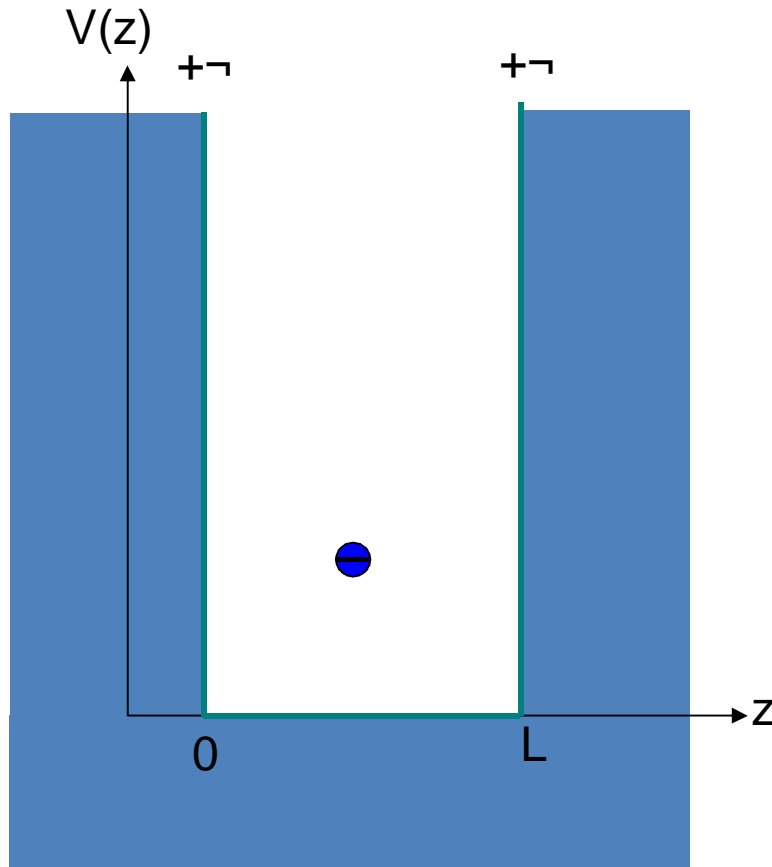
Nanostructure: The confinement potential breaks the translation symmetry in one or several direction.

$$\psi(\vec{k}, \vec{r}) = \frac{1}{\sqrt{N}} F(\vec{r}) u(\vec{k}, \vec{r}) \cong \frac{1}{\sqrt{N}} F(\vec{r}) u_c(0, \vec{r})$$

envelope  
= localized function



# Quantum well with infinite barriers



Envelope function equation :

$$\left[ \frac{\hbar^2}{2m^*} \frac{\partial^2}{\partial z^2} - E_n \right] f_n(z) = 0$$

Boundary condition :

$$f_n(0) = f_n(L) = 0$$

Solutions :

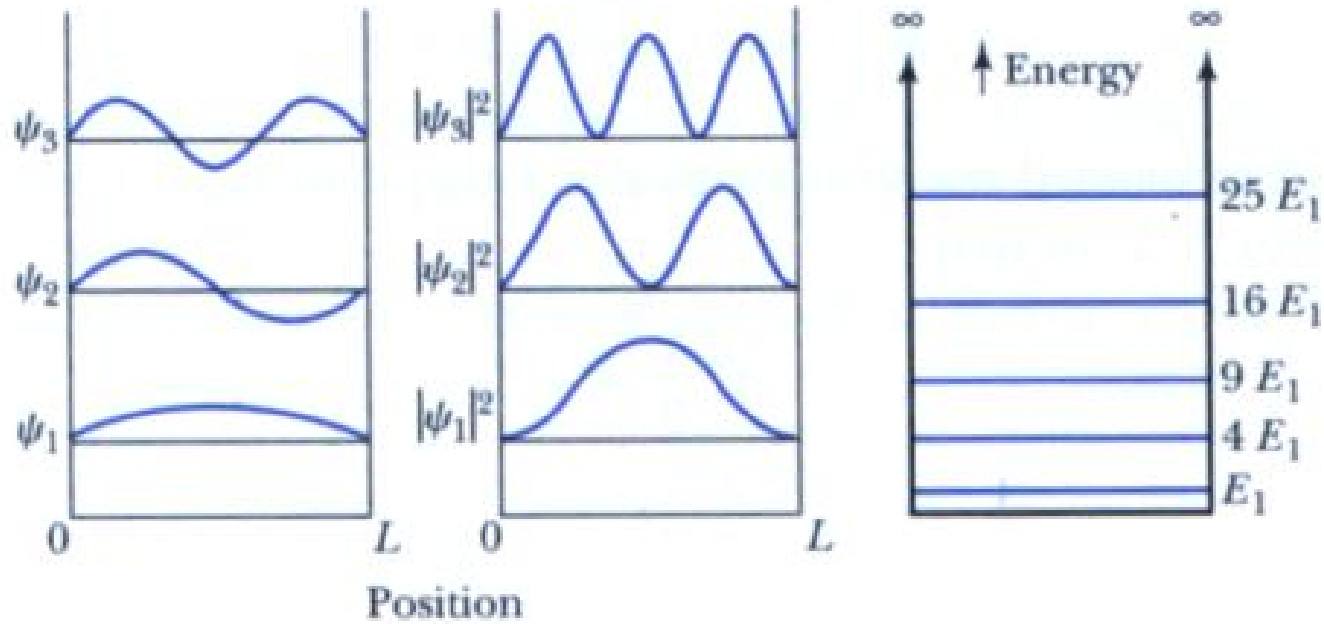
$$f_n(z) = \sqrt{\frac{2}{L}} \sin k_n z \quad \text{avec } k_n = n\pi/L$$

$$E_n = \frac{\hbar^2 k_n^2}{2m^*} = \frac{\hbar^2}{2m^*} \left( \frac{n\pi}{L} \right)^2$$

Confinement: discretization of allowed  $\mathbf{k}_z$  and discretization of energy states.

Characterizing and improving light emitters

# Envelope function and energy levels



# Optical transition in QWs

Fermi's golden rule

$$P_{i \rightarrow f \cong} = \frac{2\pi}{\hbar} \left| \langle f | p \mathcal{E} | i \rangle \right|^2 g(\hbar\omega) \cong \frac{2\pi}{\hbar} \left| \langle u_c | p \mathcal{E} | u_v \rangle \right|^2 \left| \int f_e(z_e) f_h(z_h) dz \right|^2 g(\hbar\omega)$$

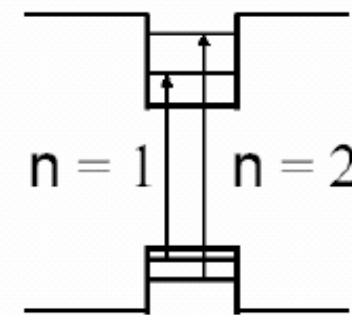
Envelope function overlap

QW with infinite barriers

$$\int f_n^e(z_e) f_m^h(z_h) dz = \frac{2}{L} \int_0^L \sin(k_n z) \sin(k_m z) dz = \delta_{n,m}$$

Allowed transition :  $\Delta n = 0$

conduction band



valence band

# Excitons in QWs

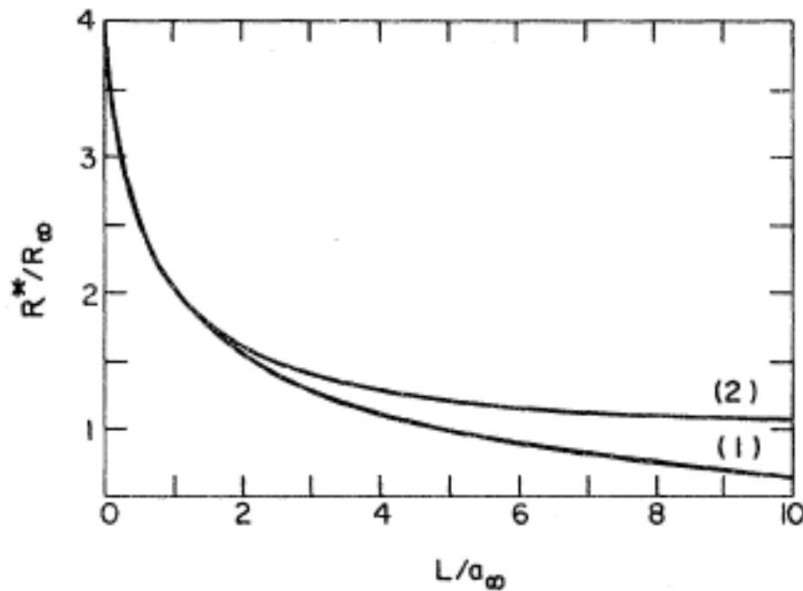
2D hydrogenoid function

$$\psi_1 = N_1 \cos \frac{\pi z_e}{L} \cos \frac{\pi z_h}{L} \exp \left[ \frac{-\rho}{\lambda_1} \right]$$

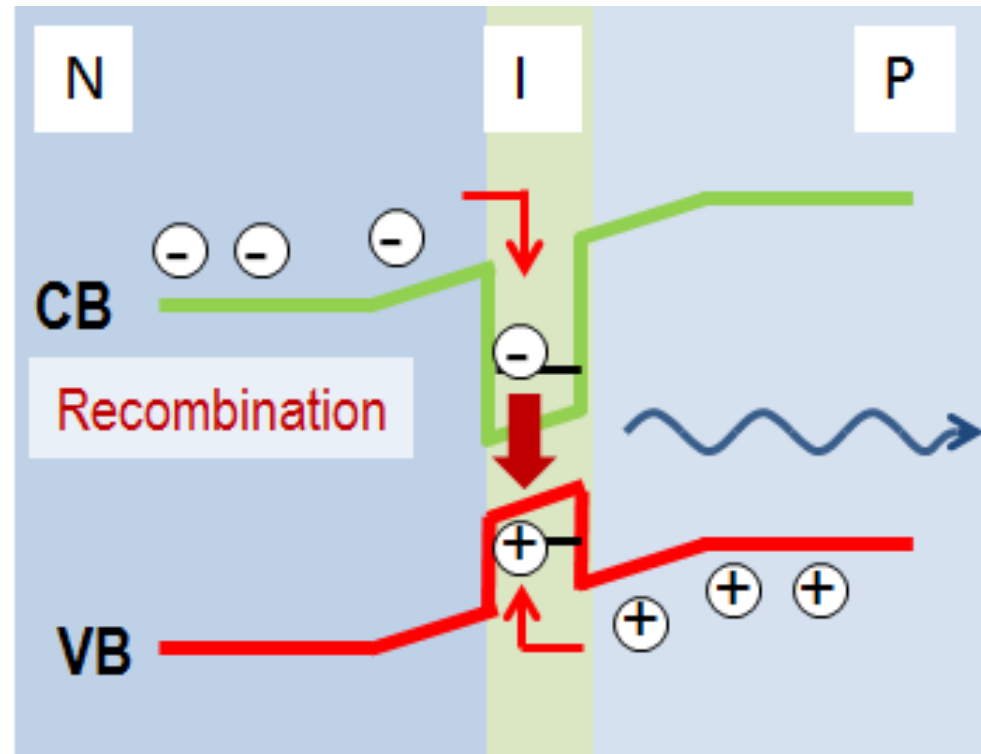
Quantum confinement enhances the exciton binding energy.

Energy spectrum of bound excitonic states

$$E_n = - R_y / (n - 1/2)^2$$



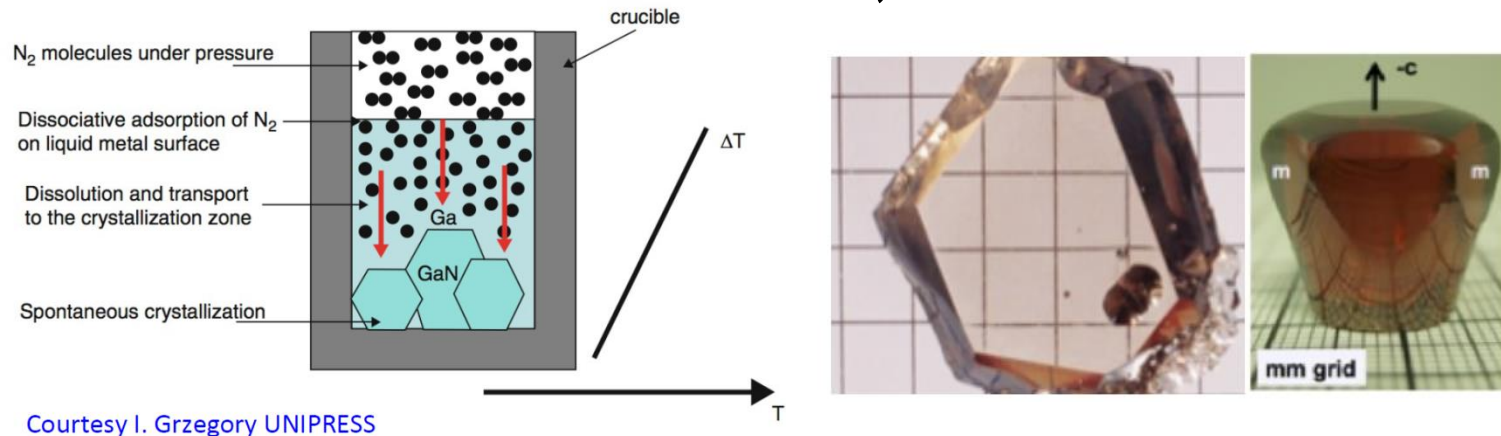
# Quantum well LED





# Growth of bulk GaN

- GaN is **extremely complicated to synthesize**.
- Growth of bulk GaN : 1600°C, 20kbar.

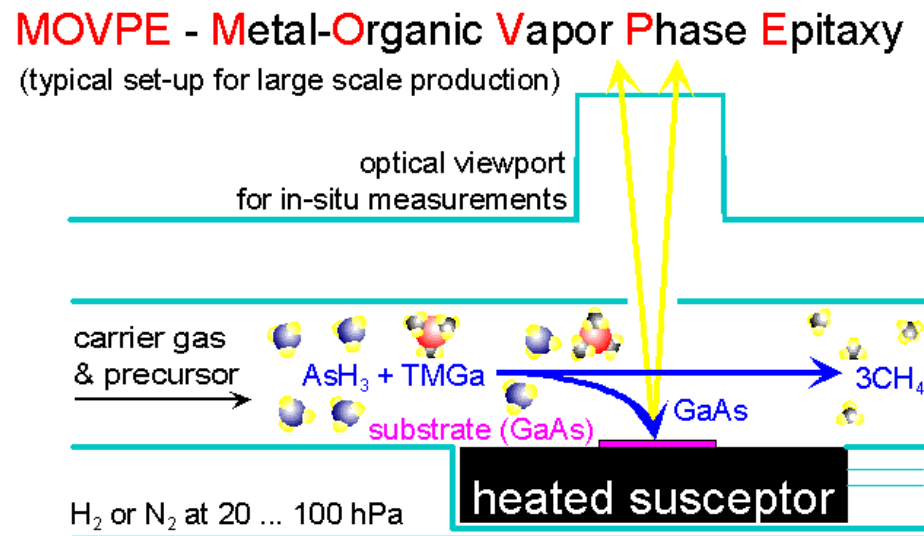


- **Hetero-epitaxy:**

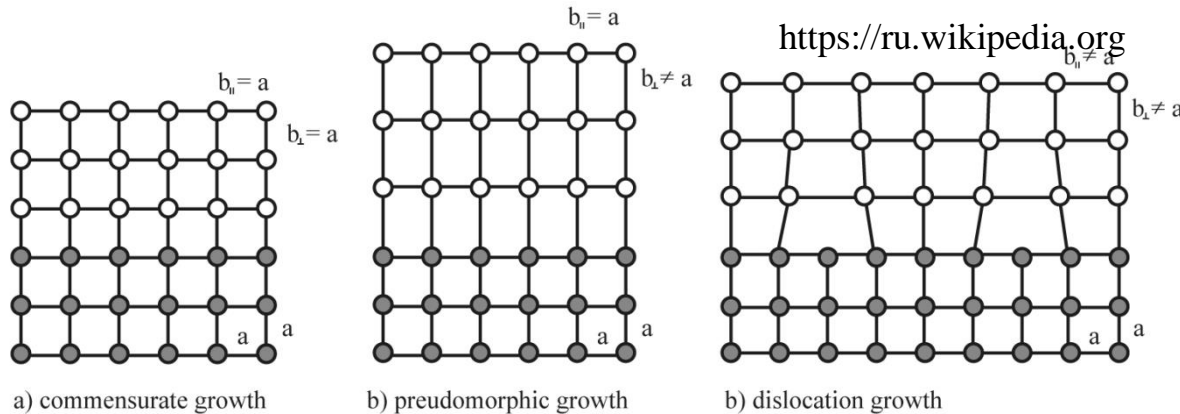
⇒ In a reactor, the chemical constituents of the crystal are deposited on a heated **monocrystalline substrate** that acts as a **germ** to initiate an **oriented crystalline growth**.

# Metal-Organic chemical vapor deposition

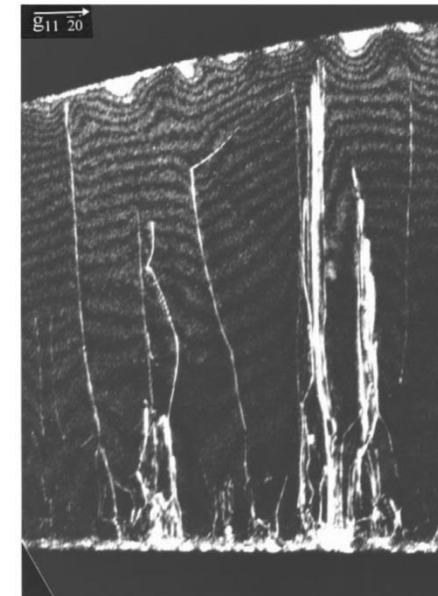
- The principle is to **pyrolyze metal-organic compounds** (triethyl-gallium or trimethyl-gallium) and hydrides ( $\text{NH}_3$ ,  $\text{SiH}_4$ ) in the flow of a carrier gas ( $\text{N}_2$  or  $\text{H}_2$ ) in the vicinity of a heated substrate.



# Hetero-epitaxy



	GaN	Al <sub>2</sub> O <sub>3</sub>	6H-SiC	Si
<b>a(Å)</b>	3.189	4.758	3.08	5.430
<b>a(T<sup>-1</sup>)</b>	5.6	7.5	4.2	3.59



N. Grandjean, J. Massies, P. Vennéguès *et al.*, *J. Appl. Phys.* **83**, 1379 (1998)

- **No matching substrate**

- ⇒ **Strained** layers

- ⇒ Structural defects (**dislocations**)

- ⇒ *Dislocations are known to act as non-radiative centers in semiconductors.*

# Optical investigations

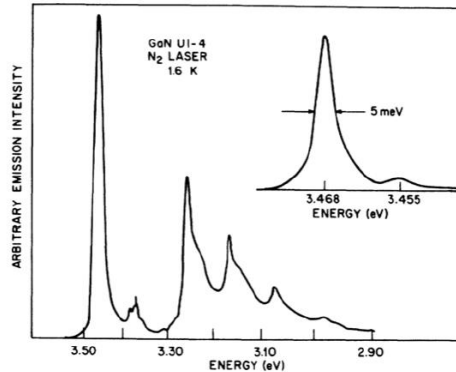
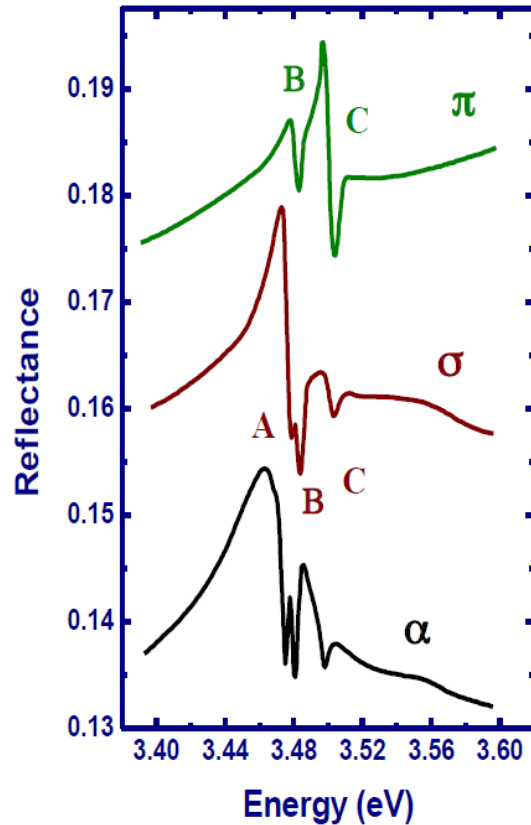
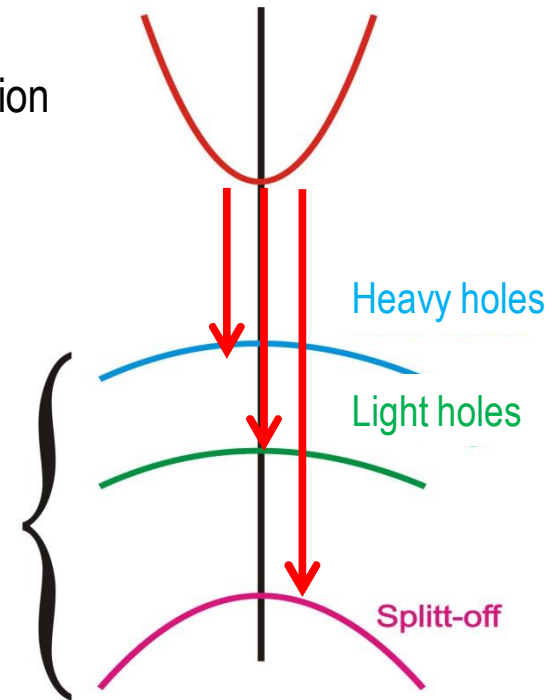


FIG. 5. Typical photoluminescence spectrum of type-I GaN at 1.6 K. The insert is a more detailed view of the strong high-energy peak near 3.468 eV, taken with low excitation intensity and high resolution.

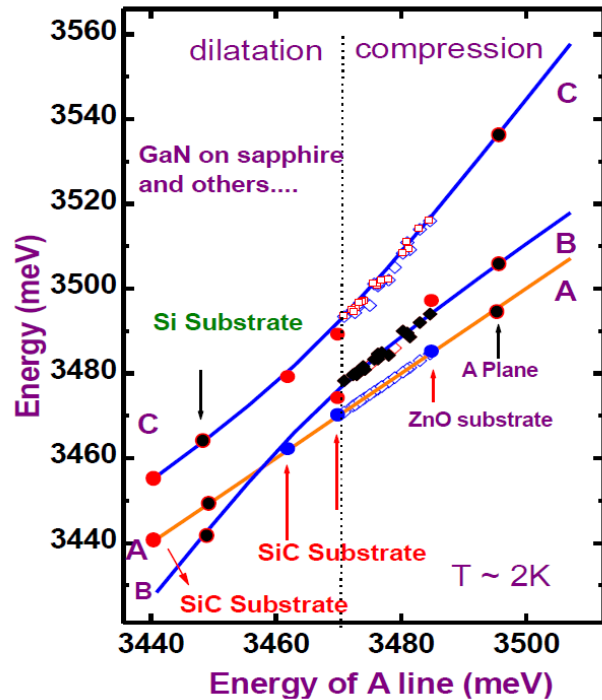
Conduction band

Valence bands

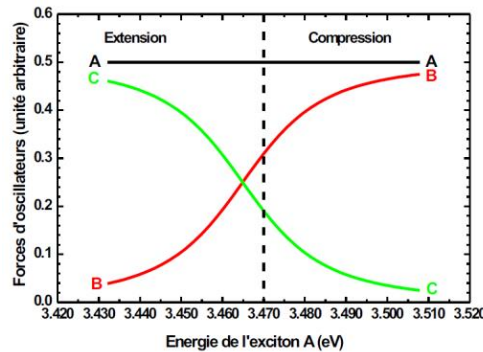


- Band gap measurement  $E_g=3,48$  eV.
- Observation of excitonic transitions.
- Valence band structure.

# Strain effect



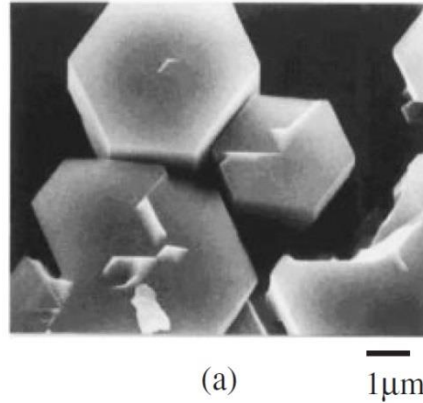
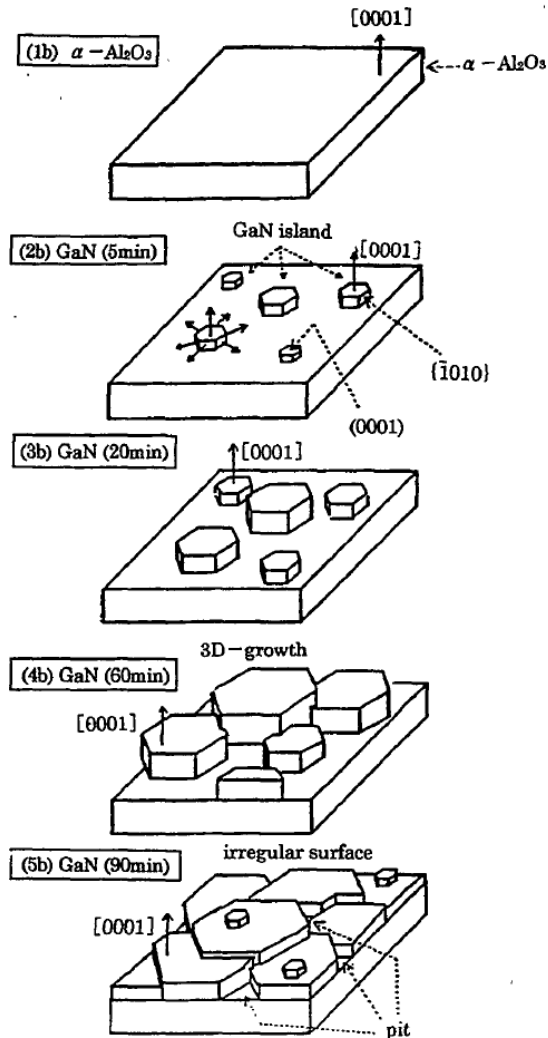
B. Gil, F. Hamdani and H. Morkoç *Phys. Rev. B* **54**, 7678 (1996)



## ■ Strain effect:

- Shifts the **energy bands**.
- Modifies the **coupling-strength** between excitons and light.
- Observable **in optical experiments**.

# Growth of GaN by MOVPE



- Hetero-epitaxy on sapphire (Al<sub>2</sub>O<sub>3</sub>)
  - Island growth.
  - Rough polycrystalline layers, large density of structural defects.
  - Strong residual n-doping.

I. Akasaki and H. Amano, *Jpn. J. Appl. Phys.* **36** 5393 (1997)

# Buffer layer

1) AlN buffer layer



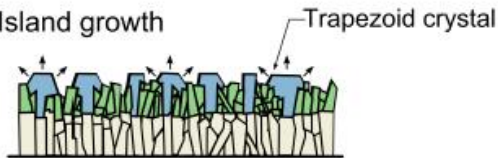
2) Nucleation of GaN



3) Geometric selection



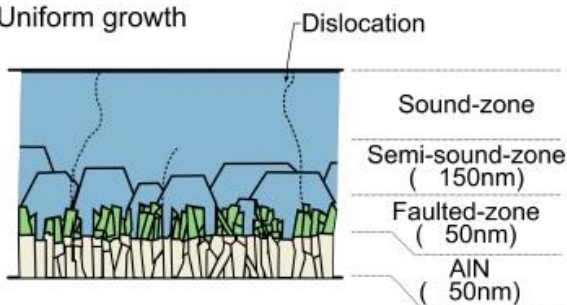
4) Island growth



5) Lateral growth

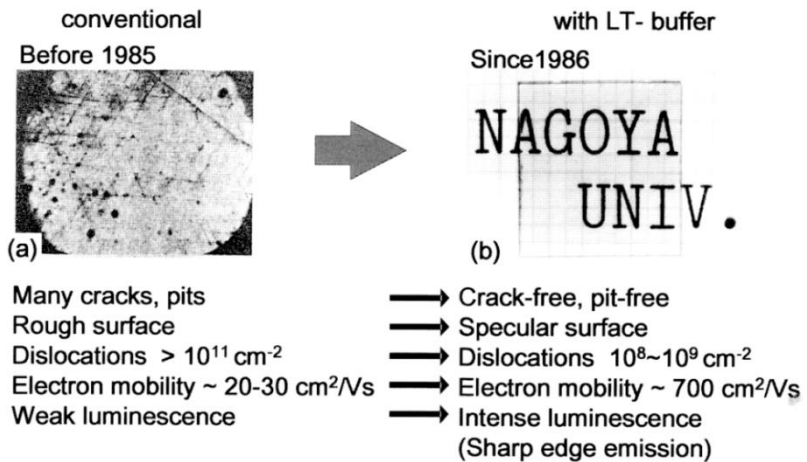


6) Uniform growth



## Benefits:

- Supply of a high-density nucleation centers having the same orientation as the substrate.
- Promotion of the lateral growth of the epitaxial film due to the decrease in interfacial free energy between the substrate and the epitaxial film.
- Augmentation of grain size, diminution of dislocation density.





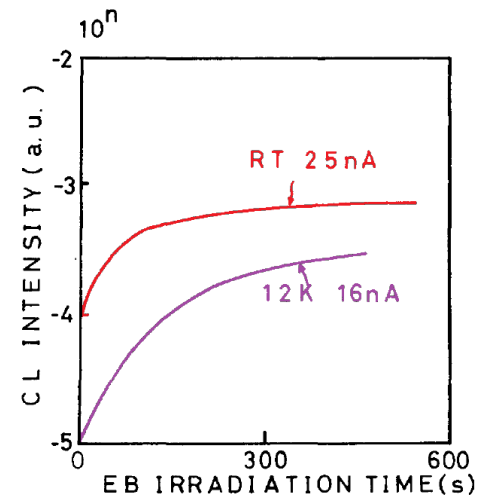
# Control of p- doping

- Using AlN buffer layers, the residual donor density was drastically decreased.
  - ⇒ *What about p-doping?*
- First attempts to incorporate p-impurities, such as Zn, showed no change in resistivity measurements.



# Doping activation

- Using AlN buffer layers, the residual donor density was drastically decreased.
  - ⇒ *What about p-doping?*
- First attempts to incorporate p-impurities, such as Zn, showed no change in resistivity measurements.
- In 1987, H. Amano found that the intensity of Zn-related luminescence increases when Zn-doped GaN is irradiated with electron beam during CL measurement.
- Activation of Zn acceptors that were added during growth.



H. Amano, I. Akasaki, T. Kozawa, K. Hiramatsu, N. Sawaki, K. Ikeda and Y. Ishii, *J. Lum.* **40&41**, 121 (1988)

H. Amano, M. Kito, K. Hiramatsu and I. Akasaki, *Jpn. J. Appl. Phys.* **28**, L2112 (1989)

# Doping activation

- In 1992, S. Nakamura found that a thermal annealing at 800°C enables to reduce the resistivity of Mg-doped GaN layers.

**Interpretation:** acceptors are passivated by an hydrogen excess. Post-growth annealing breaks the Mg-H bound and activates the doping.

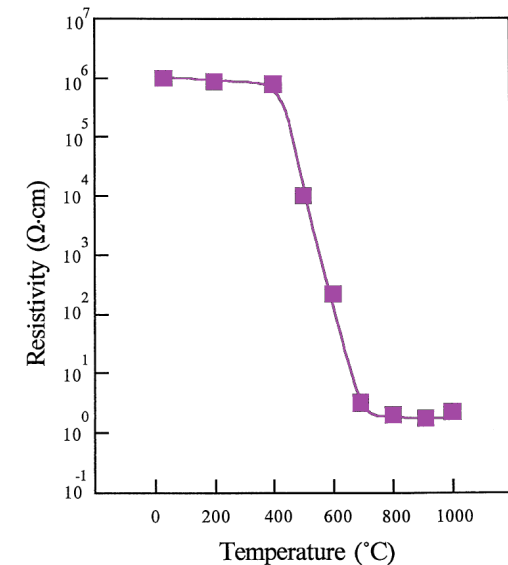


Fig. 1. Resistivity of Mg-doped GaN films as a function of annealing temperature.

S. Nakamura, T. Mukai, M.Senoh and N. Iwasa, [Jpn. J. Appl. Phys. 31, L 139-L 142 \(1992\)](#)

S. Nakamura, N. Iwasa, M.Senoh and T. Mukai, [Jpn. J. Appl. Phys. 31, 1258-1266 \(1992\)](#)

The blue diode

# First GaN-based UV diode

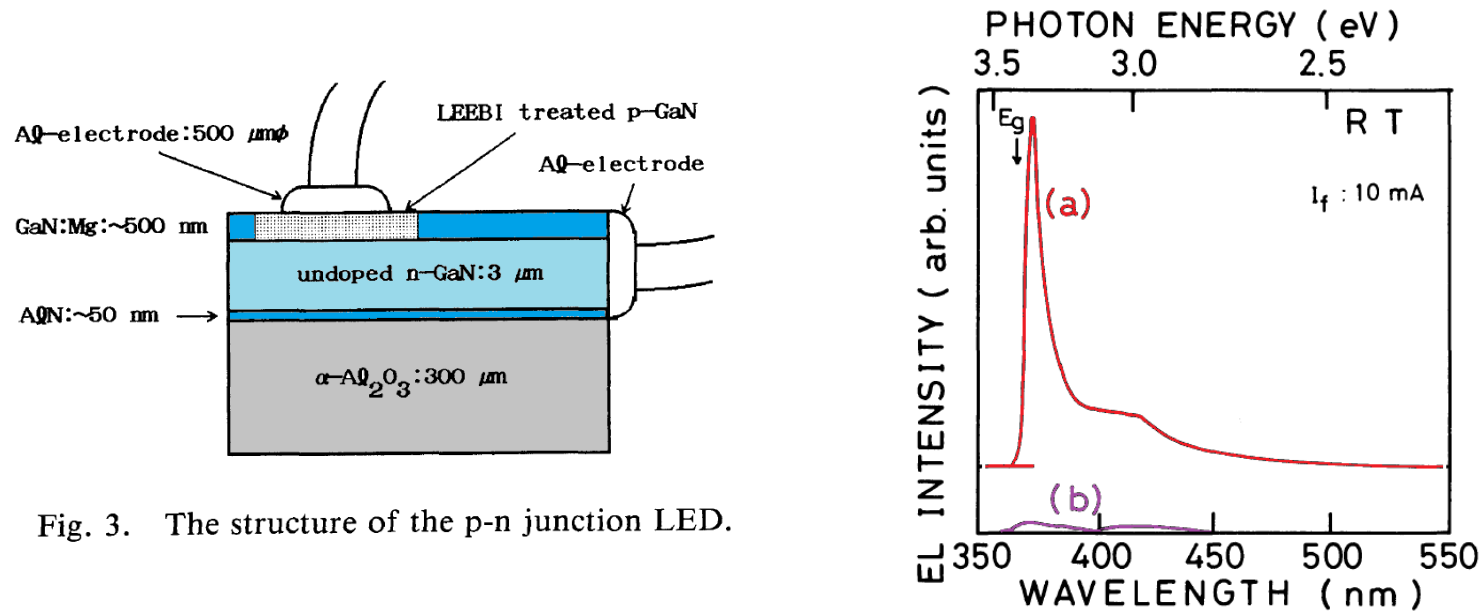


Fig. 3. The structure of the p-n junction LED.

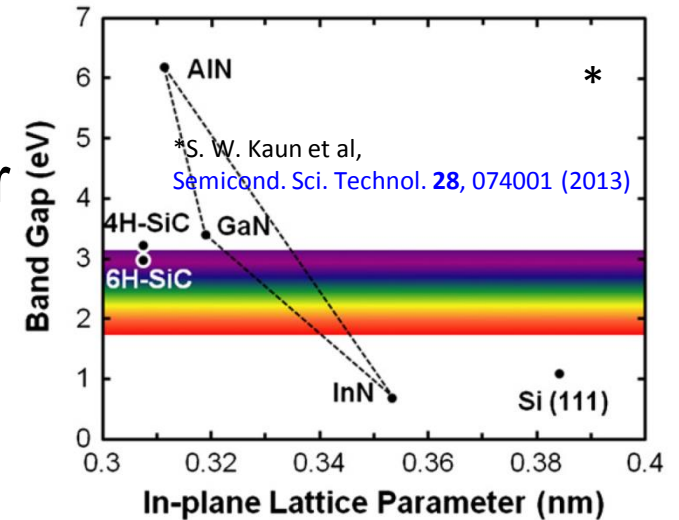
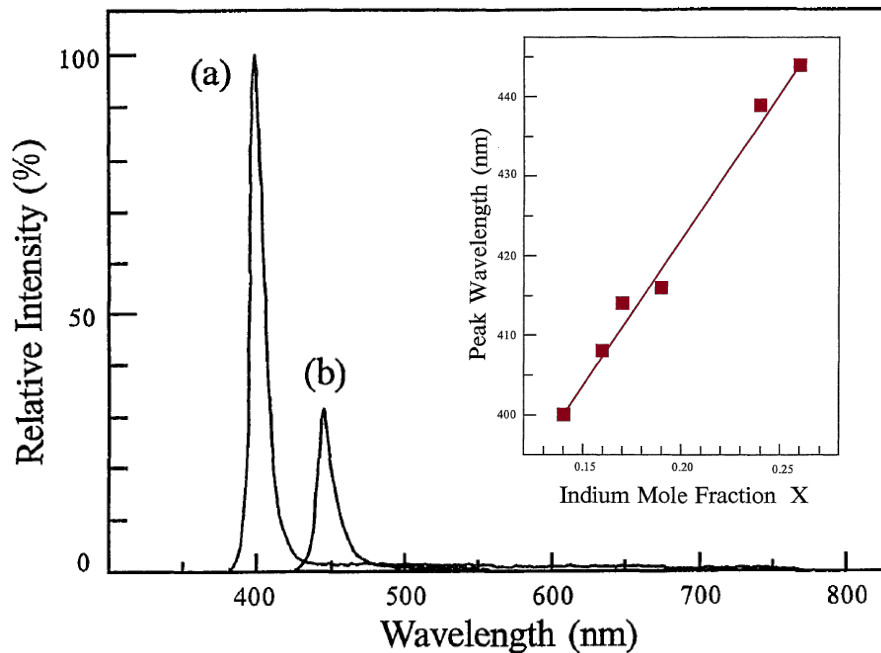
- Built around a GaN:Si–GaN:Mg homojunction.
- Emission wavelength ~380 nm at room temperature.  
⇒ *Need to decrease the bandgap energy to achieve blue emission.*

H. Amano, M. Kito, K. Hiramatsu and I. Akasaki, *Jpn. J. Appl. Phys.* 28, L2112 (1989)

The blue diode

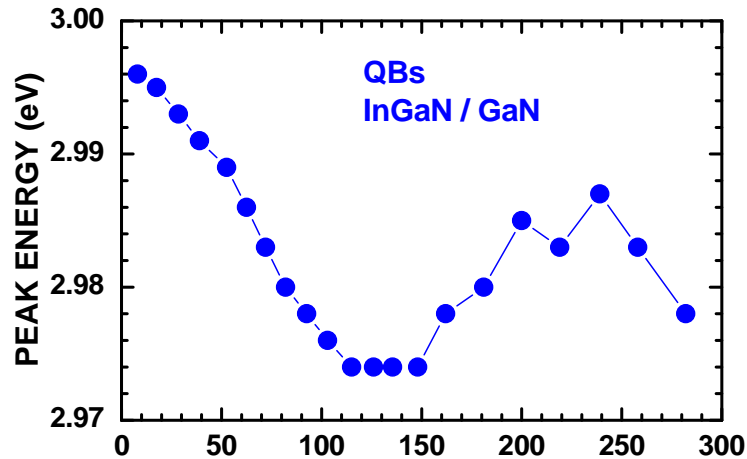
# How blue can you get?

- Adding **indium** in GaN should shift the bandgap towards the **visible range**.
- A blue emission at  $\sim 450$  nm needs to master the growth the alloy  $\text{In}_x\text{Ga}_{1-x}\text{N}$  for concentrations around **25-30%**..

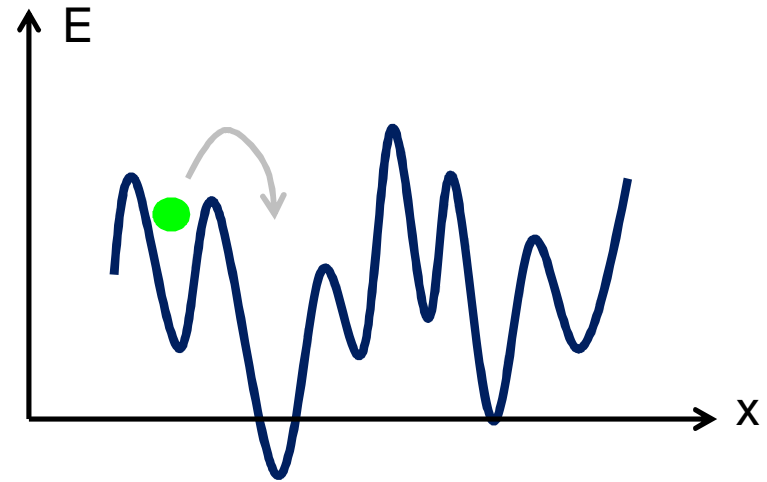
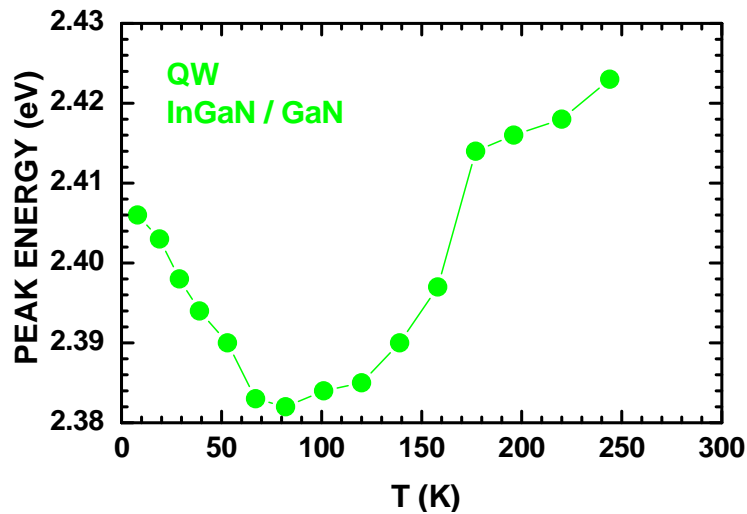


- In 1991, S. Nakamura develop a specific skill to growth InGaN using MOVPE.
- Strong photoluminescence whose wavelength depends on the indium fraction.

# Advantage of InGaN



- The radiative efficiency of InGaN layers and quantum wells are not affected by dislocations.
- Localization of electron-hole pairs on potential fluctuations due to alloy disorder.
- Confirmed by the evolution of photoluminescence with T.



# First blue diode

## P-GaN/N-InGaN/N-GaN Double-Heterostructure Blue-Light-Emitting Diodes

Shuji NAKAMURA, Masayuki SENOH and Takashi MUKAI

Department of Research and Development, Nichia Chemical Industries, Ltd., 491 Oka, Kaminaka, Anan, Tokushima 774

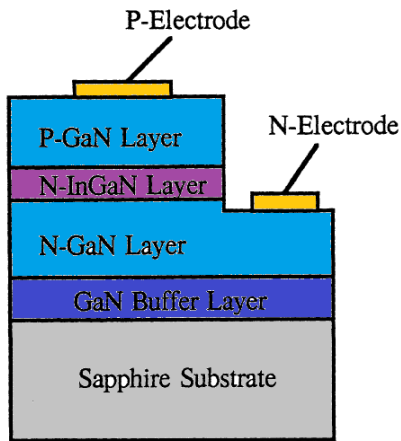


Fig. 1. The structure of the p-GaN/n-InGaN/n-GaN double-heterostructure blue LED.

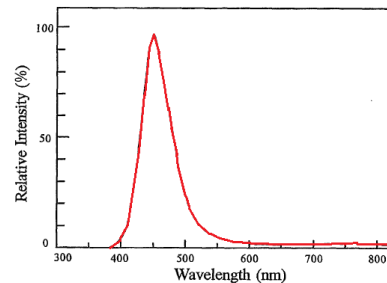


Fig. 5. Room-temperature photoluminescence spectrum of the top layer (Mg-doped p-type GaN layer) of the p-GaN/n-InGaN/n-GaN double-heterostructure after the 15 kV electron-beam irradiation treatment.

Quantum efficiency~0.2%

## Candela-class high-brightness InGaN/AlGaIn double-heterostructure blue-light-emitting diodes

Shuji Nakamura, Takashi Mukai, and Masayuki Senoh

Department of Research and Development, Nichia Chemical Industries, Ltd., 491 Oka, Kaminaka, Anan, Tokushima 774, Japan

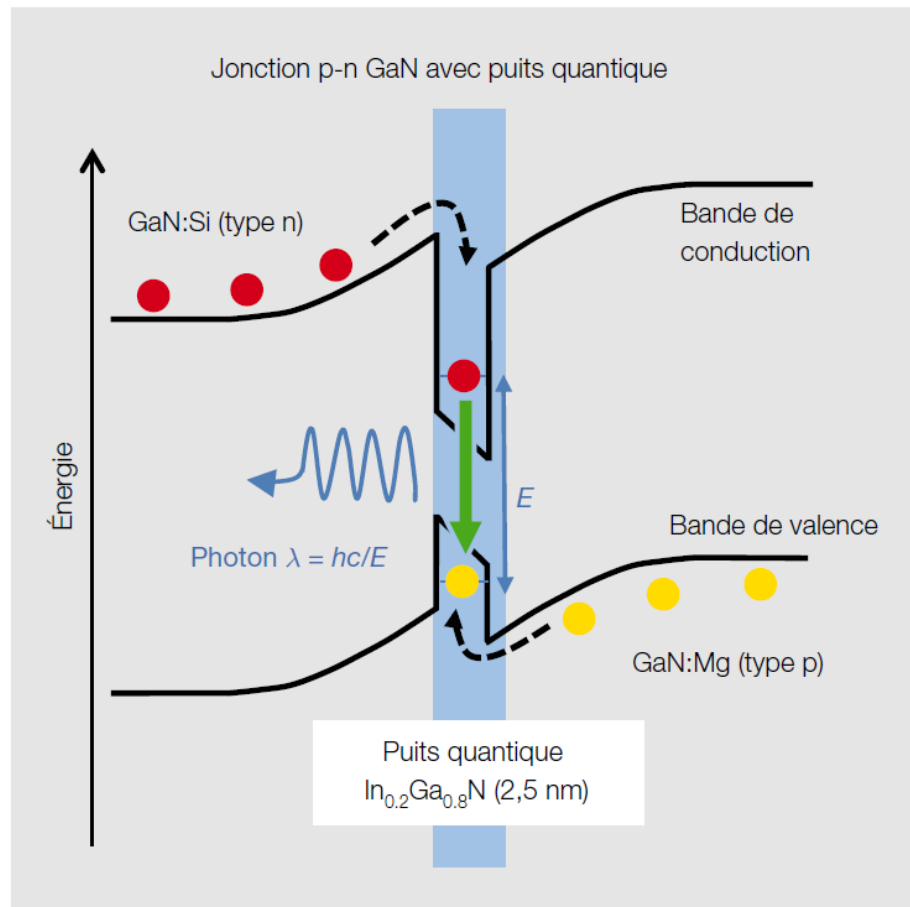
(Received 2 December 1993; accepted for publication 5 January 1994)

Quantum efficiency~3%

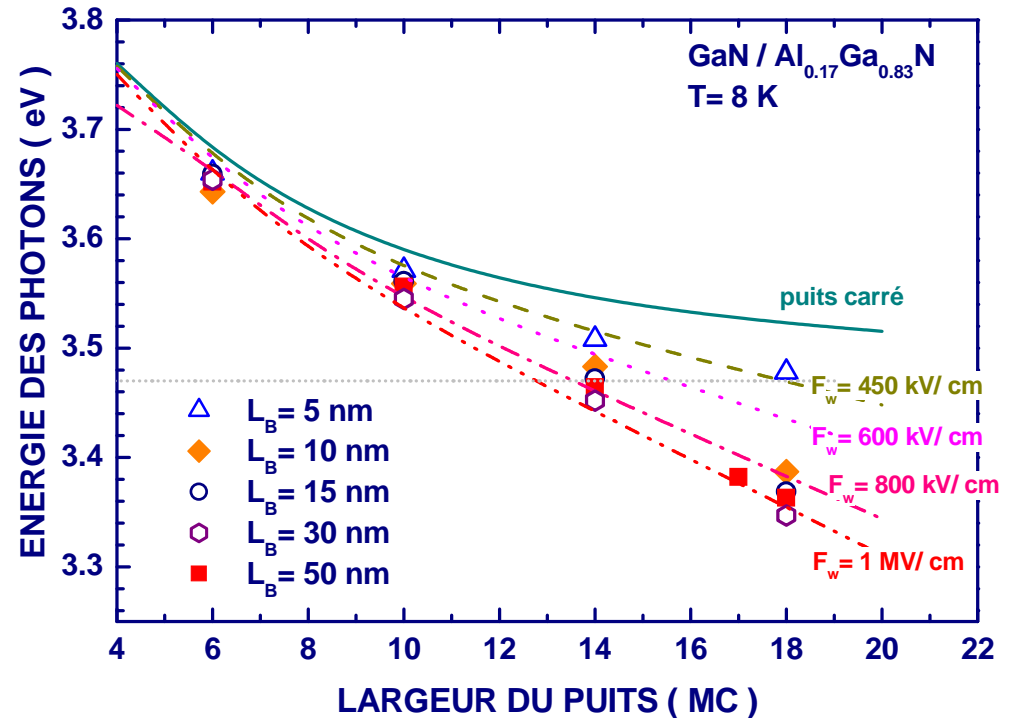
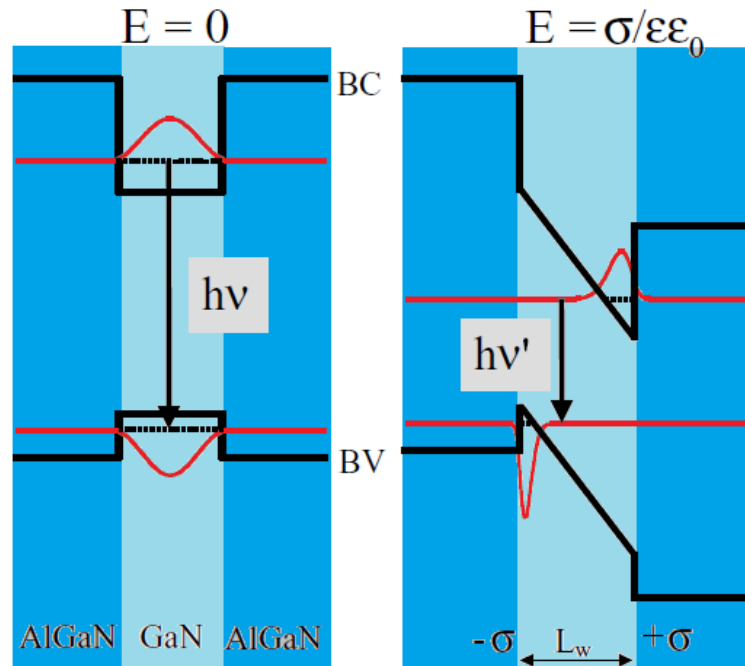
⇒ *Double heterostructure.*

⇒ *The next generation of components embeds quantum wells.*

# Quantum wells



# Quantum wells & quantum dots

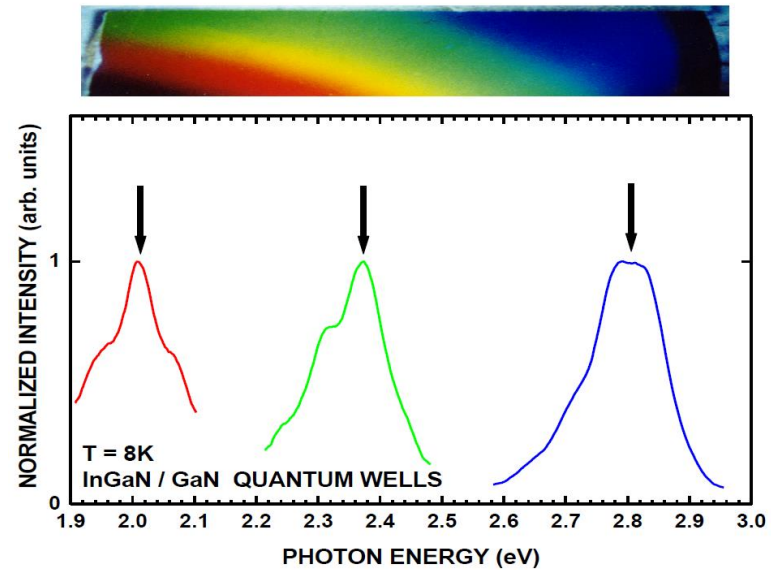
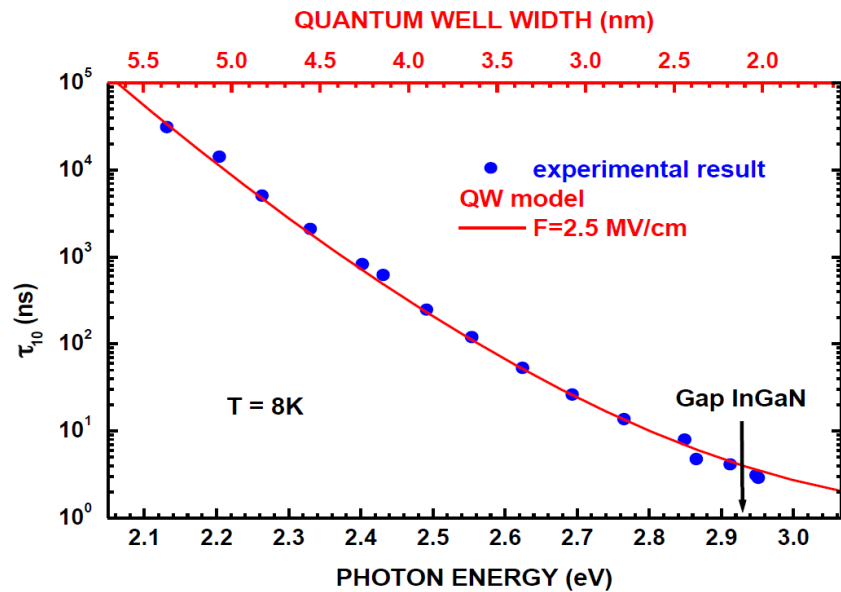


- Because of their **wurtzite structure**, nitride materials are piezoelectric.
- Existence of **built in polarization fields** in quantum wells that **separate electrons** and **holes** along the growth axis.

⇒ *Quantum confined Stark-effect.*

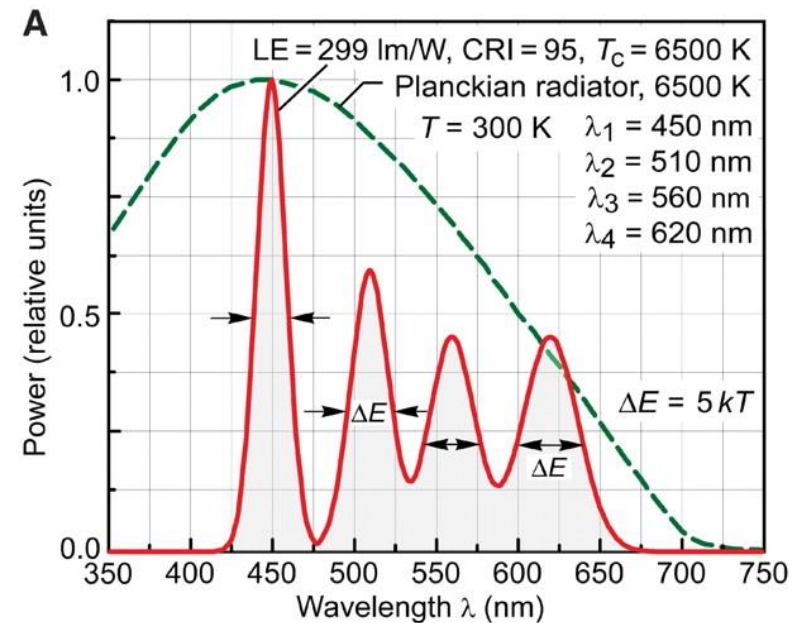
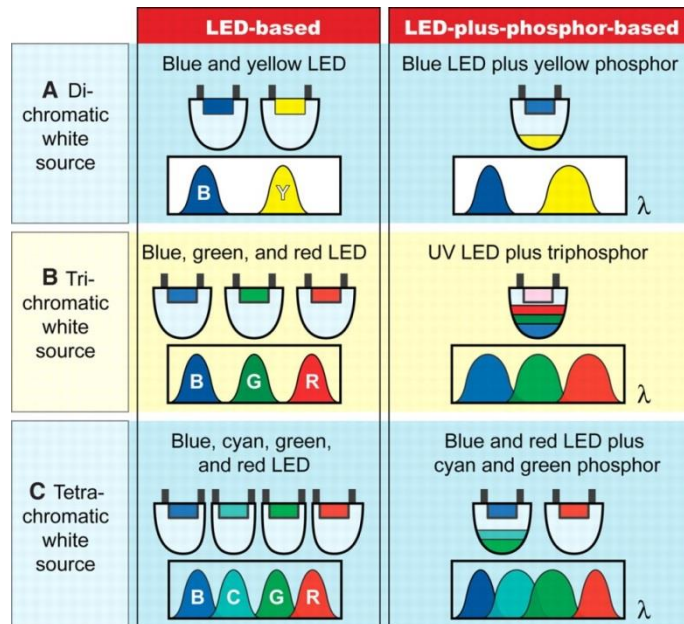


# Effect of built-in electric field



P. Lefebvre, A. Morel, M. Gallart, T. Taliercio, J. Allègre, B. Gil, H. Mathieu, B. Damilano, N. Grandjean and J. Massies, *Appl. Phys. Lett.* **78** (9), 1252-1254 (2001)

# Improving the CRI



*E. .F Schubert and J. K. Kim, Science 27 May 2005:  
vol. 308 no. 5726 1274-1278*

- Nitride based nanostructures are limited to light emission.
- Their wide bandgap and strong exciton binding energy makes them interesting for room temperature operation.
- The present degree of control of GaN-based heterostructures allows to consider a physics that was previously mostly restricted to conventional III-V.

- Nitride based nanostructures are limited to light emission.
- Their wide bandgap and strong exciton binding energy makes them interesting for room temperature operation.
- The present degree of control of GaN-based heterostructures allows to consider a physics that was previously mostly restricted to conventional III-V.

⇒ *Here are some examples.*

# Spin dynamics

## Quantum dots

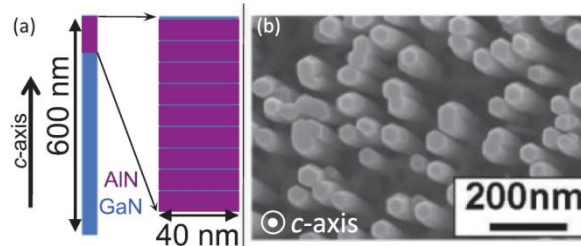
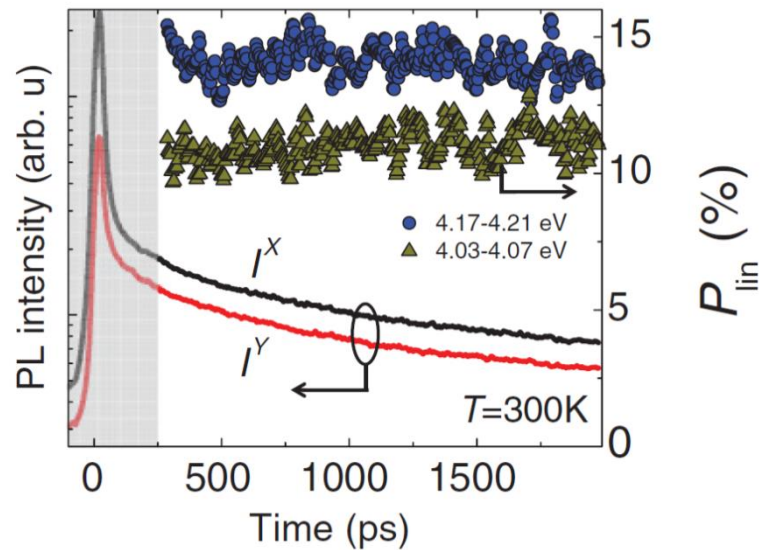
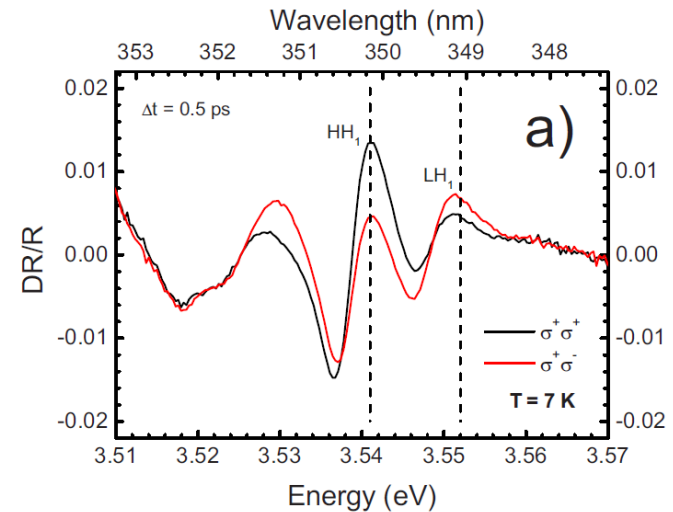
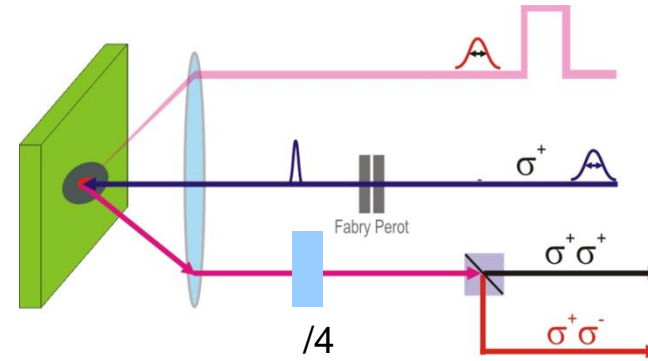


FIG. 1. (Color online) (a) Schematic representation of a single nanowire. (b) Top view SEM image of the nanowires.



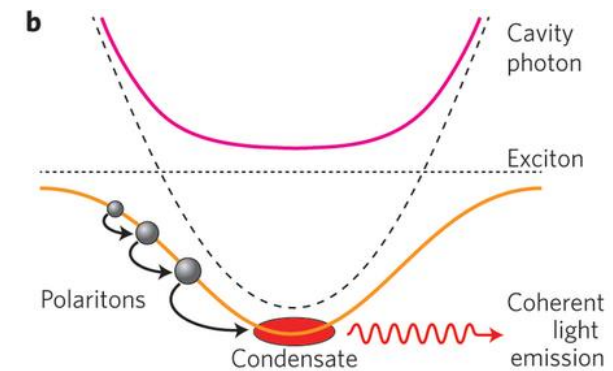
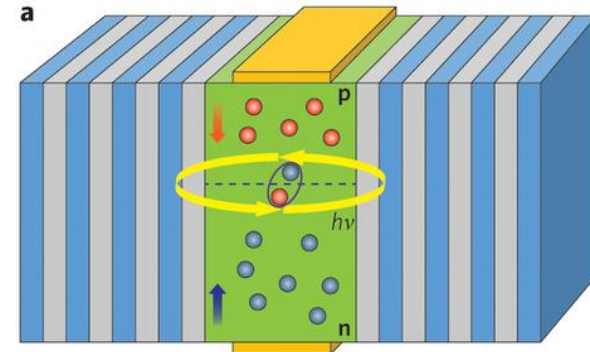
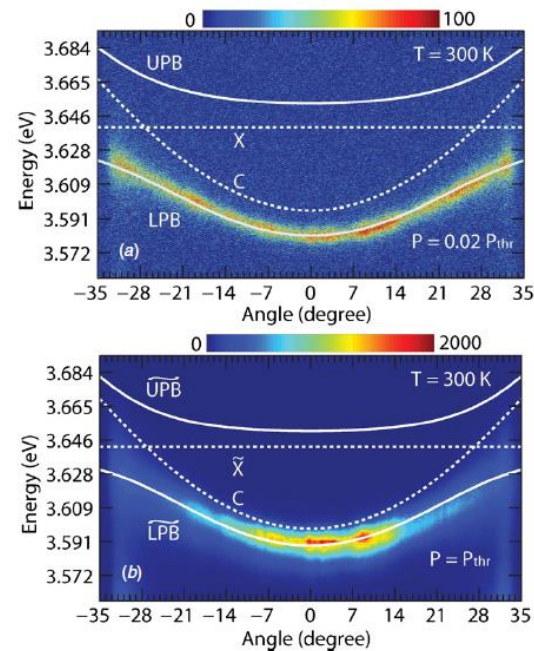
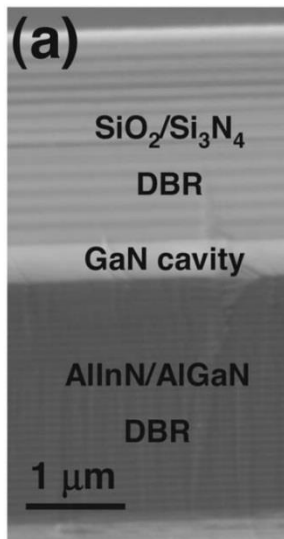
## Quantum wells



A. Balocchi, J. Renard, C. T. Nguyen, B. Gayral, T. Amand, H. Mariette, B. Daudin, G. Tourbot and X. Marie, *Phys. Rev. B* **84**, 235310 (2011)

J. Besbas, A. Gadalla, M. Gallart, O. Crégut, B. Hönerlage, P. Gilliot, E. Feltin, J. F. Carlin, R. Butte, N. Grandjean, *Phys. Rev. B* **82**, 195302 (2010)

# Polariton laser



R. Butté, G. Christmann, E. Feltn, J.-F. Carlin, M. Mosca, M. Illegems, and N. Grandjean, [Phys. Rev. B \*\*73\*\*, 033315 \(2006\)](#)

R. Butté and N. Grandjean, [Semicond. Sci. Technol. \*\*26\*\* \(2011\) 014030 \(9pp\)](#)



# Single photon emitters

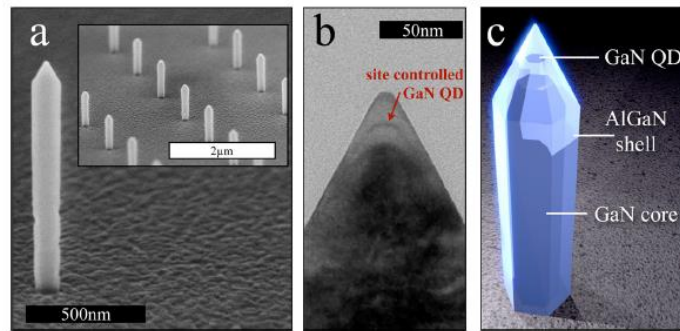


Figure 1. Images of site-controlled nanowire-QDs. (a) SEM image showing a single nanowire grown on a patterned SiO<sub>2</sub> substrate by selective area MOCVD. The inset shows an array of nanowires separated by 2 µm (a spacing of 20 µm was used for the optical experiments). (b) TEM image clearly showing the formation of a single QD near the tip of a single nanowire. (c) Schematic of a nanowire containing a single QD.

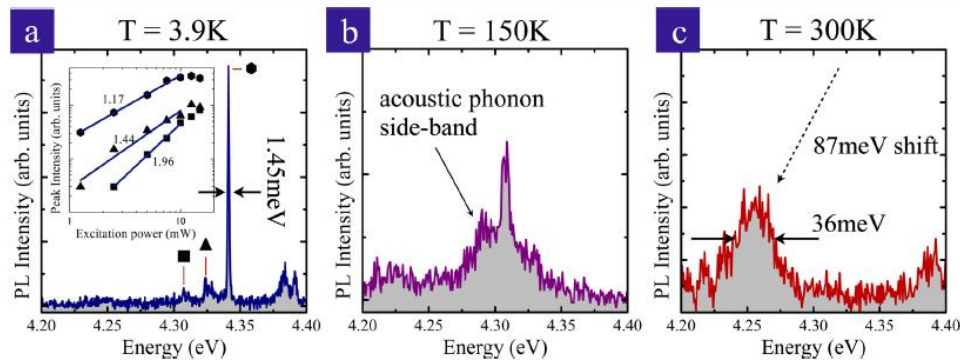


Figure 2. Emission spectra from a single nanowire-QD measured at 3.9 K (a), 150 K (b), and 300 K (c) (measured with excitation powers of 5 mW, 10 mW, and 15 mW, respectively). The inset in (a) shows the power dependence of the peak intensity of the three peaks labeled with a hexagon, triangle, and square, respectively.

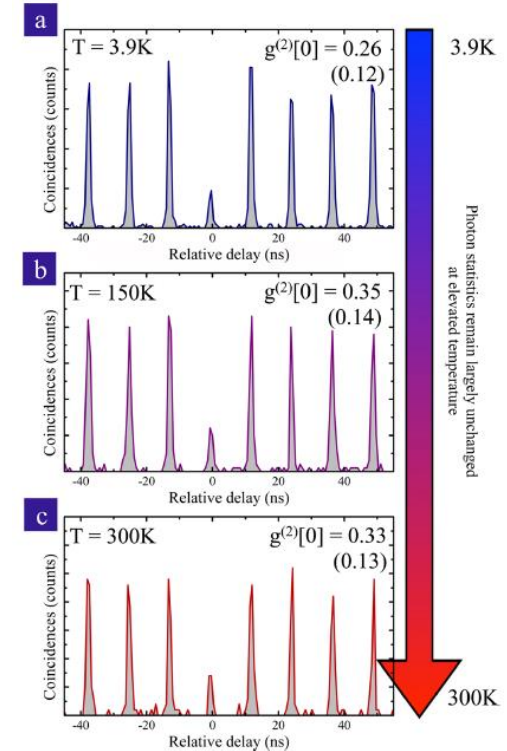


Figure 3. Coincidence counts histograms measured at 3.9 K (a), 150 K (b), and 300 K (c) from the same QD in Figure 2. The  $g^{(2)}[0]$  values in parentheses are corrected for background and detector dark counts. The suppressed height of the peak at delay 0 is clear evidence of single photon emission, and the single photon statistics remain largely unaffected by temperature.

M. J. Holmes, K. Choi, S. Kako, M. Arita, and Yasuhiko Arakawa, *Nanoletters* **14**, 982 (2014)

# Summary

- Focus on different **white light sources** and more specifically on **light emitting diodes**.
- Fundamentals of **electronic** and **optical properties** of **semiconductors** and their **quantum heterostructures**.
- **Optics** as a **sensitive probe** of matter.
- Story of the **blue LED** as an illustration of different technological steps to be taken.



# Summary

- The three laureates demonstrated **intuition** and **perseverance** that lead to a **revolution** in **lighting** and **light display**.
- The mass **marketing** of nitride based lighting devices did **not kill** the **fundamental research** related to these materials.
- The success of the blue diode created a **dynamics** that brought a class of materials to **maturity** and created **new perspectives** in **fundamental** and **applied research**.